Band Engineering in Thermoelectric Materials Using Optical, Electronic, and Ab-Initio Computed Properties

> Thesis by Zachary M. Gibbs

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Abstract

Thermoelectric materials have demanded a significant amount of attention for their ability to convert waste heat directly to electricity with no moving parts. A resurgence in thermoelectrics research has led to significant enhancements in the thermoelectric figure of merit, zT, even for materials that were already well studied. This thesis approaches thermoelectric zT optimization by developing a detailed understanding of the electronic structure using a combination of electronic/thermoelectric properties, optical properties, and ab-initio computed electronic band structures. This is accomplished by applying these techniques to three important classes of thermoelectric materials: IV-VI materials (the lead chalcogenides), Half-Heusler's (XNiSn where X=Zr, Ti, Hf), and CoSb₃ skutterudites.

In the IV-VI materials (PbTe, PbSe, PbS) I present a shifting temperature-dependent optical absorption edge which correlates well to the computed *ab-initio* molecular dynamics result. Contrary to prior literature that suggests convergence of the primary and secondary bands at 400 K, I suggest a higher convergence temperature of 700, 900, and 1000 K for PbTe, PbSe, and PbS, respectively. This finding can help guide electronic properties modelling by providing a concrete value for the band gap and valence band offset as a function of temperature.

Another important thermoelectric material, ZrNiSn (half-Heusler), is analyzed for both its optical and electronic properties; transport properties indicate a largely different band gap depending on whether the material is doped n-type or p-type. By measuring and reporting the optical band gap value of 0.13 eV, I resolve the discrepancy in the gap calculated from electronic properties (maximum Seebeck and resistivity) by correlating these estimates to the electron-to-hole weighted mobility ratio, *A*, in narrow gap materials (*A* is found to be approximately 5.0 in ZrNiSn).

I also show that CoSb₃ contains multiple conduction bands that contribute to the thermoelectric properties. These bands are also observed to shift towards each other with temperature, eventually reaching effective convergence for T>500 K. This implies that the electronic structure in CoSb₃ is critically important (and possibly engineerable) with regards to its high thermoelectric figure of merit.

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List of Symbols and Notation

Fundamental Constants

- *e* Fundamental electron charge
- \hbar Planck's constant (*h*) divided by 2π
- m_e Electron rest mass
- k_B Boltzmann constant
- c Speed of light

Transport Properties

- *V* Voltage
- *R*-Resistance
- *I* Electric Current
- V_H Hall voltage
- *B* Magnetic Field
- *n* Carrier concentration (electrons)
- *p* Carrier concentration (holes)
- R_H Hall coefficient
- r_H Hall factor (n/n_H)
- n_H Carrier concentration $(\frac{1}{R_H e})$
- σ Electrical conductivity (*ne* μ)
- ρ Electrical resistivity $(1/\sigma)$
- μ Carrier drift mobility
- μ_H Hall Mobility (σR_H)
- τ Scattering time
- λ Scattering coefficient (exponent)
- v Drift velocity of electrons
- *S* Seebeck coefficient
- |S| Thermopower (magnitude of S)
- *L* Lorenz number
- κ Total thermal conductivity
- κ_e Electronic thermal conductivity ($L\sigma T + \kappa_{bipolar}$)
- $\kappa_{bipolar}$ Bipolar thermal conductivity
- C_p Heat capacity (constant pressure)
- d Density
- *D* Thermal diffusivity
- *zT* –Thermoelectric figure of merit
- $S^2 \sigma$ –Thermoelectric power factor
- L^* Sample dimension

- *A*^{*} Sample cross sectional area
- *t* Sample thickness
- α_L Linear thermal expansion coefficient
- K_B Bulk modulus

Band Engineering

- v_l Longitudinal speed of sound
- E_{def} Deformation potential
- C_l Average longitudinal elastic moduli $(v_l^2 d)$
- τ_0 Scattering time prefactor
- μ_0 Mobility prefactor ($e\tau_0/m_c^*$)
- n_0 Carrier concentration prefactor
- σ_0 Conductivity prefactor $(n_0 e \mu_0)$
- m^* Effective mass
- m_c^* Conductivity effective mass
- m_d^* Density of states effective mass
- m_b^* Single valley effective mass
- m_P^* Effective mass obtained from electronic momentum
- m_s^* Effective mass obtained from Seebeck coefficient (single parabolic band)
- m_{\parallel}^* Effective mass along the longitudinal ellipsoid direction
- m_{\perp}^* Effective mass along the transverse ellipsoid direction
- $\mu_0 m_S^{*3/2}$ Weighted mobility

• *B*- Thermoelectric quality factor
$$\left(\frac{\mu_0 m_S^{*\frac{3}{2}}}{\kappa_L}\right)^{\frac{3}{2}}$$

- A Majority-to-minority carrier weighted mobility ratio
- N_v Valley degeneracy
- N_{v}^{*} Effective valley degeneracy
- K Ellipsoidal mass anisotropy parameter $(\frac{m_{\parallel}}{m^*})$
- K^* Effective anisotropy parameter
- f Fermi distribution function
- F_i Fermi integral of order j
- ${}^{n}F_{l}^{m}$ Generalized Fermi integral
- E_g Band gap
- $E_{g,thermal}$ Thermal band gap
- ΔE Band offset
- T_{cvg} Band convergence temperature
- β Dimensionless Kane band parameter ($\approx k_B T/E_a$)
- m_0^* Band edge effective mass (Kane bands)
- *P*-Kane band interaction matrix element

- D(E) Density of states
- N_A Number of acceptors
- N_d Number of donors
- s_i Sign of the charge carriers in the i^{th} band
- E Electron energy
- ϵ Dimensionless energy (E/k_BT)
- ξ Chemical potential
- E_F Fermi level
- η Dimensionless chemical potential (ξ/k_BT)
- A_k Fermi surface area
- V_k Volume enclosed by the Fermi surface

Optical Properties

- F(R) Kubelka Munk Function, equivalently: $\frac{\alpha}{k}$
- α Absorption coefficient
- \widetilde{K} Scattering coefficient
- ω Photon frequency
- $\hbar \omega$ Photon energy
- ϵ_c Complex dielectric constant ($\epsilon_1 + i\epsilon_2$)
- ϵ_1 Real dielectric constant
- ϵ_2 Imaginary dielectric constant
- n_r Real part of the refractive index
- κ_r Imaginary part of the refractive index
- **E** Electric field
- ω_p Plasma frequency
- ϵ_{∞} Screening dielectric constant
- $E_{g,opt}$ Optical band gap
- Δ_{RN} Band gap renormalization