### Thermodynamic Properties of Organometallic Dihydrogen Complexes for Hydrogen Storage Applications

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This work is dedicated to my family...

Steel is shaped by the hammer and anvil, but it is the care and patience of the smith that make

the shape meaningful.

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## Abstract

Hydrogen gas has been lauded as a clean fuel technology for its direct combustion to water vapor and its potential for production from renewable fuel sources. However, as a lightweight gas the low volumetric density of hydrogen makes it inappropriate for use in most fuel applications. Hydrogen storage technologies that promote the compression of the gas to useable volumes are a critical component to further development of a hydrogen fuel-based economy.

A number of solid-state storage systems based on metal and complex hydrides, chemical hydrides, and high-surface-area materials have been developed for use as storage media for hydrogen fuel. However, these technologies, which are separated broadly into absorption and adsorption complexes, display typical binding enthalpies too high or too low for practical applications, respectively. In recent years, new complexes designated as Kubas materials containing unsaturated metal centers and high surface areas have been produced and show binding energies intermediate to absorption and adsorption complexes while maintaining beneficial kinetics and binding capacities, demonstrating great promise as hydrogen storage candidates. This behavior has been attributed to the Kubas bond, which binds  $H_2$  chemically as a molecule to an unsaturated metal center based on the Dewar-Chatt-Duncanson molecular binding model. However, the bulk absorption mechanism and thermodynamic behaviors for these interactions remain unknown. Kubas materials have historically been evaluated as physisorption materials for hydrogen storage applications; however, the recent display of absorption plateaus and exact stoichiometric values in isotherm measurements has led to uncertainty in the use of the physisorption model to explain the bulk interaction mechanism, which is necessary to accurately assess thermodynamic properties.

The bulk thermodynamic mechanism of hydrogen binding in dihydrogen complexes, which dis-

play "pure" Kubas bonds, is also poorly understood in the solid state. Identifying the mechanism of Kubas bonding can lead to improved thermodynamic characterization of both dihydrogen complexes and new generation hydrogen storage materials. To address this issue, the solid-state complex  $[Mn(CO)dppe_2][BArF^{24}]$  was synthesized and the thermodynamic behavior and properties of the hydrogen absorption reaction to form the dihydrogen complex  $[Mn(\eta^2-H_2)dppe_2][BArF^{24}]$ were measured over the temperature range 313K-373K and pressure range 0-600 torr using the Sieverts method. The absorption behavior was accurately described by Langmuir isotherms, and enthalpy and entropy values of  $\Delta H^{\circ} = -52.2$  kJ/mol and  $\Delta S^{\circ} = -99.6$  J/mol-K for the absorption reaction were obtained from the Langmuir equilibrium constant. The observed binding strength was similar to metal hydrides and other organometallic complexes, despite rapid kinetics suggesting a site-binding mechanism similar to physisorption materials. Electronic structure calculations using the LANL2DZ-ECP basis set were performed for hydrogen absorption over the organometallic fragments  $[M(CO)dppe_2]^+$  (M = Mn, Tc, Re). Langmuir isotherms derived from calculation for absorption onto the manganese fragment successfully simulated both the pressure-composition behavior and thermodynamic properties obtained from experiment. Results from calculations for the substitution of the metal center reproduced qualitative binding strength trends of 5d > 3d > 4dpreviously reported for the group 6 metals.

Solid-state complexes [FeHdppe<sub>2</sub>][NTf<sub>2</sub>] and [FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][NTf<sub>2</sub>] were also synthesized and measured by x-ray diffraction and Mössbauer spectroscopy to determine the thermal decomposition pathway. Simulations at the B3LYP/TZV level of theory and experimental Mössbauer spectra confirmed the direct thermal decomposition from singlet state [FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][NTf<sub>2</sub>] to triplet state [FeHdppe<sub>2</sub>][NTf<sub>2</sub>] under vacuum conditions at 398K. Sieverts analysis showed samples to have extremely rapid H<sub>2</sub> uptake rates and significantly slower decomposition rates, with an Arrhenius activation energy of  $E_a = 32.3$  kJ/mol for hydrogen release. Evaluation of the partial quadrupole splitting values from Mössbauer spectroscopy revealed a value of  $Q(H_2) = -0.245$  mm/s, similar to the values obtained by Morris and Schlaf for similar complexes and significantly different from values obtained for hydrides, indicating an underutilized mechanism for identification of dihydrogen ligands. Langmuir simulations of hydrogen absorption onto the singlet-state complexes  $[MHdppe_2]^+$  (M = Fe, Ru, Os) revealed the same 5d > 3d > 4d trend and similar enthalpies to those observed for the group 7 complexes. Singlet-state thermodynamic values from simulation were consistent with experimental observations for Ru and Os, and ruthenium complexes were found to have thermodynamic properties within appropriate ranges for hydrogen storage applications. Simulated thermodynamic values for Fe complexes were found to significantly underestimate experimental behavior, demonstrating the importance of the magnetic spin state of the molecule to hydrogen binding properties.

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## Chapter 1

# Introduction to Hydrogen Storage Technologies

With high gravimetric energy density, clean combustion, and renewable production capability, hydrogen gas shows great potential as a sustainable alternative to fossil fuels for energy production. This potential has led to a worldwide development effort for hydrogen technology to power industrial, residential and transportation infrastructure—a concept known as the hydrogen economy.<sup>1</sup> Mitigation of fossil fuel usage for transportation applications has dominated research efforts, with development of fuel cell-based powertrains for light-duty vehicles as the primary focus.<sup>2</sup> A number of unique technological challenges are inherent to these systems, including the efficient, safe storage of sufficient volumes of hydrogen gas to provide a fuel cell vehicle with comparable range to modern gasoline vehicles. Improvement of the volumetric energy density of the fuel storage system remains one of the largest hurdles to commercial implementation of fuel cell vehicles, and significant research efforts have been employed over the last decade to overcome this challenge.

Solid-state storage materials are currently the primary focus of research into storage of hydrogen gas for fuel systems.<sup>3</sup> In these materials, hydrogen gas is reversibly bound through chemical or physical interactions, increasing the volumetric energy density of the fuel, and correspondingly the fuel tank capacity, while maintaining safe operating temperatures and pressures. Multiple classes of solid-state storage materials have been identified based on the properties of their interaction with hydrogen, with numerous potential materials in each class, and research activities have been primarily focused on identification of new storage materials and optimization of the identified systems for energy efficiency. These efforts have led to significant advancements in the kinetics, thermodynamics, and capacity of solid hydrogen storage systems, but so far have not been able to meet the stringent requirements for commercial systems<sup>1,2,4</sup>

Continuing need for commercial hydrogen storage systems and concern that current materials have met intrinsic limits have provided motivation for parallel investigations into new classes of materials that exhibit unique binding motifs with hydrogen gas.<sup>5</sup> This chapter reviews the thermodynamic requirements of vehicular hydrogen storage systems and the current solid-state hydrogenstorage technologies. The advantages and limitations of each class of materials is discussed, and the motivation for investigations into an alternative class of storage materials exhibiting Kubas binding, where a dihydrogen molecule is bound to an organometallic complex as a ligand, is presented.

### 1.1 Background

Solid-state hydrogen-storage materials are broadly divided into two classes: absorption materials and physisorption materials.<sup>1</sup> These class distinctions are based on the location of the hydrogensolid interaction; in absorption materials, hydrogen atoms dissociate at the solid surface and are subsequently bound within the bulk of the solid material, whereas in physisorption materials molecular hydrogen binds to the material surfaces. These class distinctions also roughly divide storage materials by binding strength, although some overlap can exist between their ranges. Absorption materials are characterized by strong chemical bonds between hydrogen atoms and atoms of the substrate. Physisorption materials bind hydrogen through weak van der Waals interactions, and hydrogen molecules maintain their molecular bond. Each of these broad classes is divided into several subclasses of materials, representing hundreds of potential storage materials with a wide range of properties.

Approaching the development and evaluation of these materials requires a practical method for screening promising candidates from the hundreds of potential materials available. Recent hydrogen storage research has focused on vehicular applications, directed by the US Department of Energy's Fuel Cell Technologies Program that, in 2005, introduced a research plan and series of engineering criteria to implement commercialization of fuel storage and delivery systems in hydrogen-fueled vehicles.<sup>4</sup> These criteria cover a wide range of economic, engineering, and material design specifications but mostly fall outside of the scope of basic research. However, several criteria have historically been used to direct basic hydrogen storage research,<sup>2</sup> and a review of the role of these criteria in choosing materials for study is necessary to understand the current state-of-the-art in the field.

Five of the criteria established by the US Department of Energy are commonly implemented in basic materials research: gravimetric energy density, volumetric energy density, cycling life, operating pressure and temperature, and filling kinetics. Gravimetric energy density and volumetric energy density depend upon the material structure and chemical makeup. Cycling life, operating conditions, and filling kinetics depend upon reactions within the material and the energy transfer between the solid and hydrogen gas during absorption. Procedures for evaluating these properties vary according to the binding mechanism, with different techniques for absorption materials and physisorption materials to accommodate the differences in the physical interactions within these materials.

The USDOE target ranges<sup>4</sup> for hydrogen delivery conditions to a fuel cell are 3–100 bar and 233K– 358K. These values directly correlate to the thermodynamic properties of the reaction between the solid storage medium and the hydrogen atmosphere, and the evaluation of the standard enthalpy,  $\Delta H^{\circ}$ , and standard entropy,  $\Delta S^{\circ}$ , of hydrogen absorption is the primary goal of research into hydrogen storage materials.

### **1.2** Evaluation of Thermodynamic Properties

Experimental evaluation of thermodynamic data for hydrogen storage materials is commonly performed through the collection of isothermal absorption profiles using a Sieverts apparatus.<sup>6,7</sup> A schematic of a Sieverts apparatus appears in Figure 1.1. The instrument consists of a series of rigid calibrated volumes attached to a hydrogen source and a calibrated reaction chamber containing the material under investigation. During a measurement, hydrogen gas is introduced into an evacuated calibrated volume and allowed to equilibrate while pressure and temperature are recorded, and a value for the number of moles of hydrogen in the volume is obtained using the appropriate equation of state. The calibrated volume is then opened to the evacuated reaction chamber, and hydrogen is allowed to react with the storage material. The pressure and temperature are again recorded, and the change in moles between the original measurement and the measurement after exposure to the storage material is recorded as the number of moles absorbed by the sample, giving a single pressure-temperature-composition (P-T-x) data point.

Single point isotherms are commonly reported,<sup>8–10</sup> particularly for materials that do not have strong reversibility or when simultaneous evaluation of thermodynamics and analysis of the kinetics is desired. While thermodynamic information may be extracted from these isotherms, for many materials it is difficult to differentiate between solubility of  $H_2$  in the material and reactive absorption from them, and the incorrect treatment of the physical process can sometimes be used. More complete iterative isotherms are comprised of multiple single point isotherm P-T-x data points, created by serial exposures of the material to new hydrogen pressures. Kinetic information is sacrificed to produce these isotherms, but a more complete expression of the absorption properties and thermodynamics is obtained. Thermodynamic information is obtained from the comparison of isotherms at multiple temperatures, using appropriate equations for the expected absorption mechanism.<sup>11–13</sup>

Thermogravimetric analysis and differential scanning calorimetry, respectively, are also used for evaluating the gravimetric capacity and thermodynamic properties of hydrogen storage materials.<sup>14,15</sup> The temperature of the hydrogen storage material is raised at a constant rate under a flow of inert gas, and the mass of the sample and heat flow required to maintain the rate of increase relative to a standard are recorded. Loss of mass during the experiment, identifiable as  $H_2$ gas by mass spectroscopy, indicates the desorption capacity of the material. Changes in the heat flux necessary to maintain the temperature ramp indicate the onset of the chemical reaction, and integration of the heat flux curves provides the enthalpy of the interaction. These techniques are accurate, rapid, relatively easy to perform, and require very little sample material. However, a hydrogenated material that is stable at room temperature is required, and the techniques do not allow measurements at multiple pressures. As such, these methods are inappropriate for certain types of absorption materials and for all currently known physisorption materials.



Figure 1.1: Schematic diagram of a Sieverts apparatus. Calibrated volumes are filled from a hydrogen source through a valved manifold (also calibrated) and exposed to the temperature controlled reactor for absorption measurements. Multiple calibrated volumes are used to match the expected pressure drop from absorption in the reactor. Argon expansion is used to calibrate the volumes of the reactor. Pressure and temperature measurements are obtained on the reactor and on each of the available volumes.

Electronic structure calculations are used to estimate the thermodynamic binding properties in hydrogen storage materials, most commonly with storage materials that exhibit hydrogen dissociation or chemical reaction.<sup>16, 17</sup> Density functional theory (DFT) calculations simulate the geometry and electronic energy of each component of the absorption reaction. The energies are then summed to produce a reaction energy that approximates the standard enthalpy of the reaction. This powerful technique is useful for screening potential candidates and, more broadly, mixtures of potential candidates that fall within the desired thermodynamic range.

Analysis of the data to obtain thermodynamic information is dependent on the expected interaction mechanism. Properties of absorption materials and physisorption materials, calculation methods for their thermodynamic parameters and state-of-the-art materials for each group are reviewed below.

#### **1.3** Properties of Absorption Materials

Absorption hydrogen storage materials undergo a chemical reaction with hydrogen gas, causing dissociation of the gas molecule at the material surface and migration of atomic hydrogen within the material structure.<sup>3</sup> The atomic hydrogen may then occupy interstitial sites in the crystalline solid material, as in transition metal hydrides, or form new chemical bonds between the hydrogen atoms and the atoms of the substrate. Generally, this process can be described by the chemical equation:

$$S + nH_2 \rightleftharpoons SH_{2n}$$
 (1.1)

This category of materials includes alkali and alkaline earth metal hydrides (e.g., LiH, NaH, MgH<sub>2</sub>, CaH<sub>2</sub>), transition metal hydrides (e.g., PdH<sub>0.6</sub>, LaNi<sub>5</sub>H<sub>6</sub>), alanes (e.g., AlH<sub>3</sub>, NaAlH<sub>4</sub>), boranes (e.g., LiBH<sub>4</sub>, NaBH<sub>4</sub>, NH<sub>3</sub>BH<sub>3</sub>), solid hydrocarbons, and amines and imines (e.g., LiNH<sub>2</sub>, Li<sub>2</sub>NH). These materials are primarily characterized by high gravimetric and volumetric energy densities, high binding enthalpies, slow solid-state kinetics, and moderate to poor cycling behavior.<sup>2, 1</sup>

Gravimetric energy density is the easiest of these properties to evaluate and is used extensively in screening potential absorption materials.<sup>17</sup> Except for transition metal hydrides, hydrogen binds to these materials in fixed stoichiometric ratios and the theoretical gravimetric energy density is directly obtainable from the chemical formula of the hydrogenated material. For example, in the absorption equation

#### $2\mathrm{Li} + \mathrm{H}_2 \longrightarrow 2\mathrm{LiH}$

the gravimetric energy density of LiH is obtained as the product of the standard lower heating value of hydrogen gas, 120 kJ/gram H<sub>2</sub>, and the weight ratio of hydrogen in lithium hydride as obtained from the molecular weights, 0.1268 grams H/gram LiH (MW(H) = 1.0079, MW(LiH) =

7.9489). Because the lower heating value is a standard, gravimetric energy density is often reported as the weight ratio or weight percentage of hydrogen in the material. However, for transition metal hydrides the gravimetric capacity does not necessarily follow stoichiometric rules and must be obtained experimentally, although rules of thumb based on the crystal structure exist.<sup>18</sup> The USDOE target equivalent of 4.5–7.5% wt storage has mostly limited examination of new absorption hydrogen storage materials to those comprised of elements from the first three rows of the periodic table. These limitations allow for lightweight and compact storage materials, but often at the expense of thermodynamic or kinetic efficiency.

Volumetric energy density for absorption materials is determined from the unit cell volume of the crystalline hydrogenated product. This information is also easily obtainable and tabulated from x-ray diffraction (XRD) measurements<sup>19</sup> or DFT geometry optimizations,<sup>20</sup> and is used as an additional screening parameter along with weight percentage. For most absorption materials, volumetric energy density far exceeds the targets set by the USDOE. Because of this, weight percentage of hydrogen is favored in reports.

#### **1.3.1** Absorption Thermodynamics

Thermodynamic properties of absorption hydrogen storage materials are typically evaluated through the ideal gas absorption van't Hoff equation<sup>1</sup>

$$\ln\left(\frac{P}{P_0}\right) = \frac{\Delta H_{rxn}^{\circ}}{RT} - \frac{\Delta S_{rxn}^{\circ}}{R}$$
(1.2)

where P and T are the pressure and temperature of the system,  $P_0$  is the standard or reference pressure of the system (typically 1 atm or 1 bar), R is the universal gas constant, and  $\Delta H^{\circ}$  and  $\Delta S^{\circ}$  are the enthalpy and entropy of the absorption reaction at standard pressure and temperature. This relationship is developed from the equivalence of free energy between the products and reactants in Equation 1.1 during chemical equilibrium. Given the equivalent thermodynamic system of the Sieverts reactor, schematically illustrated in Figure 1.2, the equilibrium condition between the gas



Figure 1.2: Schematic of the general thermodynamic system for hydrogen storage measurements phase and the condensed phases can be expressed mathematically as

$$G_{solid} - G_{gas} = 0 \tag{1.3a}$$

$$\partial G_{solid} - \partial G_{gas} = 0 \tag{1.3b}$$

where  $G_{solid}$  and  $G_{gas}$  represent the total molar Gibbs free energy of the solid and gas phases, respectively. This definition of equilibrium for absorption systems is universal and independent of the absorption mechanism. For absorption systems, Equation 1.3a is expanded with respect to the absorption portion of Equation 1.1, giving

$$G_{\rm SH_{2n}} - G_{\rm S} - nG_{\rm H_2} = 0$$

where  $G_{SH_{2n}}$ ,  $G_{H_2}$  and  $G_S$  are the molar Gibbs free energy reaction components. To apply the equation to isotherms, the temperature is assumed to be fixed and the standard molar Gibbs free energy is defined at the fixed temperature and a standard pressure,  $P^{\circ}$ . Separating the Gibbs free energies into standard and non-standard components gives

$$\left(G_{\mathrm{SH}_{2n}}^{\circ} - G_{\mathrm{S}}^{\circ} - nG_{\mathrm{H}_{2}}^{\circ}\right) + \int_{P_{0}}^{P} \left[ \left(\frac{dG_{\mathrm{SH}_{2n}}}{dP}\right)_{T} - \left(\frac{dG_{\mathrm{S}}}{dP}\right)_{T} - n\left(\frac{dG_{\mathrm{H}_{2}}}{dP}\right)_{T} \right] dP = 0$$

where  $G^{\circ}$  is the standard molar Gibbs free energy, T, P are the isothermal temperature and pressure of the system, and P<sub>0</sub> is the standard pressure of the system. This equation completely describes the equilibrium condition for absorption in absorption materials for any conditions. However, to obtain the van't Hoff equation given in Equation 1.2, a few assumptions regarding the absorption conditions must be made.

Solid phases in the reaction are generally assumed to be incompressible, such that their free energy has no dependence on the system pressure. This assumption takes advantage of the large values of the bulk modulus for hard condensed matter to simplify the integral and eliminate the terms containing the solids. For hydrides, complex hydrides, ammonia borane, and other small molecule, densely packing absorption material, this assumption will be valid across a wide range of pressures and can safely be made within the 0–100 bar pressure range specified in the USDOE criteria. However, for materials consisting of polymers and other soft condensed matter this assumption may not hold. With this assumption, the equation becomes

$$\Delta G_{\rm rxn}^{\circ} = \int_{P_0}^{P} \left(\frac{dG_{\rm H_2}}{dP}\right)_T dP$$
$$n\Delta G_{\rm rxn}^{\circ} = G_{\rm SH_{2n}}^{\circ} - G_{\rm S}^{\circ} - nG_{\rm H_2}^{\circ}$$

with  $\Delta G_{rxn}^{\circ}$  defining the change in the standard molar Gibbs free energy of the absorption reaction, normalized to the moles of hydrogen gas. The constant temperature derivative of the molar Gibbs free energy of a single component gas with respect to pressure is the thermodynamic molar volume:

$$\Delta G^{\circ}_{\rm rxn} = \int\limits_{P_0}^{P} V_{\rm H_2} dP$$

which can be related to the system temperature and pressure through an appropriate equation of state. The second assumption applies the ideal gas equation of state to evaluate the molar volume



Figure 1.3: Fugacity coefficient of hydrogen at relevant temperatures and pressures

and obtain expressions based on temperature and pressure. Substituting the ideal gas equation in for the molar volume gives

$$\Delta G_{\rm rxn}^{\circ} = \int_{P_0}^{P} \frac{RT}{P} dP$$
$$\Delta G_{\rm rxn}^{\circ} = RT \ln\left(\frac{P}{P_0}\right)$$

Rearranging and separating the Gibbs free energy into enthalpy and entropy components gives the van't Hoff equation, Equation 1.2, for hydrogen absorption. The derivation above and Equation 1.2 are considered from the perspective of absorption; the signs on the enthalpy and entropy terms are reversed for desorption reactions.

The correct application of this equation to hydrogen storage systems requires hydrogen to be within the range of ideal gas behavior. The deviation of hydrogen from ideal gas behavior can be estimated by calculation of the fugacity coefficient for hydrogen at various temperatures and pressures:

$$\phi = \exp\left\{\frac{G_{\mathrm{H}_2}(T, P) - G_{\mathrm{H}_2}^{\mathrm{IG}}(T, P)}{RT}\right\} = \exp\left\{\frac{1}{RT}\int_0^P \left(V_{\mathrm{H}_2} - \frac{RT}{P}\right)\partial P\right\}$$

where the superscript IG denotes the equivalent ideal gas properties. A value of one for  $\phi$  denotes that the gas behaves as an ideal gas, and deviations from a value of one represent non-idealities. These values are derived from direct measurements of the hydrogen equation of state, and have been compiled for a range of pressures and temperatures by the National Institutes of Science and Technology (NIST).<sup>21</sup> A graph of hydrogen gas fugacity coefficients at temperatures and pressures relevant for hydrogen storage appears in Figure 1.3. For pressures below 100 bar, the fugacity coefficient for hydrogen is a weak function of pressure and is a weak function of temperature above 196K, which has been shown in the past to be sufficient for use of the van't Hoff equation in evaluating absorption hydrogen storage materials to meet the USDOE requirements. For basic research involving higher pressures or lower temperatures, however, the values may deviate significantly, and Equation 1.2 would be inappropriate to describe the absorption system.

The pressures and temperatures used in the van't Hoff equation are the values that correspond to chemical equilibrium between the gas and solid phases. Experimentally, these values must be obtained from pressure-composition isotherms. For absorption processes, this equilibrium is easily determined; isotherms for absorption processes demonstrate a flat plateau, illustrated schematically in Figure 1.4, which corresponds to the equilibrium conditions of the reaction where the change of the Gibbs free energy is zero as the reactants are converted to products.<sup>1</sup> The flat shape of the plateau is characteristic of a reaction-dominated process, indicating independence of the Gibbs free energy of absorption on the composition of the phases.

#### **1.3.2** State-of-the-Art: Absorption Materials

Although no absorption materials have been commercially developed, several material types are under active investigation and are considered state-of-the-art materials for hydrogen absorption. Each material type exhibits several benefits and drawbacks for use with commercial systems. A brief overview of each material type is given below.



Figure 1.4: Idealized pressure-composition isotherm (left) and van't Hoff plot (right) for absorption processes, demonstrating the expected curve-shapes for absorption dominant processes and translation to thermodynamic quantities.

**Borohydrides**—Borohydrides are complex hydride materials which primarily store hydrogen in the form of the tetrahydridoboride anion,<sup>22, 23</sup>  $[BH_4]^-$ . Typically used as mild reducing agents in chemical applications, these materials are lightweight, earth-abundant, and are available on industrial production scales. The complexes primarily under development for hydrogen storage applications are LiBH<sub>4</sub>, NaBH<sub>4</sub>, Mg(BH<sub>4</sub>)<sub>2</sub> and Ca(BH<sub>4</sub>)<sub>2</sub>, with LiBH<sub>4</sub> dominating the literature because of its high weight percentage of hydrogen. NaBH<sub>4</sub> has received less attention in recent years because its primary mechanism for release of hydrogen is through hydrolysis of the borohydride anion, an irreversible process.<sup>24</sup>

Borohydrides suffer from a number of drawbacks that limit their applications as hydrogen storage materials. Thermodynamically, the absorption enthalpy of borohydrides is relatively strong, ranging from 67 kJ/mol for LiBH<sub>4</sub><sup>25</sup> to 53 kJ/mol for CaBH<sub>4</sub>.<sup>26</sup> Materials within this range of enthalpies require extremely high temperatures for hydrogen cycling; often temperatures greater than 550K are required to observe hydrogen release. Such high temperatures also promote side reactions and the

release of unfavorable product gases, primarily diborane<sup>27</sup> ( $B_2H_6$ ), which can foul fuel cell catalysts and surfaces and remove reversible capacity from the material.

Borohydrides also suffer from poor solid-state diffusion kinetics and a complex decomposition chain, which leads to multiple side reactions and the formation of highly stable side and end products, notably metal borides and materials containing the  $[B_{12}H_{12}]^{2-}$  anion.<sup>28</sup> Both of these features lead to poor reversibility of the desorption reaction and effective prevention of true cycling behavior. Recent research efforts have aimed at control of the size of borohydride crystallite sizes and confinement of crystallites to nanocavities,<sup>29</sup> as well as using catalytic additives,<sup>30</sup> in the hopes of limiting diffusion and promoting reversible hydrogen release at lower temperatures.

Alane/Alanates—Both the neutral alane,<sup>31</sup> AlH<sub>3</sub>, and alanate salts comprised of the tetrahydridoaluminate anion,<sup>11</sup> [AlH<sub>4</sub>]<sup>+</sup>, have been consistent candidates for hydrogen storage materials since the discovery that titanium and organotitanium catalyzes the release and reabsorption of hydrogen from sodium alanate. Neutral AlH<sub>3</sub> is thermodynamically unstable to hydrogen decomposition at room temperature, with an absorption enthalpy of only -14.85 kJ/mol H<sub>2</sub>, which corresponds to an equilibrium pressure of 5 x 10<sup>5</sup> bar at room temperature.<sup>32</sup> Alane is kinetically stabilized in the solid state, only decomposing at approximately 440K in a single step to give hydrogen and aluminum powder. Once decomposed, the resulting aluminum powder is highly resistant to reabsorption of hydrogen to regenerate alane. In situ, alane is thus considered a single-use material that must be removed and regenerated through chemical means. Ongoing research into alane as a hydrogen storage material focuses on safely controlling the release kinetics of hydrogen from the material, and methods to efficiently regenerate the hydrogenated material from spent aluminum powder.<sup>33</sup>

Alanate salts, particularly  $NaAlH_4$  and  $LiAlH_4$ , have been more widely reviewed for hydrogen storage applications. Both materials decompose by a two-step pathway, typified by the equations

$$3$$
NaAlH<sub>4</sub>  $\implies$  Na<sub>3</sub>AlH<sub>6</sub> + 2Al + 3H<sub>2</sub>  
Na<sub>3</sub>AlH<sub>6</sub>  $\implies$   $3$ NaH + Al +  $3/2$ H<sub>2</sub>

For the sodium analog, both reactions have been shown to be reversible when titanium is present,

with mild enthalpies of 37 kJ/mol and 47 kJ/mol, respectively.<sup>34</sup> However, because aluminum must be mobile within the phases for the decomposition to occur, the kinetics of desorption are slow. Titanium doping and the introduction of other impurities have been shown to improve the kinetics to reasonable levels in sodium alanate, making it one of the few complex hydrides to demonstrate reversible kinetics near room temperature. However, a combined reversible weight percentage of only 5.6 wt% for NaAlH<sub>4</sub> has prompted studies into the lithium analog, which decomposes through the same mechanism. LiAlH<sub>4</sub> was found to be considerably less stable, with initiation of desorption at 453K and extremely high absorption pressures, making reversible hydrogen storage extremely difficult.<sup>35</sup>

 $NaAlH_4$  continues to be explored as a potential vehicle fuel system material, and current research on the material is concerned with engineering properties the material.<sup>36,37</sup> Investigations into LiAlH<sub>4</sub> and other alanates are primarily concerned with increasing the stability of the compound and reversibility of the desorption.<sup>38,39</sup>

**Amides/Imides**—Lithium nitride/lithium amide has been one of the more promising materials systems for hydrogen storage since it was first investigated by Chen, et al.<sup>40</sup> Absorption of hydrogen over lithium nitride has been shown to occur in a two-step, reversible process:

$$\text{Li}_3\text{N} + \text{H}_2 \Longrightarrow \text{Li}_2\text{NH} + \text{LiH}$$
  
 $\text{Li}_2\text{NH} + \text{H}_2 \Longrightarrow \text{LiNH}_2 + \text{LiH}$ 

These two reactions give absorption of 5.5 wt% and 6.5 wt%, respectively, exhibiting comparable absorption capacity to NaAlH<sub>4</sub> with fast absorption kinetics. However, the first step was found to be irreversible, and the reversible absorption enthalpy of -66.1 kJ/mol for the imide requires temperatures greater than 673K, preventing practical use of this system. Experimental investigations of the system have also found significant quantities of ammonia<sup>14</sup> can be released, through the self-decomposition reaction of lithium amide:

$$2\text{LiNH}_2 \longrightarrow \text{Li}_2\text{NH} + \text{NH}_3$$

This ammonia release is severely detrimental to the performance of the material in fuel systems; ammonia serves to poison fuel cell catalysts and drastically reduces the lifetime of a fuel system. Ammonia release can be prevented by the presence of LiH throughout the structure to absorb the molecule to form  $\text{LiNH}_2$  and  $\text{H}_2$ . However, this kinetic effect makes the mixing of starting materials and prevention of chemical segregation crucial to the performance of the material, and unfortunately any efforts to maintain these properties cannot be expected to be maintained upon cycling.

Other amide systems, based on the lithium nitride system, are being investigated for improvement of the thermodynamic behavior and reduction of ammonia release. The  $Mg(NH_2)_2$  + LiH mixture has received attention as a replacement in recent years, showing promise both in gravimetric capacity and thermodynamic properties.<sup>41</sup> This is a form of chemical destabilization, discussed in more detail below.

Ammonia Borane and Amidoboranes—Ammonia borane ( $NH_3BH_3$ ), at 19.6 wt% hydrogen, has received a great deal of attention as a potential hydrogen storage material. Unlike most absorption storage materials, ammonia borane is air stable and release of hydrogen occurs primarily from the molten state just past the melting temperature of 377K.<sup>42</sup> Hydrogen release is also exothermic and self-promoting, with an overall decomposition enthalpy of -21.7 kJ/mol.

The release of hydrogen occurs through a complex polymerization decomposition, with three steps relevant for hydrogen storage:

$$\begin{split} nNH_3BH_3 & \longrightarrow & [NH_2BH_2]_n + nH_2 \\ [NH_2BH_2]_n & \longrightarrow & [NHBH]_n + nH_2 \\ [NH_2BH_2]_n & \longrightarrow & nN_3B_3H_6 + nH_2 \end{split}$$

These reactions create observable pressures between approximately 380K and 473K. After formation of polyaminoborane ( $[NH_2BH_2]_n$ ), the formation of polyiminoborane ( $[NHBH]_n$ ) is exothermic, spontaneous and favorable at lower temperatures. This reaction path is desirable; borazine ( $N_3B_3H_6$ ) formed through the last reaction pathway is a volatile liquid that can be released into the fuel lines, removing storage capacity from the fuel tanks and potentially fouling fuel cell catalysts. However, because the kinetics of hydrogen release are slow at lower temperatures, a compromise between borazine production and release rate often can be made to mitigate the formation of detrimental products.

Initialization of the decomposition is slow, and the decomposition is not directly reversible. Catalytic systems are under development for both improving the release rate and facilitating regeneration of the final product. Iridium organometallic complexes have been shown to be extremely effective at improving the decomposition rate, but are impractical for scale-up.<sup>43</sup> Less expensive options based on 3d transition metals are under investigation to replace iridium complexes and other precious metals in such systems.<sup>44</sup>

The mechanism for removal of the polymeric end products from fuel systems and efficient chemical regeneration of ammonia borane remain the most significant engineering and scientific challenges associated with its use. A number of catalytic regeneration cycles have been proposed, including the use of organotin catalytic cycles<sup>45,46</sup> and regeneration in pressurized, liquefied ammonia with hydrazine.<sup>47</sup> These methods have serious safety and economic hurdles to overcome before being commercially realizable, and ongoing research seeks to improve the engineering properties of these regeneration techniques, as well as further improving the efficiency of regeneration processes and increasing control of kinetic rates of release and gaseous product trapping and recovery.

**Destabilized Materials Mixtures**—One of the common features of absorption storage materials that limits their applications in fuel systems is the high enthalpy of absorption. This feature forces the use of impractically high temperatures to activate hydrogen release from the solid. From Equation 1.2, reduction of the binding enthalpy leads to exponential increases in the available pressure at a given temperature, and making small improvements in the overall system thermodynamics worth pursuing. Some recent work has centered on the use of complex hydride and simple metal hydride mixtures, designed to increase the stability of the desorption products at the sacrifice of some capacity. A prototypical example of this effect is the mixture of LiBH<sub>4</sub> and MgH<sub>2</sub>.<sup>25</sup> Individually, these materials release hydrogen through the reactions

$$LiBH_4 \longrightarrow LiH + B + 3/2H_2$$
$$MgH_2 \longrightarrow Mg + H_2$$

However, as a mixture of 2:1 LiBH<sub>4</sub>:MgH<sub>2</sub>, dehydrogenation occurs via the overall reaction

$${\rm LiBH_4} + \frac{1}{2}{\rm MgH_2} \longrightarrow \ \frac{1}{2}{\rm MgB_2} + {\rm LiH} + 2{\rm H_2}$$

The mixture, at 11.54 wt% usable capacity, represents a  $\sim 2\%$  reduction in the absorption capacity from LiBH<sub>4</sub> alone. However, formation of the more stable MgB<sub>2</sub> species reduces the dehydrogenation enthalpy to 40.5 kJ/mol from 67 kJ/mol for LiBH<sub>4</sub> alone and 75 kJ/mol for MgH<sub>2</sub> alone. This reduction leads to a decrease in the 1 bar equilibrium temperature from greater than 673K for LiBH<sub>4</sub> to 498K, representing a significant improvement in the operating properties of the storage material. Millions of such combinations of absorption hydrogen storage materials have been identified from available crystal structures, and DFT calculations have been used to screen these combinations for the highest potential for meeting the USDOE requirements.<sup>16,17</sup> Combinations identified by the theoretical studies are currently undergoing experimental investigations.<sup>48,49,41</sup>

The use of mixtures can lead to reductions in the kinetic rate of absorption and desorption, which can offset the benefits in operating temperature obtained through lower thermodynamics by kinetic stabilization of the hydrogenated products. Solid-state diffusion limitations of larger elements like boron or magnesium can easily dominate the kinetic behavior at lower temperatures and limit the release rate. A balance between the benefits of the thermodynamic destabilization and the kinetic stabilization must therefore be found for the efficient use of destabilized mixtures. Unfortunately, this information is unavailable from current DFT simulations, and must be evaluated experimentally.

Evaluation and improvement of kinetic properties is difficult in experimental mixtures, and additional, non-chemical complications can arise. For example, in the  $LiBH_4 + 1/2MgH_2$  system above, as shown in Figure 1.5, the  $LiBH_4$  melt was found to have poor surface adhesion to  $MgH_2$ , which belies an unfavorable interaction energy between the two materials that disfavors mixing and increases the diffusional barriers. This physical property prevents the formation of classic composites



Figure 1.5:  $LiBH_4$  on a compacted disk of MgH<sub>2</sub>, under an argon atmosphere. a) Powdered  $LiBH_4$  (white) at room temperature. b) The same system at 573K, after melting of the  $LiBH_4$ . Molten  $LiBH_4$  forms a large contact angle with the MgH<sub>2</sub> surface, indicating a poor wetting interaction.

from the suspension of  $MgH_2$  in molten  $LiBH_4$ , and impedes efforts at nanoconfinement and other techniques. Mitigation of such effects is a large area of research in destabilized systems.

### **1.4** Properties of Physisorption Materials

Physisorption storage materials bind hydrogen to the material surface through van der Waals forces.<sup>1</sup> These binding forces, produced by polarization of the interacting molecules through mutual repulsion of their electron clouds and random fluctuations in electron density, occur at any interface between hydrogen gas and a solid material. The bonds formed between hydrogen and the solid in these materials are typically made over long distances and are correspondingly weak and non-chemical in nature, which gives characteristically low binding enthalpies and poor storage capacity at room temperatures. However, these materials are characterized by rapid kinetics, low cost, and long cycle lifetimes. Solids designed to act as physisorption storage materials are treated to maximize the available surface area for these interactions, and thus the amount of hydrogen capable of being stored on the surface.

Evaluation of the USDOE parameters in physisorption materials is similar to the methods for absorption materials. Gravimetric energy density is again the most straightforward of the criteria to



Figure 1.6: Plot of specific surface area vs. gravimetric capacity at saturation for hydrogen absorption over adsorbent materials, demonstrating Chahine's rule of approximately 1 wt% increase in adsorption capacity for every 500 m<sup>2</sup>/g increase in specific surface area. Data obtained from Hirscher and Panella, 2007.<sup>50</sup>

determine from the material properties. The ultimate gravimetric storage capacity of physisorption materials with similar chemical composition has been shown to be directly proportional to the specific surface area of the material.<sup>50</sup> This correlation, known as "Chahine's rule", is shown graphically for various carbon-based materials in Figure 1.6. This allows rapid estimation of the gravimetric capacity from BET or other surface-area tests.

Because gravimetric capacities are typically lower for physisorption materials than for absorption materials, physisorption material chemical compositions are primarily carbon, boron and boron nitride. Limiting the basic chemical composition to these two materials minimizes the density of the material, allowing for the highest gravimetric ratio of the surface hydrogen to the bulk material. Physisorption materials can also incorporate small quantities of other elements to improve specific properties; this technique is primarily used to modify the absorption enthalpy. These additional elements include transition metals,<sup>51</sup> intercalated alkali metals or alkaline earth metals,<sup>52</sup> and fluorine adatoms.<sup>53</sup> Heavier atoms are only added in small amounts, however, to avoid reducing the gravimetric capacity too much. Volumetric capacity in physisorption materials is difficult to estimate or measure. Because hydrogen is only adsorbed onto the surface and not absorbed into the bulk, no true volumetric energy density can be defined for the material. Instead, the volumetric capacity is typically assigned as the improvement in the number of moles stored in a given tank volume when the adsorbent material is added, with appropriate adjustments for the volume displaced by the adsorbent itself.<sup>3</sup> Based on this metric, the volumetric capacity is not an independent quantity, being dependent on both the temperature and pressure of the measurement, and on the gravimetric capacity of the material. Because the volumetric capacity must be derived from measurements, is tank specific, and is not a true physical quantity, it is a less meaningful description of the properties of physisorption materials, and is rarely reported in the literature.

Diffusional barriers are largely absent from physisorption materials, and the weak inherent binding to the surface reduces activation barriers for hydrogen release. Side reactions and modifications of the solid surface during the adsorption process are also largely absent, because no true chemical bonds with the surface are formed. These properties translate into rapid kinetics for hydrogen adsorption and release and high cycling lifetimes, typically far in excess of the USDOE targets. Because of this, kinetic rate measurements are often not performed for hydrogen absorption measurements in basic research. However, these values become important in engineering applications when coupled with heat management requirements. The rapid absorption of gas onto the surface can lead to high thermal fluxes or rapid temperature increases. Both of these properties may be detrimental to the operation of an adsorption system.

#### 1.4.1 Physisorption Thermodynamics

Derivation of the surface thermodynamics for physisorption processes also begin with equivalence of the free energies of gaseous and bound hydrogen, given in Equation 1.3a and Equation 1.3b. However, unlike absorption materials the hydrogen molecule does not lose its identity, and no chemical reaction occurs. Instead, the hydrogen molecule exists in both solid and gaseous states with differing potentials, with the solid-state potential modified by interaction with the substrate. This process parallels phase transition models. Taking this approach, Equation 1.3a and 1.3b may be used directly to arrive at pressure and temperature dependence of the system. For phase equilibrium of a single species, Equation 1.3b can be rewritten as

$$V_{gas}\partial P - S_{gas}\partial T = V_{solid}\partial P - S_{solid}\partial T$$

Where gas and solid subscripts describe hydrogen within those phases. Also, using the definition G = H - TS in Equation 1.3a, the entropy can be related to the enthalpy

$$H_{gas} - TS_{gas} = H_{solid} - TS_{solid}$$

Combining these two equations yields the Clapeyron equation:

$$\left(\frac{\partial P}{\partial T}\right) = \frac{S_{gas} - S_{solid}}{V_{gas} - V_{solid}} = \frac{\Delta H_{des}}{T\Delta V}$$
(1.4)

where  $\Delta H_{des} = H_{gas} - H_{solid}$  is the desorption enthalpy, analogous to the heat of vaporization in liquid-vapor equilibrium, and  $\Delta V = V_{gas} - V_{solid}$  and P is the saturation pressure corresponding to equilibrium. The Clapeyron equation exactly describes the thermodynamic system. However, it is difficult to work with, and the specific volume of hydrogen at the surface of the substrate is difficult to define. As with absorption materials and the van't Hoff equation, assumptions are made to simplify the use of the Clapeyron equation to a more usable form.

While the volume of the adsorbed hydrogen cannot be directly assessed, the specific volume of hydrogen on the solid surface is assumed similar to the specific volume of solid hydrogen, and small relative to the specific volume of the gas. This assumption holds for pressures of less than 100 bar and temperatures greater than 77K.<sup>21</sup> At higher pressures and lower temperatures, hydrogen gas density becomes an appreciable percentage of the solid-state density, and validation of this assumption is required. Applying this assumption to the Clapeyron equation gives

$$\left(\frac{\partial P}{\partial T}\right) = \frac{\Delta H_{des}}{TV_{gas}}$$

To eliminate the volume of the gas and reduce the equation to a function of temperature and pressure only, and equation of state for hydrogen can be used, and typically the ideal gas equation is used. Applying the ideal gas equation yields the Clausius-Clapeyron equation, which is typically used in the literature to evaluate the enthalpy of physisorption storage materials<sup>54, 55</sup>

$$\left(\frac{\partial ln(P)}{\partial T}\right) = \frac{\Delta H_{des}}{RT^2}$$

The Clausius-Clapeyron equation is subject to the same restrictions for the assumption of ideal gas behavior as the van't Hoff equation, discussed earlier. These assumptions hold well for the pressures specified in the USDOE targets. Higher pressure systems may need a fugacity correction to retain accuracy in non-ideal systems. Typically,  $\Delta H_{des}$  is a weak function of temperature but a relatively strong function of surface coverage. Because of this values of  $\Delta H_{des}$  are defined along isosteres, or constant values of surface coverage and are often defined in the literature as isosteric heat.<sup>56</sup>

An additional complication to the thermodynamic discussions arises in physisorption processes.<sup>57</sup> The Clausius-Clapeyron equation is based on a liquid-vapor equilibrium model, in which the liquid and vapor are separate phases occupying their own distinct volumes. In the gas-adsorbate system, however, these boundaries are less distinct, and molecules from the vapor phase may penetrate or otherwise be included within the volume associated with the adsorbate phase due to the normal compression of the gas, as illustrated in Figure 1.7. This behavior leads to two descriptions of the adsorption: absolute adsorption, which treats the volumes as separate layers and includes some gas phase molecules within the adsorbate layer, and excess adsorption, which excludes these gas molecules from the adsorbate layer. The molar balance between absolute and excess quantities can be defined as

 $n_{\rm excess} = n_{abs} - \rho V_{\rm Gibbs}$ 



Vapor-Liquid Equilibrium

Gas-Adsorbate Equilibrium

Figure 1.7: Schematic diagram of the relationship between liquid-vapor equilibrium and gasadsorbate equilibrium. The dashed line represents the Gibbs dividing line for both phases, with spheres below the line representing the absolute condensed phase. The liquid-vapor system has a well-defined interface, while the gas-adsorbate does not. Spheres in white below the line in the gas-adsorbate system are considered part of the gas phase density and are excluded from the excess adsorption calculation.

where  $\rho$  is the density of the gas phase, and  $V_{\text{Gibbs}}$  is the volume of the adsorbate on the material surface defined by a second boundary layer called the Gibbs dividing surface. The location of this surface is defined somewhat arbitrarily as the endpoint of the extension of the potential well from the material surface. Defining the location of the Gibbs defining surface from experiment is currently beyond available imaging technologies, and the surface location is also likely a function of system conditions.

The separation of measurements into absolute and excess isotherms also result from these definitions. Idealized versions of these isotherms, based on the Langmuir adsorption isotherm, appear in Figure 1.8. Experimental measurements can only yield excess isotherms, as these represent the pressure drop from expected values. Absolute isotherms are only available from theoretical calculations, requiring a definition of the Gibbs dividing surface. Excess isosteric heats are typically reported for physisorption materials for both theoretical and experimental studies, to maintain consistency


Figure 1.8: Idealized adsorption curves showing the relationship between absolute adsorption, excess adsorption, and gas density and typical lineshapes for the three types. Curves are based on the Langmuir isotherm and the ideal gas equation of state.

within measurements. These values typically lie between 5 kJ/mol and 20 kJ/mol, representing the relatively weak binding in these materials.<sup>58</sup>

#### 1.4.2 State-of-the-Art: Physisorption materials

As with absorption materials, no physisorption materials have yet been developed for commercial vehicular hydrogen storage materials. A number of state-of-the-art material types exist for physisorption materials which have the potential to fill this role. These types are reviewed briefly below.

Activated and Templated Carbons—The largest class of adsorbent materials for hydrogen storage, carbon-based materials are lightweight, easily produced and commercially available. Research into carbonaceous sorbent materials has focused on improving techniques for producing high specific surface areas, and thus improving overall adsorption capacity for hydrogen.<sup>58,55</sup>

Carbon sorbent materials are produced by pyrolysis of organic molecules at high temperatures in reduced oxygen environments. Organic molecules used for this process can be either natural or artificial in origin, and research into the benefits to the properties of the final carbon material resulting from various starting materials, particularly polymers such as polyether ether ketone<sup>59</sup> (PEEK), is ongoing.

For most carbon materials, the resulting pyrolyzed material is then "activated" to produce the high surface areas for adsorption. Several processes for performing this operation exist, and many proprietary methods are claimed by commercial producers. Two common laboratory techniques for activation are steam explosion, which rapidly flashes superheated water introduced into the carbon structure to physically rend the structure, and carbon dioxide maceration, which used high temperature reaction of the carbon surface with  $CO_2$  to produce gaseous CO, causing chemical erosion of the surface. These processes are capable of producing carbon materials with surface areas up to ~3000 m<sup>2</sup>/g, but consume a large portion of the starting material in the process. These high surface-area materials, such as AX-21 and MSC-30, are used commonly as a standard for comparison of new physisorption materials.<sup>60, 61</sup>

Recently, efforts to increase the surface area of carbon materials beyond activated carbons and to control the microporosity have led to templated carbon materials. These materials contain carbon in regular nanostructures that provide very high surface areas, and still largely obey Chahine's rule. Investigations into carbon nanotubes and nanotube bundles quickly led to more advanced regular nanostructured materials<sup>63, 65, 64</sup> and carbon aerogels.<sup>66</sup> The latest materials in this area are based on zeolite frameworks.<sup>60, 62</sup> Pyrolysis of polymers within the channels of a zeolite leads to a carbon negative of the zeolite, which can be extracted by dissolving the zeolite material in hydrofluoric acid. This produces a high surface area carbon with very uniform micropores and low skeletal density, and has shown similar uptake to the high surface area MSC-30 with an improved adsorption profile.

Unfortunately, the limits to surface area modification are being approached for pure carbon materials. Research efforts in this area relative to hydrogen storage have been increasingly directed toward engineering applications and reduction of the cost associated with producing these materials. Carbon nanoscaffolds and aerogels are also used frequently to improve the kinetics of absorption materials, and the interactions of the scaffold with these materials is under investigation.<sup>67–69</sup>

Metal Decorated, Doped, and Intercalated Carbons—The low adsorption enthalpies for hydrogen in carbon-based physisorption materials have led to continuous efforts to dope or otherwise transform these materials without adding significant weight to the structure. The most common technique used is to incorporate a small fraction of adatoms or inclusions, which have strong bulk interactions with hydrogen gas, into the material's structure. These dopants may then increase the overall interaction potential of the material.

Initial doping strategies used platinum or palladium nanocluster incorporations onto carbon structures, to produce a material similar to supported-metal catalysts.<sup>51,71</sup> Because of the high cost of precious metals, later studies have focused on the use of 3d transition metals, particularly titanium,<sup>70</sup> to produce similar effects. The metallic clusters in these materials were unfortunately found to act independently in most cases, with the metal clusters maintaining their own surface coverage of chemisorbed hydrogen atoms at stronger energies than the remaining carbon structure. Several groups have reported *spillover* in such materials, where dissociated hydrogen is transferred from metallic clusters to the carbon structure, providing higher surface area coverage of strongly bound hydrogen atoms.<sup>72,73</sup> These interactions have been observed at high temperature for platinum and ruthenium catalysts on carbon supporting structures through neutron spectroscopy,<sup>74</sup> and enhanced adsorption capacity at room temperature has been demonstrated. However, these enhancements appear to be a minor effect, and spillover processes are unlikely to produce increases in adsorption enthalpies that are significant for storage applications.

Additional techniques involve direct doping of the carbon structure, rather than surface modifications. Boron and nitrogen doping of the graphitic layers have both shown to increase the adsorption capacity of activated carbons.<sup>75–77</sup> However, whether the increase is due primarily to modification of the surface potential or to increased accessible surface area due to bowing of graphitic layers remains unknown. Intercalation of alkali metals into graphitic structures have also been used, and shown to both modify the interlayer spacing in the structure and increase the hydrogen adsorption enthalpy.<sup>52</sup> These materials, however, are difficult to produce and tend to produce alkali metal hydrides after a few cycles. Relatively few studies continue to examine these methods for improving the binding enthalpy in hydrogen storage applications, but continue to be relevant for chemical catalysis applications. Studies are largely directed at examination of metal substitutions and reduction of metal size and increase in metal dispersion to reduce the amount of metal loading necessary to produce desired effects.

Metal-Organic Frameworks—Metal organic frameworks (MOFs) have received a large amount of attention in physisorptive hydrogen storage research, both for the very high specific surface areas available in these materials and for large variability in the possible frameworks.<sup>78</sup> These crystalline materials, composed of metal atoms linked together with organic molecules in a three-dimensional framework, contain uniform pores and structures which allow for greater control over the adsorption capacity and enthalpy. MOFs have consistently shown meaningful adsorption capacities at room temperature, making them promising candidates for further development.<sup>79–81</sup>

Hundreds of MOFs have been identified, with varying pore sizes and chemical makeup, and these variations are being actively pursued for gas adsorption applications. Overall, trends in hydrogen binding show an increase in hydrogen adsorption capacity with smaller pore sizes or higher specific surface area, in accordance with Chahine's rule. Incorporation of mid-row transition metals and alkaline earth metals at the metal site have also been shown to increase binding enthalpies, increasing hydrogen storage at high temperatures.<sup>82</sup> Examination of various chemistries and binding motifs are still ongoing. Besides basic materials work, research into improvements in the scale-up and large scale production of MOFs are needed before these materials can be implemented outside of the laboratory.

#### 1.5 Kubas Binding<sup>1</sup>

Recently, a new classification of hydrogen storage materials has been identified: Kubas materials. These materials, first identified in early work on MOFs<sup>80,83,84</sup> (MOFs) and Prussian blue analog<sup>13</sup> physisorption materials, showed enhancement in the isosteric heat of adsorption by approximately 2 kJ/mol H<sub>2</sub> over similar materials if unsaturated metal centers were included in the framework. These materials appeared to take advantage of a new binding motif, designated as Kubas binding, which binds a dihydrogen molecule chemically to the unsaturated metal center, in an arrested dissociative absorption. This interaction provides thermodynamics in the ideal range between the ranges of standard physisorption and absorption materials, with the rapid kinetics associated with physisorption materials and long cycle lives. Materials designed to exploit this effect in hydrogen storage materials have shown enhancement of physisorption enthalpies by up to 35 kJ/mol.<sup>85</sup> Moreover, these properties are believed to be tunable, with alterations to the ligands attached to the metal center providing modifications to the binding enthalpy.

Because Kubas binding was initially discovered in MOFs and other physisorption materials during hydrogen storage measurements, the interaction has consistently been treated as physisorption. However, recent experimental studies<sup>86</sup> have also demonstrated unusual behaviors arising in Kubas materials, such as activation energies and plateaus within the isotherms and widely varying adsorption enthalpies, that are not typically associated with physisorption processes. Because an understanding of the mechanism is important to proper calculation of the thermodynamic properties, these oddities have prompted further investigation into the mechanism of these interactions to enable a detailed understanding of these materials and enable rational design.

Mechanistic studies of Kubas binding require materials that exhibit this binding type almost exclusively. The original complexes exhibiting this behavior, the dihydrogen complexes first reported<sup>87</sup> by Gregory Kubas, provide such a model. In the next chapter, the properties of dihydrogen complexes and the complexes useful for thermodynamic studies in the solid state are reviewed.

<sup>&</sup>lt;sup>1</sup>Parts of this section have been reproduced with permission from Evaluation of the Thermodynamic Properties of H<sub>2</sub> Binding in Solid State Dihydrogen Complexes  $[M(\eta^2-H_2)(CO)dppe_2][BArF^{24}]$  (M = Mn, Tc, Re): An Experimental and First Principles Study by David G. Abrecht and Brent Fultz. Copyright 2012 American Chemical Society.

# Chapter 2 Review of Dihydrogen Complexes

The unexpected discovery of the first dihydrogen complex,  $W(\eta^2-H_2)(CO)_3(PCy_3)_2$ , has led to the subsequent reports of hundreds of complexes containing the dihydrogen ligand, including the reclassification of many previously reported as polyhydrides.<sup>88</sup> These complexes demonstrate the full range of molecular bond breaking in the hydrogen molecule to form a metal dihydride. Furthermore, the degree of bond breaking in the ligated dihydrogen molecule has been shown to be a function of the degree and type of electron donation from other ligands attached to the metal center.<sup>89,93</sup> This property has led to considerable interest in dihydrogen complex research, both for furthering the basic understanding of metal-hydrogen interactions and H-H bond breaking, and for applications in catalysis, biochemistry, and hydrogen storage and production.

In this chapter, the properties and synthesis of dihydrogen complexes are reviewed, and their applications as model compounds for Kubas binding to further basic understanding and rational design of Kubas materials for hydrogen storage applications are discussed. Complexes and properties relevant to hydrogen storage are the primary focus of this review; no attempt is made to cover the extremely rich expanse of dihydrogen complex chemistry, which includes such diverse topics as protein activity and extreme low temperature chemistry.

#### 2.1 General Properties, Structure, and Bonding

Dihydrogen complexes cover an extremely diverse range of properties and chemistries, and while the majority share common properties, there are few rules of thumb associated with these complexes that



Figure 2.1: Ball-and-stick model<sup>91</sup> of the organometallic dihydrogen fragment  $[FeH(\eta^2-H_2)dppe_2]^+$ , showing octahedral configuration of the central iron atom and side-on coordination of the hydrogen molecule to the metal center. Alkyl and aryl hydrogens have been removed for clarity.

do not have demonstrated exceptions.<sup>88</sup> Organometallic dihydrogen complexes normally contain one dihydrogen molecule bound chemically to a transition metal center, although occasionally two dihydrogen ligands may be present. Most known dihydrogen complexes are octahedrally coordinated,<sup>90</sup> and all known dihydrogen complexes contain hydrogen bound in a side-on ( $\eta^2$ ) configuration, as shown in Figure 2.1. These configurations preferentially position the molecular orbitals of both the hydrogen molecule and the organometallic molecule for bonding interactions. Dihydrogen binds as a neutral, or L-type, ligand and does not change the formal oxidation state of the metal center or charge on the molecule.<sup>87</sup> However, dissociation of the dihydrogen molecule at the complex forms a dihydride, which increases the oxidation state of the metal by two. Instability of the corresponding dihydride oxidation state may be a factor in stability of the dihydrogen complex, as in iron (II) dihydrogen complexes like [FeH(H<sub>2</sub>)dppe<sub>2</sub>][BF<sub>4</sub>].<sup>96</sup>

Bonding theory for dihydrogen complexes was developed relatively quickly after their discovery.<sup>92</sup> The binding mechanism for dihydrogen complex formation has been described primarily through analogy to the Dewar-Chatt-Duncanson model for  $\pi$  complexes, which introduces the concept of backdonation of electron density. In this model, the  $\pi$  electrons associated with an olefinic double bond are partially donated to an empty d-shell of a metal center, forming a coordination complex in a similar fashion to the lone pair donation in classic Werner complexes. Backdonation from the metal center to the  $\pi^*$  orbital of the olefin, which becomes a bonding orbital with respect to the complex, prevents complete rupture of the  $\pi$  bond and stabilizes the complex.

Dihydrogen complexes follow a similar model, donating electron density from bonding electrons in the  $\sigma$  orbital of the hydrogen molecule to an empty d-shell of the metal, while electron density from filled d-shells of the metal center is backdonated to the  $\sigma^*$  orbital of the hydrogen molecule to stabilize the three-center bond and redistribute charge. Molecules exhibiting this binding have been designated  $\sigma$ -complexes.<sup>92</sup> Unlike  $\pi$  complexes, donation from the absorbed molecule in  $\sigma$  complexes occurs through donation of the main bonding electrons, and can lead to complete bond rupture and dissociation if donation from the absorbed molecule or backdonation from the metal center becomes too strong. This balance allows direct control over bond breaking and formation of dihydrides in dihydrogen complexes, such that the entire reaction coordinate can be represented.<sup>88</sup>

Lability of the dihydrogen molecule has been directly related to the degree of bond dissociation in H<sub>2</sub> complexes.<sup>113,114</sup> Additional backdonation to the  $\sigma^*$  orbital causes elongation of the H-H bond towards bond breaking, which has been linked to reduced lability of the dihydrogen complex and more hydridic character. Increased



Figure 2.2: Bonding model for orbitals during transition metal-dihydrogen binding.

backdonation results from higher electron density at the metal center, and is a factor of both the identity and valency of the central metal atom and donation from the accompanying ligands, particularly the ligand *trans* to the H<sub>2</sub> binding site. This effect has been cataloged computationally by Li and Ziegler<sup>89</sup> in the Mo(CO)<sub>x</sub>(PR<sub>3</sub>)<sub>5-x</sub>(H<sub>2</sub>) system, where substitution of strongly  $\pi$  accepting carbonyl groups by weakly  $\pi$  accepting phosphines leads to increasing dissociation of the H-H bond. A strong  $\sigma$  donor trans to the dihydrogen complex, such as hydride, also reduces H-H bond elongation and increases dihydrogen character by forcing electron density to remain within the  $\sigma$  orbital of the hydrogen molecule, despite increased electron density at the metal center. This effect allows dihydrogen ligands to be present despite highly electrophilic metal centers.

Reduced electron density at the metal center leads to decreased backdonation to the hydrogen molecule. This effect leads to a reduction in bond stability in  $\pi$  accepting ligands like N<sub>2</sub> and instability of the complex. However, the reduced backdonation in electron-poor dihydrogen complexes is often compensated by increased  $\sigma$  donation from the H<sub>2</sub> molecule. Li and Ziegler found a decrease in the energetic contribution to the molecule from backdonation from -21.2 kcal/mol in Mo(CO)(PH<sub>3</sub>)(H<sub>2</sub>) to -11.9 kcal/mol in Mo(CO)<sub>5</sub>(H<sub>2</sub>), representing an increased instability with less backdonation, but a corresponding and near equivalent energetic stabilization from  $\sigma$  donation from the H<sub>2</sub> molecule, leading to nearly equivalent decomposition energies for the entire series.<sup>89</sup> This allows the hydrogen ligand to bind effectively to relatively electron-poor metals, and allows stabilization of dihydrogen complexes by reducing the total electron density available in the molecule. For this reason, cationic dihydrogen complexes are more stable, and make up the majority of reported dihydrogen complexes. The same effect may be produced by using electron-poor metals (i.e., early transition metals) and by using metals with poor backdonation ability.

#### 2.2 Synthesis and Characterization

Synthesis and characterization of dihydrogen complexes described in the chemical literature are primarily concerned with the solution state. Since the metal centers are isolated within a large coordination sphere created by the ligands in both solution and solid states, crystal fields and intermolecular interactions are not expected to cause major deviations in the binding behavior for small molecules at the central site, and in many cases the properties can be translated directly. However, ligand rearrangements and other kinetic processes are often considerably slower in the solid state, and reaction mechanisms that take advantage of a dynamic equilibrium between geometric arrangements may be inaccessible when considering these complexes for hydrogen storage materials. Most dihydrogen complexes are air-sensitive, and must be prepared under an inert atmosphere. Inert gases such as argon or helium must be used when formation of the  $N_2$  adduct is undesirable. Excess moisture should also be avoided, although certain dihydrogen complexes are entropically stabilized against substitution of the  $H_2$  ligand by  $H_2O$  at room temperature.<sup>88</sup>

#### 2.2.1 Synthetic Routes

Stable dihydrogen complexes are primarily formed through one of four synthetic routes:<sup>87,94–97</sup> direct hydrogenation of an unsaturated precursor, hydrogen displacement of a neutral or charged ligand, and protonation of a hydride complex:

$$L_{n}M + H_{2} \Longrightarrow L_{n}M(H_{2})$$

$$L_{n}ML' + H_{2} \Longrightarrow L_{n}M(H_{2}) + L'$$

$$L_{n}MX + H_{2} \Longrightarrow [L_{n}M(H_{2})]^{+} + X^{-}$$

$$L_{n}MH + HX \Longrightarrow [L_{n}M(H_{2})]^{+} + X^{-}$$

The first equation, representing direct hydrogenation of an unsaturated precursor, is the most relevant for hydrogen storage and is the expected reaction during cycling in solid-state fuel storage applications. However, this reaction is one of the most difficult for initial production of the complexes, due to solution formation of complexes and the general instability of unsaturated organometallic clusters towards addition of solvent or other small molecules. As a result, this synthetic mechanism is reported for only a small portion of dihydrogen complexes.

Unsaturated precursors capable of absorbing hydrogen directly in solution are stabilized by the presence of agostic interactions, where C-H bonds from pendant groups on attached ligands weakly occupy the open coordination site, entropically stabilizing the complex towards addition of a molecule.<sup>98,99,88</sup> For example, in W(CO)<sub>3</sub>(PCy<sub>3</sub>)<sub>2</sub> or [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] cyclohexyl and phenyl groups located on the phosphines, respectively, have been shown through crystallographic methods to preferentially arrange to coordinate the *ortho* proton to the open site.<sup>99,100</sup> While H<sub>2</sub> has been shown to actively displace the agostic interaction in W(CO)<sub>3</sub>(PCy<sub>3</sub>)<sub>2</sub>, [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] in the agostic configuration has been shown to be inactive to hydrogen absorption in the solid state despite ready absorption of hydrogen in solution, and similar steric effects have been shown to inhibit dihydrogen binding in the solid state.<sup>86</sup> Separate evaluations of direct hydrogenation of complexes containing agostic interactions are therefore needed for solid-state applications.

Displacement of small molecule ligands, such as  $H_2O$  or  $N_2$ , also produces  $H_2$  complexes amenable to hydrogen storage applications.<sup>94,95</sup> Dihydrogen complexes commonly have corresponding dinitrogen complexes<sup>101,100,107,97,102</sup> due to the similarity in  $\pi$  acceptor ability of the  $H_2$  and  $N_2$  ligands, and the weak  $\sigma$  donation ability of  $N_2$  relative to  $H_2$  favors dihydrogen coordination in many complexes, allowing easy displacement. This method provides effective protection of the binding site during synthesis, and the small, gaseous  $N_2$  molecule can be removed from the coordination site in the solid state more easily than agostic interactions arising from bulky ligands.  $N_2$  release from initial charging would be benign to fuel cell catalysts and other catalytic converters in the system, and removal of the protective  $N_2$  adduct can correspondingly be performed after the fuel tank is charged.

Displacement of a charged ligand is typically not desirable for hydrogen storage applications because of the weight of the persistant anion; however, methods exist to reduce the complex *in situ* to separate the ionic pair. Direct displacement of a charged ligand is difficult without a precipitating cation such as sodium to drive the reaction,<sup>103, 104</sup> although examples exist: chlorine is displaced by H<sub>2</sub> without an intermediate in Re(CN-t-Bu)<sub>3</sub>(PCy<sub>3</sub>)Cl in dichloromethane.<sup>105</sup> Formation of a neutral complex is possible if a hydride is used as a reducing agent:

$$L_nMX_m + mH^- + H_2 \longrightarrow L_nM(H_2)H_m + mX^-$$

This method has successfully been used to produce several dihydrogen complexes,<sup>117,106</sup> and NaH, NaBH<sub>4</sub>, or LiAlH<sub>4</sub> may be used as hydride sources. NaH is ideal chemically for these reactions, although poor solubility in organic solvents often prevents its direct use. Use of borohydrides and alanates may produce side products because of the strong reductive power of LiAlH<sub>4</sub> and the coordination potential of borohydride.

Protonation of a hydride complex has received a great deal of attention as a synthetic route because of both the higher stability of cationic dihydrogen complexes and the relevance of these processes to modeling the behavior of hydrogenase enzymes.<sup>88</sup> However, because a large, poorly coordinating anion, such as  $BPh_4$ ,<sup>96</sup> remains, these methods are less favorable for hydrogen storage applications except to produce model complexes to test properties of Kubas binding. Direct acidification of a hydride by the strong acid of a poorly coordinating anion in an organic solvent, producing a dihydrogen salt that is poorly soluble in the organic solvent, allows clean production and purification of the complexes and is used for nearly all dihydrogen complex production because of the simplicity of the synthetic method.

#### 2.2.2 Dihydride/Dihydrogen Identification

Characterization of dihydrogen complexes is often difficult, as belied by their late discovery. Spectroscopy remains the most definitive method for determining the existence of a dihydrogen ligand.<sup>88</sup> The existence of the H-H bond produces isotope-dependent vibrational and rotational modes in the organometallic complex that are not present in hydrides, and uniquely identify the dihydrogen molecule.<sup>108</sup> These modes produce significant, observable shifts in IR and Raman spectroscopy band energies between  $H_2$  and  $D_2$  isotopes, on the order of hundreds of wavenumbers. Additionally, substitution of  $H_2$  by HD leads to unique mode energies lying between  $H_2$  and  $D_2$  rather than a splitting of the two homoisotope bands.

Of the six normal modes for the dihydrogen molecule, only the symmetric stretching mode and H-H stretch produce strong bands in IR and Raman spectroscopy, and H-H stretching modes in the 2080–3200 cm<sup>-1</sup> range often overlap with the C-H stretching modes of other attached ligands.<sup>88,109</sup> Observation of the weak mode bands can be facilitated by the use of inelastic neutron scattering (INS) techniques with perdeuteration of the remaining ligands to isolate the H-H modes.<sup>110,111</sup> This technique, although extremely powerful for determination of the vibrational modes in dihydrogen ligands, is used less frequently because of the expense associated with the measurement.

The existence of a dihydrogen ligand may also be inferred from crystallographic characterization



Figure 2.3: Idealized NMR spectra peaks for a)  $H_2$  complex, showing broadened signal and b) HD substitution, showing 1:1:1 triplet splitting of the peaks and sharpened signal.  $J_{\rm HD}$ , the coupling constant for the spin-spin coupling of the hydrogen and deuterium atoms is measured as shown and is proportional to the H-D bond distance, and is absent in the absence of a bond. HD and  $H_2$  signals may share the same chemical shift, or the HD signal may be shifted by a small amount due to isotope effects on the bonding properties.

techniques. X-ray and neutron diffraction studies used to characterize the bond lengths and angles of the molecule to provide insight into the bonding orbitals and electronic character can simultaneously provide separation distances between hydrogen atoms. Distances of less than 1.3 Å between hydrogen atom positions indicate potential dihydrogen complexes, and distances greater than 1.3 Å are unlikely to retain any dihydrogen character.<sup>88, 112, 113</sup> However, these techniques cannot be used to confirm the existence of a bond between the hydrogen atoms, and spectroscopic methods are needed for confirmation. Interatomic distances determined from crystallographic techniques are also used to determine the existence of agostic complexes in unsaturated complexes used for direct hydrogenation formation of dihydrogen complexes.<sup>99</sup>

NMR spectroscopy is used extensively for identification of dihydrogen complexes.<sup>96, 115, 97, 88, 90, 113</sup> Substitution of the  $H_2$  molecule with HD produces a 1:1:1 triplet splitting pattern at the chemical

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shift corresponding to the dihydrogen complex in <sup>1</sup>H NMR spectra, as shown in Figure 2.3. This splitting occurs due to spin-spin coupling between the atoms along the residual H-D bond, and is absent if insufficient electron density remains between the two molecules. This technique provides rapid, inexpensive evidence for the presence of an  $H_2$  ligand in both solid and solution states, and is included in the majority of literature reports for dihydrogen complexes. Correlations have also been established between the H-H distance and the dipole-dipole relaxation time ( $T_1$ ) in solution NMR,<sup>116,117</sup> but temperature dependence and interference from adjacent ligands makes this technique less favorable.<sup>118</sup>

Some compound-specific techniques for identification of dihydrogen complexes have also been reported. Morris and Schlaf<sup>119</sup> reported unique Mössbauer isomer shifts and quadrupole splittings for dihydrogen complexes with iron centers, and unique partial quadrupole splittings for the dihydrogen ligand relative to hydride ligands. King et al. report<sup>100</sup> carbonyl shifts for complexes containing carbonyl *trans* to the dihydrogen complex similar to those for dinitrogen or dicarbonyl complexes, due to  $\pi$  acceptor ability of the dihydrogen ligand that is absent in hydride ligands. These techniques can provide strong support for the presence of a dihydrogen complex, but are limited to specific chemistries.

#### 2.3 Binding Strength and Hydrogenation Thermodynamics<sup>1</sup>

The binding strength of the H<sub>2</sub> molecule in dihydrogen complexes is typically evaluated from the stretching of the H-H bond when bound to the metal center. This property is a measure of the electron density removed from the bonding  $\sigma$  orbital of the hydrogen molecule and donated to the anti-bonding  $\sigma^*$  orbital by the metal center, representing the degree of bond breaking within the H<sub>2</sub> molecule and the strength of the electronic interaction. H-H distances less than ~1.0 Å, representing the "true" dihydrogen complexes, are considered weakly bound and easily removed from the dihydrogen complex. Bond distances of greater than ~1.0 Å, representing complexes with

<sup>&</sup>lt;sup>1</sup>Parts of this section have been reproduced with permission from Evaluation of the Thermodynamic Properties of H<sub>2</sub> Binding in Solid State Dihydrogen Complexes  $[M(\eta^2-H_2)(CO)dppe_2][BArF^{24}]$  (M = Mn, Tc, Re): An Experimental and First Principles Study by David G. Abrecht and Brent Fultz. Copyright 2012 American Chemical Society.

more dihydride character, are considered more tightly bound and less likely to reversibly absorb the dihydrogen complex.<sup>112,113</sup>

X-ray diffraction and neutron diffraction provide the only truly definitive method for determining H-H bond distance. However, x-ray scattering from the low mass hydrogen atom relative to large metal atoms is often weak and indeterminate, and neutron diffraction experiments are expensive and difficult to perform. Estimates of the H-H bond separation are available from NMR spectroscopy measurements, and are the most commonly reported method of approximating the binding strength. Maltby<sup>121</sup> and Luther<sup>122</sup> independently developed relationships correlating the H-H bonding distance from x-ray and neutron diffraction measurements to the solid-state NMR coupling constant,  $J_{\rm HD}$ , of isotopomeric HD complexes. The relationship given by Maltby is:

$$d_{\rm HH} = 1.42 - 0.0167 J_{\rm HD}$$

where  $d_{\rm HH}$  is the H-H bond distance, in angstroms, and  $J_{\rm HD}$  is given in hertz. The equation given by Luther is the same as the equation above, with values within experimental error of those found by Maltby. These relationships have also been shown to hold approximately in solution NMR studies, despite the potential for interference from solvent molecules and increased molecular motion. Because of the ease and clarity of solution state NMR spectra relative to solid-state NMR spectra or organometallic crystallographic methods, this method has come to dominate the literature for estimates of binding strength.

While the interatomic distances can provide relative measures of the dihydrogen binding strength in similar families of complexes, these values only provide approximations to the binding strength and are not universally applicable as measures of thermodynamic properties. The dihydrogen molecule often exists in equilibrium with its corresponding dihydride, with a low energy barrier separating the two states. Removal of dihydrogen may be possible through the transient formation of a weak dihydrogen complex, as seen in  $W(CO)_3(P(^i-Pr)_3)_2(H_2)$ , despite the complete dissociation of the  $H_2$  molecule to form the dihydride.<sup>120</sup> The existence of additional hydrides ligated to the metal surface can also add entropic stabilization to the complex through intramolecular site exchange, stabilizing weaker complexes.<sup>96</sup> Varying ratios of  $\sigma$  and  $\pi$  donation ability created by the metalligand pairs lead to poor correlation between the observed lability of the dihydrogen ligand and the H-H distance for molecules with widely disparate chemistries. Additionally, the bond distance provides no measure of the stability of the dehydrogenated complex; rearrangement of the molecular geometry after dehydrogenation can strongly affect the thermodynamic properties of hydrogen absorption and release, and consequently the equilibrium pressure and temperature.<sup>88</sup>

Thermodynamic information on dihydrogen complexes has largely been focused on the heterolytic cleavage of the dihydrogen molecule and the protonation ability of the complexes,<sup>96</sup> in order to obtain greater information on the activity of hydrogenases.<sup>88</sup> These interactions are unfortunately detrimental to direct hydrogenation hydrogen storage applications, as they represent a potentially irreversible side reaction to form stable hydride complexes, and thermodynamic values presented in these studies do not typically include the unsaturated complex for assistance in evaluating direct hydrogenations. The available thermodynamic data on direct hydrogenations to form dihydrogen complexes remains scarce. Calorimetric studies have reported solution-state binding enthalpies of  $\Delta H^{\circ} = -41.8$  and -46.9 kJ/mol for the complexes W(CO)<sub>3</sub>(PCy<sub>3</sub>)<sub>2</sub> and W(CO)<sub>3</sub>(P-i-Pr<sub>3</sub>)<sub>2</sub> in toluene, respectively.<sup>123,124</sup> A similar study in THF, attempting to develop relationships between the group 6 metals in the complexes  $M(CO)(PCy_3)_2$  (M=Cr, Mo, W), found  $\Delta H^{\circ}$  between -27.2 kJ/mol and -41.8 kJ/mol, and entropy values  $\Delta S^{\circ}$  between -100 and -110 J/mol-K.<sup>95</sup> Solid-state measurements are limited and show deviations from solution behavior, with one study giving  $\Delta H^{\circ} = -13.22$  kJ/mol and  $\Delta S^{\circ} = -9.62 \text{ J/mol-K}$  for hydrogen absorption over the complex<sup>125</sup> [Ir(cod)(PPh\_3)\_2]SbF\_6, representing a significant reduction in binding strength and an unexpected increase in the entropy of the bound state. While most of the data are consistent with values obtained from hydrogen storage materials, the current measurements represent a wide range of metals and ligands providing different chemical properties that, apart from the systematic study performed by Gonzalez, et al. for the group 6 complexes, do not provide chemical trends to aid in the design of new materials.

Difficulties in obtaining systematic thermodynamic data for dihydrogen complexes arise from their inherent stability, with the equilibrium pressures of most known materials falling outside of acceptable ranges for traditional chemistry techniques. New techniques for measuring the thermodynamic properties of dihydrogen complexes, especially at elevated pressures, are necessary to evaluate their potential as hydrogen storage materials. In the next chapters, the mechanism of pressurized hydrogenation to form dihydrogen complexes in the solid state is examined, and techniques for establishing the thermodynamic properties of these materials are established.

## Chapter 3

## Studies of $[M(\eta^2 - H_2)(CO)dppe_2]^+$ (M = Mn, Tc, Re) Complexes<sup>1</sup>

As discussed in the previous chapter, technical difficulties have largely prevented direct quantification of thermodynamic properties for dihydrogen binding in organometallic complexes. To help resolve these difficulties and enable the generation of systematic thermodynamic data relevant to the hydrogen storage community, we have used the Sieverts method to obtain experimental isotherm measurements on the interaction of hydrogen gas with the dihydrogen complex [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] (dppe = 1,2-bis(diphenylphosphino)ethane, BArF<sup>24</sup> = tetrakis-(3,5-trifluoromethyl)phenylborate). In addition, we report electronic structure calculations on the hydrogenation of the fragments  $[M(CO)dppe_2]^+$  (M = Mn, Tc, Re) to examine trends in the binding energy within the group 7 metals. We interpret our results in terms of the solid-state binding mechanism for hydrogen in these materials, and show a relatively rapid and facile means to quantitatively evaluate thermodynamic properties and establish chemical trends that are useful for materials design.

#### 3.1 Experimental

Unless otherwise stated, all reactions were performed under a dinitrogen atmosphere using either a controlled atmosphere glovebox or Schlenk line techniques. Manganese pentacarbonyl bromide, magnesium metal and sodium tetrafluoroborate were purchased from Alfa-Aesar and used without

<sup>&</sup>lt;sup>1</sup>This chapter has been reproduced with permission from Evaluation of the Thermodynamic Properties of  $H_2$ Binding in Solid State Dihydrogen Complexes  $[M(\eta^2-H_2)(CO)dppe_2][BArF^{24}]$  (M = Mn, Tc, Re): An Experimental and First Principles Study by David G. Abrecht and Brent Fultz. Copyright 2012 American Chemical Society.

further purification. 1,2-bis(diphenylphosphino)ethane was purchased from Strem Chemicals and used without further purification. 3,5-bis(trifluoromethyl)bromobenzene was purchased from Sigma-Aldrich and was degassed using three freeze-pump-thaw cycles and dried over activated alumina before use. Research-grade gases were purchased from Matheson and used directly. All solvents were dried and deoxygenated by purging with dry dinitrogen gas for 15 minutes before passing through packed columns of activated alumina and activated copper. Mass spectroscopy of off-gassing products was performed using a Stanford Research Systems RGA-200 residual gas analyzer with observable pressure ranges from  $10^{-8}$  to  $10^{-5}$  torr. After synthesis, materials were stored under dry argon in an atmosphere-controlled glovebox until used for sorption measurements.

Solution-state NMR spectra were recorded on a Varian 300 MHz instrument with <sup>1</sup>H shifts reported relative to the residual solvent peak, and <sup>31</sup>P peaks reported relative to 85% H<sub>3</sub>PO<sub>4</sub>. Deuterated NMR solvents were purchased from Cambridge Isotopes Laboratories. Deuterated dichloromethane was vacuum distilled from a sodium/benzophenone solution before use. Deuterated acetone was vacuum distilled twice from CaSO<sub>4</sub> and stored over 4A molecular sieves before use.

Preparation of Na[BArF<sup>24</sup>]—This procedure was a modification of one in the literature.<sup>131</sup> 3,5bis(trifluoromethyl)bromobenzene (5 mL, 8.55 grams, 2.918 mmol) was added to a suspension of anhydrous NaBF<sub>4</sub> (0.8173 grams, 7.446 mmol) and magnesium metal (5.028 grams, 206.87 mmol, excess) in 150 mL of diethyl ether under a nitrogen atmosphere, causing the solution to turn olive green. The suspension was stirred for 24 hours, producing an orange solution with white precipitate. The solution was exposed to air and quenched with 20 mL of saturated aqueous sodium carbonate solution, followed by filtration over a coarse frit to remove precipitated salt. The ether layer was removed, and the aqueous layer was washed with 50 mL of fresh ether. The ether layers were combined, dried over sodium sulfate and filtered. The solvent was removed *in vacuo* and the resulting residue was dissolved in benzene and distilled in a Dean-Stark apparatus under nitrogen for three hours to remove residual water. The benzene was then removed *in vacuo*, and the residue was washed with dichloromethane and filtered to produce the product as an off-white powder. <sup>1</sup>H NMR (300 MHz) in d<sub>6</sub>-acetone:  $\delta$  7.67 (s, 4H); 7.79 (t, 8H). **Preparation of fac-MnBr(CO)**<sub>3</sub>**dppe**—This procedure was a modification of one in the literature.<sup>132</sup> A solution of the phosphine ligand 1,2-bis(diphenylphosphino)ethane (dppe, 3.5326 grams, 8.866 mmol) in 100 mL benzene was added to crystalline Mn(CO)<sub>5</sub>Br (1.2180 grams, 4.430 mmol) in a 250 mL quartz round-bottom flask and allowed to stir for one hour, causing Mn(CO)<sub>5</sub>Br to slowly dissolve and release CO gas to produce a yellow solution containing both the remaining free ligand and the product. The compound was observed by NMR in CD<sub>2</sub>Cl<sub>2</sub> but not isolated. <sup>1</sup>H NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$  2.80 (m, 4H); 3.16 (m, 4H); 7.41(m, 20H). <sup>31</sup>P NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$ 69.6 (s).

**Preparation of trans-MnBr(CO)dppe**<sub>2</sub>—This procedure was a modification of one in the literature.<sup>132</sup> The solution of *fac*-MnBr(CO)<sub>3</sub>dppe and free ligand in benzene from the previous synthesis was irradiated with UV light for 2 hours under an evacuated headspace and strong stirring in a 250 mL fused quartz round-bottom flask, causing the precipitation of 0.8 grams of red-orange solid that adhered to the flask walls and blocked further irradiation. The solvent was removed *in vacuo*, and the resulting residue was washed with fresh benzene and filtered to obtain the product as an orange powder. The filtrate was collected and the procedure was repeated until no additional precipitate was formed. Total yield 3.365 grams (3.506 mmol, 79.1%) on manganese. <sup>1</sup>H NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$  2.52 (m, 4H); 2.80 (m, 4H); 7.04 (t, 16H); 7.10 (m, 8H); 7.22 (quart., 8H), 7.32 (m, 8H). <sup>31</sup>P NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$  72.1.

**Preparation of**  $[Mn(CO)dppe_2][BArF^{24}]$ —This procedure was a modification of one in the literature.<sup>115</sup> Solid Na[BArF<sup>24</sup>] (3.059 grams, 3.452 mmol) was mixed with solid *trans*-MnBr(CO)dppe<sub>2</sub> (3.266 grams, 3.403 mmol) in a 100 mL round-bottom flask and 50 mL of dichloromethane was added to the solid mixture, immediately forming an emerald green solution that was stirred for one hour. After one hour the solution was filtered over celite to remove excess insoluble Na[BArF<sup>24</sup>] and bromide salts. The filtrate was collected and the solvent removed *in vacuo*, causing a color change from emerald green to aquamarine blue. The residue was redissolved in toluene, producing a green solution, and filtered to remove any remaining bromide salts. The filtrate was collected and the solvent removed *in vacuo*, producing a blue-green powder containing mixed adducts and toluene within the structure, as described by King et. al.<sup>115</sup> Removal of adducts and toluene was performed under high vacuum conditions at 373K until no additional toluene was observable by residual gas mass spectroscopy, producing the five-coordinate product in 90% purity. <sup>1</sup>H NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub> under argon atmosphere:  $\delta$  2.78 (s, 8H); 6.20 (s, broad, 8H); 6.95–7.40 (m, 32H); 7.56 (s, 4H); 7.73 (s, 8H). <sup>31</sup>P NMR (300 MHz) in CD<sub>2</sub>Cl<sub>2</sub> under argon atmosphere:  $\delta$  82.47 (s, broad). **Reaction of [Mn(CO)dppe\_2][BArF**<sup>24</sup>] with H<sub>2</sub> for NMR spectroscopy - Confirmation of the dihydrogen absorption ability of [Mn(CO)dppe\_2][BArF<sup>24</sup>] was performed by solution NMR. CD<sub>2</sub>Cl<sub>2</sub> was vacuum transferred into a J. Young tube containing solid [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>], forming a deep blue solution. One atmosphere of hydrogen gas was introduced into the tube, and upon shaking the solution rapidly turned yellow. <sup>1</sup>H NMR (300MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$  -7.22 (s, broad, 1.5H); 2.24 (s, 4H); 2.52 (s, 4H); 6.88-7.44 (m, broad, 40H); 7.56 (s, 4H); 7.73 (s, 8H). <sup>31</sup>P NMR (300MHz) in CD<sub>2</sub>Cl<sub>2</sub>:  $\delta$  85.2 (s, broad).

#### 3.1.1 H<sub>2</sub> Absorption Measurements

Iterative isotherm and kinetic rate measurements were performed on 2.9 grams of degassed  $[Mn(CO)dppe_2]$  [BArF<sup>24</sup>] using a custom-built Sieverts apparatus, described in detail previously.<sup>133</sup> Samples were loaded into a 14 mL stainless steel reactor under an argon atmosphere and sealed before transfer to the instrument. Swagelok VCR copper filter gaskets with 2  $\mu$ m stainless steel filters were used to confine the powdered sample to the reactor during measurements. After transferring the reactor to the assembly, samples were allowed to pump down overnight to  $3.1 \times 10^{-7}$  torr at the pump inlet before measurements were performed. Pressure on the reactor and manifold during isotherm measurements was measured using an MKS model 120AA-25000RBJ capacitance transducer with a resolution of 0.25 torr in the range of interest. Higher pressures during kinetic measurements were recorded using an MKS model 870B33PCD2GC1 capacitance transducer with a 0-3000 psi range. The time resolution of the instrument was 1500 ms during kinetic measurements.

Manifold and reactor volumes were calibrated prior to absorption measurements through iterative expansion from a calibrated volume with argon gas. Isotherms and kinetic rate measurements were recorded by expanding an equilibrated amount of hydrogen gas from the manifold into the reactor, and the quantity of hydrogen absorbed was determined by ideal gas mole balance between the two volumes after equilibration. This procedure was repeated until no additional hydrogen uptake was observed. After completion of the measurement, the reactor and manifold were evacuated to  $10^{-7}$ torr before heating to the next temperature.

#### 3.1.2 Computations

Electronic structure calculations were performed for the singlet state of the cationic fragments  $[M(\eta^2 - H_2)(CO)dppe_2]^+$  and  $[M(CO)dppe_2]^+$  (M = Mn, Tc, Re), and the hydrogen molecule using the GAMESS-US software package.<sup>134</sup> Geometry optimizations for the organometallic fragments were performed using fully spin-restricted (RHF) density functional theory calculations, with the B3LYP exchange-correlation functional<sup>135, 136</sup> and the LANL2DZ basis set.<sup>137-140</sup> An additional *p* polarization shell for light atoms and *d* polarization shell for heavier atoms was added to augment the basis set. Effective core potentials<sup>138-140</sup> representing the core 10 electrons for manganese and phosphorous atoms, the core 28 electrons for technetium and the core 60 electrons for rhenium were used. SCF convergence was set to 5.0 x 10<sup>-6</sup> for all calculations. Geometry optimizations were performed to a tolerance of  $10^{-4}$  au.

The initial structure for  $[M(\eta^2-H_2)(CO)dppe_2]^+$  was taken from structural data reported previously for the manganese compounds by King, et al.<sup>100</sup> Two structures representing both an "open" geometry, in which the hydrogen molecule was removed from the H<sub>2</sub> adduct geometry, and the rearranged "agostic" geometry reported by King et al., in which ortho hydrogens from phenyl groups on the ligands form long-range bonds with the open coordination site, were used for  $[M(CO)dppe_2]^+$ . Differences between the relaxed structures obtained from calculations and the reported experimental geometries for the manganese complexes were unremarkable, beyond a contraction of all bond lengths within the structure attributable to thermal effects. The H-H bond in each dihydrogen complex was found to align parallel to the phosphorus-metal-phosphorus axis, as shown by the relaxed structure for  $[Mn(CO)(\eta^2-H_2)dppe_2]^+$  appearing in 3.1.



Figure 3.1: Relaxed structure of the  $[Mn(\eta^2-H_2)(CO)dppe_2]^+$  cation, showing the hydrogen molecule aligned along the P-Mn-P axis. Alkyl and aryl hydrogens have been removed for clarity.

Ab-initio ground state electronic energies were calculated using the spin-component scaled Møller-Plesset second-order perturbation<sup>141</sup> (SCS-MP2) scheme with the LANL2DZ-ECP basis set from the optimized geometries found from DFT calculations, which has been shown by Tomàs et al. to lead to accurate calculations for dihydrogen complexes.<sup>114,142</sup> To obtain thermal corrections to the energy and normal mode frequencies, the Hessian matrix for each fragment was calculated at the B3LYP/LANL2DZ-ECP level of theory using seminumerical methods, with contributions from positive and negative displacements of 0.01 Bohr. Vibrational analysis in the rigid rotor approximation was performed to obtain partition functions and normal modes of the fragments,  $q_{tot}$ , at 1 atm pressure. A scaling factor of 0.96 was used for the vibrational analysis to correct known errors in the LANL2DZ basis set, consistent with common practice.<sup>143</sup> The chemical potential of hydrogen gas was calculated from the partition function for the hydrogen molecule obtained from GAMESS, through the relationship:

$$\mu_{\rm gas}(T,P) = -kT\ln\left(q_{\rm tot}(T,P^\circ)\right) + \ln\left(\frac{P}{P^\circ}\right) \tag{3.1}$$

where k is Boltzmann's constant, T is the temperature in Kelvin, P is the pressure of the system, and  $P^{\circ}$  is the standard pressure of the system, taken to be 1 atm. Rotational degeneracy for the hydrogen molecule is included in the calculated partition function from the GAMESS software.

The corrected energy, E, for each fragment was obtained as the sum of the ground-state energy,  $\epsilon_0$ , and the thermal correction to the energy obtained from vibrational analysis,  $\epsilon_{corr}(T)$ , which contains contributions from translational, vibrational, and rotational motions of the molecule. The binding energy of the hydrogen molecule to the organometallic site was calculated from the corrected energies as:

$$\Delta E(T) = E_{\rm M-H_2}(T) - E_{\rm H_2}(T) - E_{\rm M}(T)$$
(3.2)

where the subscripts M-H<sub>2</sub>, H<sub>2</sub>, and M represent the organometallic adduct fragment, hydrogen gas, and the bare organometallic fragment, respectively.

#### **3.2** Results and Discussion

#### 3.2.1 Experimental Results

Absorption isotherms were measured for  $H_2$  onto  $[Mn(CO)dppe_2][BArF^{24}]$  at temperatures of 313K, 333K, 343K, 353K, 363K, and 373K, and are shown in 3.2. Reversible absorption capacity for all temperatures was approximately 0.35  $H_2$  molecules per manganese center. No irreversible absorption was observed. This capacity value was consistent throughout measurements at all temperatures, and was stable after multiple absorption/desorption cycles performed during the course of isotherm and kinetic measurements. These properties are consistent with previous reports for solid-state



Figure 3.2: Absorption isotherms for hydrogen gas onto  $[Mn(CO)dppe_2][BArF^{24}]$ . Solid lines are Langmuir isotherm fits to the data.

dihydrogen materials.<sup>125</sup>

Initial rate kinetics measurements were performed at 298K and 318K to determine the dependence of the absorption rate on pressure and quantity absorbed. The results for 318K appear in 3.3. The rate of change of the pressure was determined to be linearly dependent on the pressure and the remaining capacity, such that:

$$\frac{\partial P}{\partial t} = -kP(1-\theta)$$

where  $\theta$  is the fraction of the reversible absorption capacity absorbed, and k is the rate constant. Rate constants for 298K and 318K were 0.3090 s<sup>-1</sup> and 0.3116 s<sup>-1</sup>, respectively. The Arrhenius activation energy obtained from these values is  $E_{\rm a} = 330$  J/mol, indicating a very low activation barrier to absorption.

The isotherms presented in 3.2 do not demonstrate a maximum like with physisorptive materials,



Figure 3.3: Initial rate measurements for hydrogen absorption onto  $[Mn(CO)dppe_2][BArF^{24}]$  at 318K. Absorption kinetics were found to be linearly dependent on pressure and capacity. Linear regression gives a rate constant of k = 0.3661,  $R^2 = 0.9545$ .

but rather achieve a stable plateau at  $0.35 \text{ H}_2$  molecules per manganese. This behavior is particularly evident in the isotherm at 313K. The existence of the stable plateau in the experimental isotherms is consistent with total absorption<sup>144</sup> and indicates that hydrogen absorbs into the bulk material instead of adsorbing onto the surface. However, the absorption rate is comparable to physisorption materials, despite no additional processing used to increase the surface area of the solid or the exposure of manganese centers to the gas. This rapid absorption is likely due to the loose packing of the bulky organic ligands in the solid, which allows the small hydrogen molecule to diffuse rapidly through the structure.<sup>100</sup>

The rapid kinetics and high reversibility observed in the absorption process suggest a site-binding mechanism for the binding interaction of  $H_2$  with solid [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] that is similar to physisorptive materials. In organometallic dihydrogen complexes, the binding orientation of the ligands is considered fixed by the atomic orbitals of the metal center, producing distinct binding sites

for each ligand at the metal center. The isolation of the metal centers by the organic constituents ensures both non-interacting sites and chemical uniformity, suggesting the binding interaction at each site is isoenergetic. With this interpretation of the coordination site, we investigated the binding behavior by fitting the experimental isotherms to the Langmuir model. The Langmuir isotherm model assumes an established equilibrium between a mobile phase of guest molecules and fixed sites capable of binding a single molecule, with homogeneous, non-interacting, and isoenergetic interactions, representing an appropriate description of the organometallic coordination site if the behavior is similar to other site-binding mechanisms.

For the assumed ideal gas reaction:

$$[M(CO)dppe_2][BArF^{24}] + H_2 \rightleftharpoons [M(\eta^2 - H_2)(CO)dppe_2][BArF^{24}]$$

$$(3.3)$$

in which a single hydrogen molecule binds to the open coordination site on the metal, represented by M, the Langmuir isotherm is given by:

$$\frac{n}{N} = \frac{\alpha P}{1 + \alpha P} , \qquad \alpha = \frac{K_{\rm eq}}{RT}$$
(3.4)

where *n* is the number of bound hydrogen molecules, *N* is the total number of active binding sites, *P* is the pressure of the gas in atmospheres, *R* is the universal gas constant, *T* is the system temperature in Kelvin, and  $K_{eq}$  is the equilibrium constant for Equation 3.3.

Langmuir fits to the experimental data are shown as the solid lines in 3.2. The fit for 313K was obtained by varying both the total number of active binding sites, N, and  $\alpha$  as fitting parameters. A value of 0.344 H<sub>2</sub> per manganese center was found for N at 313K, and was held constant for the fits at all other temperatures. The values for  $K_{eq}$  derived from the fitted values for  $\alpha$  and the R<sup>2</sup> values of the fits appear in 3.1. As can be seen from the figure and the R<sup>2</sup> values, the Langmuir model very accurately describes hydrogen absorption by the manganese complex, indicating that the open coordination sites within [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] interact with hydrogen gas through a site-binding mechanism. The standard enthalpy and entropy of the absorption interaction are readily obtained from the values of the equilibrium constant in Equation 3.4 through the general form of the van't Hoff equation:

$$\ln K_{eq} = -\frac{\Delta H^{\circ}}{RT} + \frac{\Delta S^{\circ}}{R}$$
(3.5)

where  $\Delta S^{\circ}$  and  $\Delta H^{\circ}$  are the entropy and enthalpy of the absorption reaction, as defined by Equation 3.3, at standard conditions of one atmosphere pressure and 298K. A van't Hoff plot of the equilibrium constants from the Langmuir fits appears in 3.4. Linear regression gives values of  $\Delta H^{\circ} = -52.2$ kJ/mol and  $\Delta S^{\circ} = -99.6$  J/mol-K for hydrogen absorption over [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>]. This enthalpy is similar to experimental values found by Gonzalez and Hoff<sup>95</sup> for hydrogen binding to the group 6 complexes M(CO)<sub>3</sub>(PCy<sub>3</sub>)<sub>2</sub> (M = Cr, Mo, W) by solution calorimetry, and represents an interaction of comparable strength to chemisorption seen in the formation of metal hydrides. In addition, a significant increase in the absorption entropy from the standard<sup>145</sup> -131 J/mol-K typical for hydrogen storage materials is observed. The higher enthalpy and entropy values allow [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] to bind hydrogen effectively near room temperature while exhibiting the rapid absorption kinetics and long cycle life characteristic of physisorption materials, a combination of properties that are highly favorable for hydrogen storage applications.

The kinetic absorption behavior is consistent with observations made by Gonzalez, et al.<sup>123</sup> that the kinetics of absorption and dissociation of  $H_2$  with  $W(CO)_3(PCy_3)_2$  in solution were several times faster than oxidative (dissociative) addition, which also supports the site-binding model. The

Table 3.1: Equilibrium constants and  $R^2$  values from experimental Langmuir fits and simulated Langmuir isotherms for  $H_2$  absorption onto  $[Mn(CO)dppe_2]^+$ 

Temperature	$K_{eq,exp}, atm^{-1}$	$K_{eq,sim}, atm^{-1}$	$\mathbf{R}_{exp}^2$	$\mathbf{R}^2_{sim}$
313K	3241.74	3117.22	0.9702	0.9692
333K	1013.12	926.34	0.9882	0.9902
343K	581.82	528.31	0.9919	0.9892
353K	371.65	309.63	0.9937	0.9766
363K	214.78	186.08	0.9962	0.9800
373K	124.19	114.37	0.9914	0.9808



Figure 3.4: Van't Hoff plot of the equilibrium constants derived from the Langmuir model. Linear regression gives  $\ln K_{eq} = 6280.6/\text{T} - 11.986$ ,  $R^2 = 0.9985$ , giving thermodynamic properties  $\Delta H^{\circ} = -52.2 \text{ kJ/mol}$  and  $\Delta S^{\circ} = -99.6 \text{ J/mol-K}$ .

solution results also imply that the site-binding mechanism is not limited to the solid state. Qualitative assessments of affinities based on competitive ligand binding would thus require control of the local hydrogen concentration, particularly in solution state measurements where the concentration of dissolved hydrogen depends on the solvent used. This additional aspect needs to be considered before using such assessments to disqualify materials as potential candidates.

#### 3.2.2 Computational Results

The physics of the Langmuir model are well defined at the molecular scale and calculation of the isotherms from simulated binding energies is straightforward, allowing for direct comparison between experimental and computational results to establish the accuracy of computational methods. From statistical mechanics, the fractional coverage of absorbed molecules at the binding sites in the Langmuir model is:

$$\frac{\langle n \rangle}{N} = -\left(\frac{\partial W}{\partial \mu}\right)_{\beta} = \frac{q_s e^{\beta(\mu_{gas} - \Delta E)}}{1 + q_s e^{\beta(\mu_{gas} - \Delta E)}} \tag{3.6}$$

where  $\beta = (kT)^{-1}$ , W is the grand potential of the bound system,  $\langle n \rangle$  represents the ensemble average number of absorbed molecules, and  $q_s e^{\beta(\mu_{gas} - \Delta E)}$  is the partition function of the molecular species at the binding site. The pressure dependence of the fractional occupancy is expressed through the chemical potential of the gas, as shown in Equation 3.1. The values of  $\epsilon_{corr}(T)$  obtained from GAMESS contain the contributions to the partition function from the normal modes of hydrogen at the binding site. However, because symmetry is calculated based on the entire organometallic fragment, the output from GAMESS does not treat rotational symmetry of the hydrogen molecule appropriately, and the variable  $q_s$  in Equation 3.6 is used as a correction term to account for degeneracy in the rotational mode.

Carbonyl stretch frequencies obtained from the vibrational analysis were found to be 1824.32, 1820.56, and 1849.69 cm<sup>-1</sup> for the open configuration of  $[Mn(CO)dppe_2]^+$ , the agostic configuration, and  $[Mn(CO)(\eta^2-H_2)dppe_2]^+$ , respectively, in good agreement with the literature values.<sup>100,115</sup> The standard Gibbs free energy for the transformation from the open configuration to the agostic configuration was also calculated from the vibrational analysis, and was found to be  $\Delta G^\circ = -39.6$ kJ/mol, similar to values found for other agostic interactions.<sup>123</sup>

Six primary normal modes for dihydrogen motion at the binding site were defined as the modes having the largest magnitude eigenvectors on the dihydrogen molecule as obtained from the output of vibrational analysis on each adduct fragment. For the manganese complex, these modes correlated well with literature values obtained from inelastic neutron scattering.<sup>100</sup> In all three cases only the rotational mode, with frequencies of 196.79 cm<sup>-1</sup>, 269.2 cm<sup>-1</sup>, and 556.7 cm<sup>-1</sup> for Mn, Tc, and Re, respectively, was found to be significantly populated at the temperatures of interest. The symmetry of the rotational mode was determined from analysis of the components of the eigenvector associated with the atoms of the hydrogen molecule, and the mode was defined as symmetric if the x, y, and z components were opposite in sign and had differences in their magnitude of less than 0.1 millidyne/angstrom. Within this definition, dihydrogen rotational modes in the manganese and



Figure 3.5: Comparison of simulated isotherms for  $H_2$  absorption onto  $[Mn(CO)dppe_2]^+$  (solid black lines) to experimental data for  $[Mn(CO)dppe_2][BArF^{24}]$ .

technetium adducts were found to be symmetric, while rotation in the rhenium adduct was asymmetric with respect to the center of mass of the hydrogen molecule. To correct for the degeneracy of the symmetric modes in the manganese and technetium adducts,  $q_s$  was defined such that:

$$\frac{1}{q_{\rm s}} = \frac{1}{2} \left( \frac{e^{-\frac{\Theta_v}{2T}}}{1 - e^{-\frac{\Theta_v}{T}}} \right) \tag{3.7}$$

where  $\Theta_v = h\nu/k$  is the characteristic mode temperature obtained from the rotational mode frequency,  $\nu$ . This correction removes half of the mode contribution from the total partition function, accounting for the reduction in states due to symmetric rotation. For rhenium, a value of  $q_s = 1$ was used, since no degeneracy exists in the asymmetric rotation mode.

Simulated equilibrium constants and standard binding energies for absorption onto both the open and agostic configurations appear in 3.1 and 3.2, and simulated isotherms for the manganese fragment in the open geometry appear as solid lines against the experimental data in 3.5. Simulated isotherms

Compound		$\Delta E^{\circ}$	$\Delta H^{\circ}$	$\Delta S^{\circ}$	$S_{rot}(298K)$	$S_{rot}(373K)$
$[Mn(CO)dppe_2]^+$						
	open	-43.0	-53.5	-103.8	3.20	5.04
	agostic	-1.5	-11.5	-103.9		
$[Tc(CO)dppe_2]^+$						
	open	-38.1	-49.3	-103.5	0.86	2.60
	agostic	-8.51	-18.7	-103.0		
$[Re(CO)dppe_2]^+$						
	open	-50.6	-57.5	-97.4	2.17	3.39
	agostic	-20.4	-27.1	-96.7		

Table 3.2: Calculated energies, enthalpies, and entropies of the  $H_2$  absorption reaction at 1 atm, 298K and rotational entropy of the bound states at 298K and 373K. Energy and enthalpy values are given in kJ/mol, and entropy values are given in J/mol-K.

for the open geometry are nearly identical to fits for the experimental data, indicating that the open geometry sites are the active sites for the absorption reaction. The simulated enthalpy and entropy values, calculated from Equations 3.4, 3.5, and 3.6 are  $\Delta H^{\circ} = -53.5$  kJ/mol and  $\Delta S^{\circ} = -103.8$ J/mol-K, demonstrating that both pressure-composition behavior and thermodynamic properties can be accurately simulated through electronic structure calculations using the B3LYP/LANL2DZ level of theory for geometry optimization and vibrational analysis, and the MP2/LANL2DZ level of theory for the electronic energy used to obtain the binding energy.

Calculations for absorption onto the agostic geometry validate the observation by King et al.<sup>100</sup> that direct hydrogenation of the agostic manganese complex in the solid state fails under one atmosphere of hydrogen pressure. Evaluation of Equation 3.6 for the agostic complex at one atmosphere hydrogen pressure and 313K gives a fractional coverage of  $1.443 \times 10^{-5}$ , indicating that molecules in the agostic configuration do not contribute to hydrogen binding in the experimental results. The experimental result of 0.344 for the fraction of active manganese centers evidently represents the fraction of the total manganese centers in the open configuration. This number was maintained throughout all temperatures studied, indicating that no exchange between the agostic and open configurations occurs in the solid, despite the favorable value of the free energy for the agostic complex found from calculation. Solution NMR results reported by King, et al. for the same compound<sup>115</sup> in CH<sub>2</sub>Cl<sub>2</sub> found a similar 37% of the manganese centers were active for N<sub>2</sub> absorption at room temperature. It is likely that the precipitation of the N<sub>2</sub> adduct and subsequent release of the

 $N_2$  molecule produce the open configuration in the solid state necessary for  $H_2$  absorption. Once precipitated, the thermal stability of the active site fraction suggests that kinetic stabilization in the solid state prevents exchange between the two configurations. This stabilization would prevent hydrogen absorption over agostic complexes in the solid state even if favorable thermodynamics were predicted, due to the inability of the ligand to leave the binding site.

With the accuracy of the calculation method established, standard energies, enthalpies, and entropies were obtained from calculation for hydrogen absorption over  $[Tc(CO)dppe_2]^+$  and  $[Re(CO)dppe_2]^+$ . These values appear in 3.2. The resulting binding enthalpies for the open configuration are similar in strength to findings for other dihydrogen complexes. The trends in the binding enthalpies and entropies in the group 7 metals from these simulations are in good agreement with the solution calorimetry results for group 6 metals reported by Gonzalez and Hoff,<sup>95</sup> with the enthalpy increasing as 5d > 3d > 4d. Quantitatively, the energetic difference between 3d and 4d metals matches with the value found for the group 6 metals, despite the differences in group, molecular charge, and chemical environment. A slightly lower energetic difference of 4 kJ/mol between manganese and rhenium complexes was found relative to the reported 11 kJ/mol difference between chromium and molybdenum complexes. The good quantitative agreement implies a consistent trend in the thermodynamic values within the transition metals. Unfortunately, no direct cause for this relationship was evident from the calculations. The relationships are expected to rely upon the relative  $\sigma$ - and  $\pi$ - donation ability at the metal center, a complex property dependent upon the metal-ligand interactions and geometry.<sup>113</sup> Although several studies have examined the governing principles behind these properties, their exact nature remains uncertain.<sup>146–148</sup>

The individual contribution of excited rotational modes to the site entropy can be calculated from Equation 3.7 through the expression:

$$S_{rot} = k \left( \ln q_s + T \frac{\partial \ln q_s}{\partial T} \right)$$
(3.8)

Values for the rotational entropy of the bound state at 298K and 373K for each of the complexes appear in 3.2. Since only the bound state is considered, the value is equivalent for both the open

and agostic configurations. The softer rotational mode in the manganese adduct leads to increased entropic contributions at all temperatures, and an increased dependence on temperature relative to the other two complexes. The near equivalence of the standard entropies of technetium and manganese, despite a much lower contribution from rotation of the hydrogen molecule in technetium, is due to increased contributions from other modes within the organometallic complex. The asymmetry of the rotation in the rhenium complex leads to higher contributions to the entropy than in the technetium adduct, despite the high stiffness in the mode. The much stiffer, asymmetric rotational mode in the rhenium complex, along with the higher binding enthalpy, suggest that the H<sub>2</sub> molecule may have undergone oxidative addition and dissociated to form the dihydride complex. However, because both hydrogen atoms remain bound to the same metal center, the assumptions of the Langmuir isotherm are not violated. Effects from this transition are modeled in the Langmuir isotherm as a contribution to the site binding energy. This allows the absorption enthalpy to be modeled accurately by the Langmuir isotherm in cases of oxidative addition, but does not allow determination of whether the compound is a dihydrogen complex or a dihydride.

#### 3.3 Conclusions

We report experimental and computational results on the properties of dihydrogen complexes, and interpretations with the Langmuir isotherm model. Langmuir fits for the absorption of hydrogen onto [Mn(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] to form the dihydrogen complex [Mn( $\eta^2$ -H<sub>2</sub>)(CO)dppe<sub>2</sub>][BArF<sup>24</sup>] were found to accurately describe the experimental pressure-composition behavior, giving an enthalpy and entropy for the binding interaction of  $\Delta H^{\circ} = -52.2$  kJ/mol and  $\Delta S^{\circ} = -99.6$  J/mol-K, in general agreement with previous observations of model Kubas complexes. While the complexes within this study do not meet gravimetric or volumetric capacity expectations for vehicular hydrogen storage, they exhibit a unique binding motif capable of storing hydrogen at desirable temperatures and pressures with kinetics rivaling physisorption materials. The combination of these properties holds strong promise for the development of lower weight organometallic storage materials for applications.

Computationally derived isotherms using a mixed calculation scheme of B3LYP/LANL2DZ-

ECP for geometry optimization and vibrational analysis and MP2/LANL2DZ-ECP for total energy were also found to accurately describe the experimental behavior of the manganese complex and have provided additional insight into the configuration of the active binding site and the individual contributions from configurational and rotational entropy to the binding properties. Calculations for substitution of the metal center with other group 7 metals found similar trends in the thermodynamic properties to experimental reports for group 6 metals, with increasing binding enthalpy such that 5d > 3d > 4d, suggesting an overall trend in properties for transition metals moving down the periodic table.

## Chapter 4

# Studies of $[MH(\eta^2-H_2)dppe_2]^+$ (M = Fe, Ru, Os) Complexes

In the previous chapter, the Langmuir isotherm model was found to accurately describe the experimental behavior for hydrogen absorption onto the complex  $[Mn(CO)dppe_2][BArF^{24}]$  to form the dihydrogen complex  $[Mn(\eta^2-H_2)(CO)dppe_2][BArF^{24}]$ . Electronic structure simulations of hydrogen absorption over the fragment  $[Mn(CO)dppe_2]^+$  were also successful in quantitatively reproducing the experimental behavior and thermodynamic properties, and simulations were used to extend the analysis to the other group 7 metal fragments  $[Tc(CO)dppe_2]^+$  and  $[Re(CO)dppe_2]^+$  to develop trends for the hydrogen absorption behavior within the periodic table. In this chapter, the analysis is extended to describe hydrogen binding to the isoelectronic organometallic fragments  $[MHdppe_2]^+$ (M = Fe, Ru, Os) to form the dihydrogen complexes  $[MH(\eta^2-H_2)dppe_2]^+$ .

The group 8 complexes containing the cationic fragment  $[MH(\eta^2-H_2)dppe_2]^+$  (M = Fe, Ru, Os) are among the most well-studied dihydrogen compounds available in the literature. The crystal structures for all three dihydrogen complexes have been determined by x-ray diffraction<sup>149–152</sup> and found to be isostructural, and neutron diffraction experiments have been performed for the iron and ruthenium complexes to determine dihydrogen positions.<sup>149,150</sup> Extensive NMR work has been performed<sup>96</sup> to examine the dynamics of dihydrogen exchange, heterolytic cleavage, and intramolecular exchange of the hydride and dihydrogen within all three dihydrogen complexes. Multiple spectroscopic studies have been performed to characterize the iron complex, including Mössbauer spectroscopy<sup>119</sup> and analysis of the vibrational modes through inelastic neutron scattering.<sup>110</sup>
Thermodynamic properties of the dihydrogen interaction with the metal center remain undetermined, largely because of the high stability of the dihydrogen products and strong reactivity of the five-coordinate species in the iron and osmium compounds that prevent characterization. While Mössbauer, UV-vis and IR spectroscopy have been reported for the directly synthesized fivecoordinate iron complex<sup>107</sup> and  $H_2/D_2$  exchange has been shown to occur through a stable intermediate,<sup>96</sup> the thermal decomposition products of the iron and osmium species have not been directly identified and have only been assigned through the analogous [RuHdppe<sub>2</sub>]<sup>+</sup> cation, which is stable at room temperature. No characterization studies are available for [OsHdppe<sub>2</sub>]<sup>+</sup> or [RuHdppe<sub>2</sub>]<sup>+</sup> complexes, despite the stability of the ruthenium species.

Identification of the complexes involved in reversible hydrogen absorption is the first step required for accurate thermodynamic simulations. Mössbauer spectroscopy, in particular, provides a probe of the valence and spin state of the central iron atom in iron complexes, as well as a measure of the charge transfer between the metal center and attached ligands. The availability of this information and the atomic selectivity of the technique makes Mössbauer spectroscopy uniquely suited for investigating group 8 dihydrogen complexes, and for determining mechanisms in Kubas binding. Evidence for the identification of the thermal decomposition product of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  $(NTf_2 = bis(trifluoromethlysulfonyl)imide)$  through Mossbauer spectroscopy and other techniques is presented below, and simulated thermodynamic parameters are reported for the hydrogen binding interaction in the group 8 fragments  $[MH(\eta^2-H_2)dppe_2]^+$  (M = Fe, Ru, Os).

#### 4.1 Experimental

Unless otherwise stated, all reactions were performed under a dinitrogen atmosphere using either a controlled atmosphere glovebox or Schlenk line techniques. 1,2-bis(diphenylphosphino)ethane was purchased from Strem Chemicals and used without further purification. Crystalline bis(trifluoro-methanesulfonyl)imide was purchased from Acros Organics and used without further purification. Research-grade gases were purchased from Matheson and used directly. All solvents were dried and deoxygenated by purging with dry dinitrogen gas for 15 minutes before passing through packed

columns of activated alumina and activated copper. After synthesis, materials were stored under dry argon in an atmosphere-controlled glovebox until their use in testing.

Solution-state NMR spectra were recorded on a Varian 300 MHz instrument with <sup>1</sup>H shifts reported relative to the residual solvent peak, and <sup>31</sup>P peaks reported relative to 85% H<sub>3</sub>PO<sub>4</sub>. Deuterated NMR solvents were purchased from Cambridge Isotopes Laboratories. Deuterated benzene was purified by vacuum distillation from a sodium/benzophenone solution before use. Deuterated bromobenzene was dried over CaH<sub>2</sub> and vacuum distilled before use.

**Preparation of** *cis*-**FeH**<sub>2</sub>**dppe**<sub>2</sub>—This procedure was a modification of existing literature procedures.<sup>96, 153</sup> Anhydrous FeCl<sub>2</sub> (319.8 mg, 2.523 mmol) was suspended in 2 mL THF. A solution of 1,2-bis(diphenylphosphino)ethane (2.0052 g, 5.033 mmol) in 6 mL THF was added and the suspension was stirred until a milky white precipitate formed. NaBH<sub>4</sub> (220.0 mg, 5.680 mmol) in 8 mL of dry ethanol was then added, producing strong outgassing and causing all components to immediately dissolve to form a dark red solution. The solution was stirred for 3 hours during which outgassing continued and a yellow powder precipitated. The remaining solution was decanted from the precipitate, and the precipitate was dissolved in benzene and filtered over celite to remove salt impurities. Removal of the solvent *in vacuo* produced the product as a fine yellow powder. Yield 1.8533 g (73.5% on iron). <sup>1</sup>H NMR spectra was broad, indicating a fluxional molecule. <sup>1</sup>H NMR (300 MHz) in C<sub>6</sub>D<sub>6</sub>:  $\delta$  -12.76 (quad., 2H); 1.80 (s, 4H); 2.14 (s, 2H); 2.40 (s, 2H); 6.44 (s, 4H); 6.58–7.12 (m, 20H); 7.14–7.80 (m, 12H); 8.67 (s, 4H). <sup>31</sup>P NMR (300 MHz) in C<sub>6</sub>D<sub>6</sub>:  $\delta$  91.5 (s); 103.3 (s).

**Preparation of**  $[\text{FeH}(\eta^2-\text{H}_2)\text{dppe}_2][\text{NTf}_2]$ —This procedure was a modification of an existing literature procedure.<sup>96</sup> Bis(trifluoromethylsulfonyl)imide (327.8 mg, 1.166 mmol) in 5mL diethyl ether was added to a solution of *cis*-FeH<sub>2</sub>dppe<sub>2</sub> (1.015 g, 1.188 mmol) in 12 mL diethyl ether and allowed to stir for one hour, forming a light green precipitate. The precipitate was filtered over a medium frit and washed twice with diethyl ether. Removal of residual diethyl ether *in vacuo* produced the product as a light green powder. Yield 1.098 mg (81.8% on iron). Some N<sub>2</sub> adduct was included in the product. <sup>1</sup>H NMR (300 MHz) in C<sub>6</sub>D<sub>5</sub>Br:  $\delta$  -12.61 (s, broad, 1H); -7.76 (s, broad, 2H); 2.13 (s, 8H); 7.16–7.51 (m, 40H). <sup>31</sup>P NMR (300 MHz) in C<sub>6</sub>D<sub>5</sub>Br:  $\delta$  92.0 (s).

Thermal decomposition of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$ — $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  was placed under vacuum and heated at 125° for 12 hours, producing an extremely reactive blue-green powder. This powder was found to react with all available solvents, preventing direct characterization by NMR.

#### 4.1.1 X-Ray Diffraction

Diffraction quality crystals of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  were grown by layering diethyl ether over a saturated solution of the complex in benzene. The resulting yellow needles were mounted on glass fiber with Paratone oil and spectrum were recorded over 12 hours with a Bruker KAPPA APEXII x-ray diffractometer under flowing nitrogen at 298K using Mo K $\alpha$  radiation.

#### 4.1.2 Mössbauer Spectroscopy

Isotopically enriched [<sup>57</sup>FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][NTf<sub>2</sub>] and its thermal decomposition products were made according to the procedures above. <sup>57</sup>Fe metal was purchased from Isoflex USA. <sup>57</sup>FeCl<sub>2</sub> was made by reacting shavings of <sup>57</sup>Fe (0.13 mg, 0.228 mmol) metal with excess concentrated hydrochloric acid (1.0 mL, 12M) in 5 mL methanol. The resulting yellow solution was reduced and dried at 160° under vacuum for 8 hours to produce the anhydrous starting material.

Samples were loaded into a custom-built, O-ring sealed PTFE sample cell under an argon atmosphere. Sealed samples were stable for up to 24 hours in air at room temperature. Spectra were recorded over six hours in the transmission configuration using a gas scintillation detector and source radiation from the decay of <sup>57</sup>Co in a rhodium matrix. Isomer shifts were referenced to bcc iron.

#### 4.1.3 Computation

Electronic structure calculations were performed for the singlet state of the cationic fragments  $[MH(\eta^2-H_2)dppe_2]^+$  and  $[MHdppe_2]^+$  (M = Fe, Ru, Os), and the hydrogen molecule using the GAMESS-US software package.<sup>134</sup> Geometry optimizations for the organometallic fragments were

performed using fully spin-restricted (RHF) density functional theory calculations, with the B3LYP exchange-correlation functional<sup>135,136</sup> and the LANL2DZ basis set.<sup>137–140</sup> An additional p polarization shell for light atoms and d polarization shell for heavier atoms were added to augment the basis set. Starting structures for optimizations were taken from XRD experiments reported previously for  $[FeH(\eta^2-H_2)dppe_2][BF_4]^{152}$  and on calculated singlet structures<sup>107</sup> for  $[FeHdppe_2]^+$ . Effective core potentials<sup>138–140</sup> representing the core 10 electrons for iron and phosphorus atoms, the core 28 electrons for ruthenium and the core 60 electrons for osmium were used. SCF convergence was set to  $5.0 \times 10^{-6}$  for all calculations. Geometry optimizations were performed to a tolerance of  $10^{-4}$  au.

Ab initio ground state electronic energies were calculated using the spin-component scaled Møller-Plesset second order perturbation<sup>141</sup> (SCS-MP2) scheme with the LANL2DZ-ECP basis set from the optimized geometries found from DFT calculations. Thermal corrections to the energy and normal mode frequencies were obtained from vibrational analysis of the Hessian matrix for each fragment calculated using the B3LYP/LANL2DZ-ECP level of theory with seminumerical methods, with contributions from positive and negative displacements of 0.01 Bohr. Partition functions of the fragments,  $q_{tot}$  at 1 atm pressure were also obtained from vibrational analysis. A scaling factor of 0.96 was in calculated the thermodynamic properties and normal mode frequencies to correct known errors in the LANL2DZ basis set, consistent with common practice.<sup>143</sup> Symmetry of the rotational modes was determined by analysis of the components of the mode eigenvector associated with the two hydrogen atoms, and the mode was considered symmetric if the x, y, and z components of the force constants for each atom were of opposite sign and deviated in value by less than 0.1 millidyne/angstrom.

The chemical potential of hydrogen gas was calculated from the partition function for the hydrogen molecule, through the relationship:

$$\mu_{\rm gas}(T,P) = -kT\ln\left(q_{\rm tot}(T,P^\circ)\right) + \ln\left(\frac{P}{P^\circ}\right) \tag{4.1}$$

where k is Boltzmann's constant, T is the temperature in Kelvin, P is the pressure of the system, and  $P^{\circ}$  is the standard pressure of the system, taken to be 1 atm. Rotational degeneracy for the hydrogen molecule is included in the calculated partition function from the GAMESS software.

The energy for each fragment, E, was obtained as the sum of the ground-state energy,  $\epsilon_0$ , and the thermal correction obtained from vibrational analysis,  $\epsilon_{corr}(T)$ , which contains contributions from the electronic, translational, vibrational, and rotational motions of the molecule. The binding energy of the hydrogen molecule to the organometallic site,  $\Delta E$ , was calculated as:

$$\Delta E(T) = E_{M-H_2}(T) - E_{H_2}(T) - E_M(T)$$
(4.2)

where the subscripts M-H<sub>2</sub>, H<sub>2</sub>, and M represent the organometallic adduct fragment, hydrogen gas, and the bare organometallic fragment, respectively.

To obtain the Mössbauer parameters for the iron fragments, fully *ab initio* geometry optimizations and total energy calculations were performed using the TZVP level of theory for  $[FeH(\eta^2-H_2)dppe_2]^+$ and  $[FeHdppe_2]^+$ , with p and d polarization shells as described above. Calculations were also performed on the nitroprusside anion,  $[Fe(NO)CN_5]^{2-}$ , to be used as a reference state for isomer shifts. Isomer shift and electric quadrupole splitting values were obtained from the electronic charge density and electric field gradient at the iron nucleus according to the method described by Blaha.<sup>154</sup> Isomer shifts were converted from the nitroprusside scale to the bcc iron scale by subtracting 0.257 mm/s.

#### 4.1.4 Kinetic Isotherms

Kinetic isotherms were recorded for the thermal release of the H<sub>2</sub> ligand from  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$ using a custom-built Sieverts apparatus. Solid  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  (257 mg) was loaded into a 14mL stainless steel reactor under an argon atmosphere and sealed before transfer to the instrument. Swagelok VCR copper filter gaskets with 2  $\mu$ m stainless steel filters were used to prevent loss of the powdered sample from the reactor during measurements. After transferring the reactor to the assembly, samples were allowed to pump down overnight to  $3.1 \times 10^{-7}$  torr at the pump inlet before measurements were performed.

The rate of hydrogen release was measured by placing the sample under vacuum conditions

for nominally one hour at the desired temperature, followed by expansion of hydrogen gas from a calibrated volume into the reactor. Residual gas mass spectroscopy was performed during the evacuation of the reactor to identify the decomposition products. Ideal gas mole balance was used to determine the quantity reabsorbed after exposure of the degassed sample to 0.5 atm hydrogen gas. Time and pressure resolution of the instrument were 500 ms and 5 torr, respectively.

#### 4.2 **Results and Discussion**

#### 4.2.1 Experimental Results

Thermodynamic isotherms such as those measured for the isoelectronic manganese complexes could not be obtained for  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  in reasonable timescales. Equilibrium pressures from the samples were below detectable levels for all temperatures measured, and reabsorption of hydrogen under 0.5 atmospheres occurred faster than the time resolution of the instrument, preventing quantification of the absorption properties. Alternatively, kinetic rates of desorption were measured between 350K and 410K, and the results appear in Figure 4.1. An Arrhenius fit to the kinetic data gave an activation energy of  $E_A = 32.3 \text{ kJ/mol}$  for the release process, representing a significantly higher release barrier than expected from the previous manganese studies but lower than reported kinetics of dihydrogen release rates for  $W(CO)_3(PCy_3)_2$  in solution,<sup>155</sup> suggesting a wide range of kinetic behavior for dihydrogen complexes. During the course of kinetic testing, the complex underwent multiple absorption/desorption cycles without observed decomposition in the capacity, and residual gas mass spectroscopy confirmed that hydrogen gas was the only volatile decomposition product produced. Irreversible decomposition occurred at 420K, producing a black solid mixture of multiple decomposition products and hydrogen gas.

Single crystals of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  were grown to confirm the structure of the complex and assess any effects of the anion on the structure. The structure refined from the x-ray diffraction pattern appears in Figure 4.2. As shown, the organometallic cation retains the same conformation as in structures reported<sup>152, 149</sup> for  $[FeH(\eta^2-H_2)dppe_2][BF_4]$  and  $[FeH(\eta^2-H_2)dppe_2][BPh_4]$ . The bistri-



Figure 4.1: Kinetic measurements for hydrogen release from the reversible thermal decomposition of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$ . The sample was desorbed at the temperature shown from a state of full coverage for nominally one hour before exposure to hydrogen gas. The rate of release was calculated as the amount of gas reabsorbed divided by the time under vacuum. All data points were acquired on the same sample. Arrhenius fit gives  $E_A = 32.3 \text{ kJ/mol}$ ,  $R^2 = 0.8659$ .

flimide anion is highly disordered and positioned directly over the hydrogen coordination sites on the iron, but remains outside of the first coordination sphere. During reversible hydrogen release from the solid state, kinetic barriers to structural rearrangement are expected to prevent recombination of the cation-anion pair and produce the organometallic species [FeHdppe<sub>2</sub>][NTf<sub>2</sub>], as has been shown in the analogous ruthenium species.<sup>96</sup> However, the strong disorder associated with the bistriflimide anion, which is typically used for ionic liquid formulations, may lead to a lower activation barrier and a lower recombination temperature in the solid state than would be expected for a comparably sized anion, which may be the cause of the observed irreversible transition at 420K.

Thermal decomposition by direct heating of the  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  crystals under vacuum was attempted by the methods described in the procedures. This process produced blue needles of the same shape as the original crystals, with no degradation of the crystal quality observable by



Figure 4.2: ORTEP structural drawing of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  obtained from single crystal x-ray diffraction measurements at 298K. Ellipsoids represent 50% probability. Hydrogen atom positions were not resolvable. An additional benzene molecule co-crystallized within the structure has been removed for clarity.

optical microscopy. The new crystals were stable under an argon atmosphere for several hours but were highly reactive to trace air and moisture, changing color to form red crystals within a flame-sealed capillary over 24 hours through reaction with residual contaminants. As a result of this extraordinary reactivity, x-ray diffraction patterns could not be obtained and the structure of the decomposition product could not be determined.

Mössbauer spectra were obtained from samples of thermally decomposed [ $^{57}$ FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][NTf<sub>2</sub>] before and after reabsorption of hydrogen to help identify the cycling products. These spectra appear in Figure 4.3 and the values for the isomer shifts and electric quadrupole splittings, along with the literature values for the directly synthesized [FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][BF<sub>4</sub>] and [FeHdppe<sub>2</sub>][BPh<sub>4</sub>], appear in Table 4.1. Despite the harsh conditions used, only partial decomposition of the dihy-

Complex	IS, $mm/s$	EQS, $mm/s$
$[FeH(\eta^2-H_2)dppe_2][NTf_2]$	+0.0532	0.5695
$[FeH(\eta^2-H_2)dppe_2][BF_4], lit.^{119}$	-0.097	-0.56
$[FeH(\eta^2-H_2)dppe_2]^+$ , sim.	+0.3411	-0.7774
$[FeHdppe_2][NTf_2]$	+0.1552	1.5104
$[FeHdppe_2][BPh_4], lit.^{107}$	+0.2274	1.5178
$[FeHdppe_2]^+$ , singlet, sim.	-0.3693	-0.3961
$[FeHdppe_2]^+$ , triplet, sim.	-0.2726	+1.3383
$[Fe(NO)CN_5]^{2-}$ , lit. <sup>160</sup>	reference	+1.726
$[Fe(NO)CN_5]^{2-}$ , sim.	reference	+1.5620

Table 4.1: Isomer shift and electric quadrupole splitting values of iron complexes obtained from Mössbauer spectroscopy. For EQS values, a sign designation is absent if the sign was not directly determined by the measurement.

drogen complex was observed. Electric quadrupole splitting (EQS) values for the major peaks in the thermally decomposed and reabsorbed samples correspond well to literature values for the fivecoordinate species and the H<sub>2</sub> adduct, respectively. The spectra for the partially decomposed sample contains the spectra for [FeHdppe<sub>2</sub>][NTf<sub>2</sub>] and [FeH( $\eta^2$ -H<sub>2</sub>)dppe<sub>2</sub>][NTf<sub>2</sub>] as the active species, suggesting either a direct transition between the two species or that intermediates produced during the decomposition are unstable. The small, broad peak visible at ~2 mm/s in both spectra indicates the presence of unreacted <sup>57</sup>FeCl<sub>2</sub>.

EQS values describe perturbations of the electric field around the metal center due to variations in the chemical environment arising from the attached ligands, specifically asymmetric perturbations, and are less sensitive to changes in the cation-anion pairing. The similarity of the EQS values to the literature values thus provides strong evidence for the presence of the [FeHdppe<sub>2</sub>]<sup>+</sup> cation in the decomposed sample. The large EQS value for this complex is characteristic of nonbonding electrons in a high-spin d<sup>6</sup> configuration, which corresponds to the paramagnetism of the [FeHdppe<sub>2</sub>]<sup>+</sup> cation found through magnetic susceptibility measurements performed by Franke, et al.<sup>107</sup> However, while isomer shift values for the study complexes deviate from the literature values, they remain within the range associated with low-spin Fe<sup>II</sup> complexes, suggesting that the typical reduction in the s-electron density in high spin states is compensated for by donation from the attached ligands.



Figure 4.3: Mössbauer spectra for a) thermally decomposed  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  and b) thermally decomposed  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$ . Thermally decomposed spectrum shows partial decomposition to the five-coordinate  $[FeHdppe_2][NTf_2]$  species. Dashed lines correspond to the peak positions for the dihydrogen species. Both spectra also contain a small, broad doublet between 0.5 and 2.0 mm/s, corresponding to unreacted  ${}^{57}FeCl_2.{}^{156}$ 

#### 4.2.2 Computational Results

#### 4.2.2.1 Mössbauer Simulations

With the crystal structure for [FeHdppe<sub>2</sub>]<sup>+</sup> complexes unavailable from experiments, computational methods must be used to help identify the structure and electronic character of the molecule. Computational studies to determine the ground-state structure of [FeHdppe<sub>2</sub>]<sup>+</sup> have previously been performed by Franke, et al.,<sup>107</sup> and the singlet and triplet states of the molecule fragment were reported to be nearly isoenergetic, with the singlet state slightly more favorable. These states were also reported to have very different molecular geometries, which are expected to have a strong impact on the EQS values obtained through Mössbauer spectroscopy. Therefore, comparison between simulated Mössbauer parameters for the singlet and triplet states and experimental values can be

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used to determine which of the spin states is preferred at room temperature.

Simulations of the Mössbauer EQS values were performed, and the results appear in Table 4.1. Geometries of the singlet and triplet five-coordinate states were similar to those found by Franke, et al. The principal component of the electric field gradient was found to be oriented primarily along the iron-hydride axis in each of the simulations, with a slight skew in the triplet-state five-coordinate complex towards the empty coordination state, which no longer aligns with the hydride in the skewed geometry. The expected differentiation in the EQS values between the singlet state and triplet state was observed, and the value for the triplet state was found to be consistent with experimental Mössbauer studies. Simulations of the EQS values for singlet  $[FeH(\eta^2-H_2)dppe_2]^+$  and the nitroprusside anion were also performed, and found to have similar deviations from their experimental values, within the expected range of error for simulations of Mössbauer properties.<sup>154</sup> These results provide strong evidence for predominance of the triplet-state configuration of  $[FeHdppe_2]^+$  at room temperature, and for the direct conversion from the triplet five-coordinate species to the singlet hydrogen adduct.

The simulations used to calculate EQS values allow for determination of the sign of the principal component of the electric field gradient, and correspondingly the sign of the EQS. These signs are presented with their values in Table 4.1. With identification of the sign, the method of partial quadrupole splittings presented by Bancroft,<sup>157</sup> can be used to determine the individual contribution of the H<sub>2</sub> ligand to the EQS from the experimental value. The expression for the EQS based on partial quadrupole splittings (Q) is

$$EQS = \sum_{i} Q(L_i)(3\cos^2\theta_i - 1)$$

$$\tag{4.3}$$

where  $Q(L_i)$  is the partial quadrupole splitting of ligand i, and  $\theta_i$  is the angle the ligand-metal bond makes with the principal component axis of the electric field gradient. Assuming perfect octahedral geometry and the principle axis aligned along the H-Fe-H<sub>2</sub> axis, the expression for the EQS becomes

$$EQS = 2 Q(H^{-}) + 2 Q(H_{2}) - 4 Q(dppe/2)$$
(4.4)

Using the values of  $Q(H^-) = -1.04 \text{ mm/s}$  and Q(dppe/2) = -0.50 mm/s, this expression leads to a value of  $Q(H_2) = -0.245 \text{ mm/s}$ , which correlates well with the value of -0.22 mm/s presented by Morris and Schlaf for  $[FeH(\eta^2-H_2)dppe_2][BF_4]$ .<sup>119</sup> The low Q value of dihydrogen as a ligand, compared to the large value of the hydride ligand, should be universally obtainable for Fe<sup>mathrmII</sup> complexes because of the additive nature of the partial quadrupole splitting model. This distinct feature of dihydrogen as a ligand may be useful in identifying dihydrogen complexes and Kubas binding in Mössbauer active materials, as well as defining the onset of dihydrogen-dihydride transitions.

#### 4.2.2.2 Isotherm Simulations

The importance of the magnetic transition in the iron complexes can be inferred from calculations of the singlet-state species, and, based on general trends in the periodic table, the analogous ruthenium and osmium complexes are expected only to exist in the singlet states. In this context, we have performed calculations to simulate the Langmuir absorption behavior onto the singlet-state complexes [MHdppe<sub>2</sub>]<sup>+</sup> (M = Fe, Ru, Os), similar to the calculations performed for the group 7 complexes in the previous chapter.

An additional complication to the simulations not present in the group 7 complexes arises from the hydride ligand *trans* to the coordination site. Bautista et al.<sup>96</sup> found through NMR experiments that the dihydrogen and hydride ligands can rapidly exchange positions at room temperature in the iron and osmium complexes through the formation of the transient trihydride  $[MH_3dppe_2]^+$ . In the solid state, where the orientation of the molecule with respect to the rest of the solid is fixed, this intramolecular conversion produces "up" and "down" bound states that introduce additional entropy, and an adjustment of the Langmuir model is necessary to account for this effect. Because the unbound states are unable to alter their configuration, this inversion only affects the entropy of the bound states. The partition function describing this behavior can be formulated by assigning an orientation to the hydrogen molecules, such that "up" or "down" molecules bind to the open site with equivalent binding energies. This convention produces a new system partition function, expressed as:

$$\Xi = \sum_{n_{\uparrow}}^{N} \sum_{n_{\downarrow}}^{N-n_{\uparrow}} \frac{N!}{n_{\uparrow}! n_{\downarrow}! (N - n_{\uparrow} - n_{\downarrow})!} e^{-\beta \Delta E(n_{\downarrow} + n_{\uparrow})} e^{\beta \mu_{\text{gas}}(n_{\downarrow} + n_{\uparrow})}$$
(4.5)

where the subscript arrows denote the orientation of the binding molecule, with the total number of bound dihydrogen molecules  $n = n_{\downarrow} + n_{\uparrow}$ . From this partition function, a new Langmuir isotherm was derived, giving:

$$\frac{\langle n \rangle}{N} = \frac{2q_{\rm s} \mathrm{e}^{\beta(\mu_{\rm gas} - \Delta E)}}{1 + 2q_{\rm s} \mathrm{e}^{\beta(\mu_{\rm gas} - \Delta E)}} \tag{4.6}$$

Equation 4.6 was used for iron and osmium simulations. In the same study, Bautista et al. found that inversion did not occur significantly in the analogous ruthenium complex, and 3.6 was used for ruthenium simulations. Analysis of the rotational mode eigenvector components indicated that the rotation of the H<sub>2</sub> molecule about its center of mass was asymmetric for each complex, and a value of  $q_s=1$  was used for all simulations.

As with the group 7 complexes, the vibrational analysis revealed that occupied normal modes of the hydrogen molecule in the group 8 H<sub>2</sub> adducts were dominated by the rotational modes, with frequencies of 385.7, 385.5, and 202.1 cm<sup>-1</sup> for Fe, Ru, and Os species, respectively. The rotational mode at 385.7 cm<sup>-1</sup> for the iron complex was previously assigned by Eckert et al.<sup>110</sup> as a translation mode based on the presence of additional peaks at 225 and 255 cm<sup>-1</sup> in the vibration spectra from inelastic neutron scattering experiments. These lower energy peaks were also reproduced by the GAMESS simulations at 237.6 and 254.3 cm<sup>-1</sup>, and were found to be associated with strong translation modes of the iron center towards the hydride which couple to both the hydride and the dihydrogen atoms. The broad peak at 400 cm<sup>-1</sup> in the neutron scattering experiment thus likely corresponds to the primary rotational mode of the dihydrogen molecule.

The standard enthalpy and entropy obtained from the  $K_{eq}$  values of the simulated Langmuir isotherms appear in Table 4.2. As with the group 6 complexes reported by Gonzalez and Hoff<sup>123</sup> and the group 7 complexes reported in the previous chapter, the binding enthalpy in the singletstate group 8 complexes increases as 5d > 3d > 4d. Interestingly, the difference between the

Table 4.2: Calculated energies, enthalpies, and entropies of the  $H_2$  absorption reaction at 1 atm, 298K and rotational entropy of the bound states at 298K and 373K. Energy and enthalpy values are given in kJ/mol, and entropy values are given in J/mol-K.

Compound	$\Delta E^{\circ}$	$\Delta H^{\circ}$	$\Delta S^{\circ}$	$S_{rot}(298K)$	$S_{rot}(373K)$
$[FeH(\eta^2-H_2)dppe_2]^+$	-41.2	-48.0	-91.4	4.19	5.74
$[RuH(\eta^2-H_2)dppe_2]^+$	-33.9	-41.4	-98.5	4.19	5.74
$[OsH(\eta^2-H_2)dppe_2]^+$	-76.3	-80.6	-82.9	8.76	10.59

enthalpies and entropies of the osmium species and the other group 8 complexes is much higher than the corresponding differences in the group 6 and 7 complexes. This result suggests stronger backdonation from the osmium center and stabilization of the trihydride species, which is consistent with reported NMR studies in solution that indicate the trihydride is present.<sup>96</sup>

Simulated thermodynamic values and isotherms for the osmium and ruthenium complexes correlate well with experimental observations.<sup>96</sup>  $[OsH(\eta^2-H_2)dppe_2]^+$  was found to be extremely stable to loss of the dihydrogen ligand at all temperatures, with greater than 97% coverage down to 0.25 torr at 373K. This behavior is consistent with observations of exchange kinetics in the complex, which indicate that  $[OsH(\eta^2-H_2)dppe_2][BF_4]$  is highly stable to exchange of the H<sub>2</sub> ligand with N<sub>2</sub>, D<sub>2</sub>, or CH<sub>3</sub>CN in the solid state or in solution. Simulated isotherms for absorption over the ruthenium complex appear in Figure 4.4. As shown in the figure, the ruthenium complex has only ~80% coverage at room temperature and significant overpressure is required to maintain the dihydrogen complex, consistent with experimental reports. At higher temperatures the stability of the dihydrogen complex is reduced, with only ~40% coverage at 373K and 5 bar. This behavior is ideal for hydrogen storage applications, which require cycles with materials showing room temperature stability but strong release of the absorbed gas upon mild heating.

The absorption enthalpy for the singlet iron fragment  $[FeHdppe_2]^+$  is lower than that of the manganese fragment  $[Mn(CO)dppe_2]^+$  reported in the previous chapter, which significantly underestimates the binding strength in the iron complex relative to our experimental observations and reported behavior in the solid state. This deviation demonstrates the importance of the magnetic spin state to the binding behavior in these materials, despite previous reports of the nearly isoenergetic nature of the singlet and triplet states for the iron complex. Such spin state transitions



Figure 4.4: Simulated isotherms for hydrogen absorption over [RuHdppe<sub>2</sub>]<sup>+</sup> up to 5 bar pressure

may be responsible for other complications known to exist selectively in iron dihydrogen complexes. For instance, Bianchini et al.<sup>158,159</sup> found that during the catalysis of alkyne hydrogenation, the tetraphosphine complex  $[(PP_3)FeH(H_2)][BPh_4]$  preferentially removes a phosphine arm before releasing the dihydrogen ligand, while the ruthenium analog  $[(PP_3)RuH(H_2)][BPh_4]$  first releases the dihydrogen molecule for the same catalytic process. If present in the tetraphosphine complexes, the loss of hydrogen to form a high-energy triplet state may be less favorable than the removal of a phosphine arm to form a singlet state, and this effect would not be present in the analogous ruthenium complex. The potential for such magnetic transitions should be assessed carefully when evaluating these materials for both chemical and hydrogen storage applications.

#### 4.2.3 Conclusions

Mössbauer parameters for the thermal cycling products of  $[FeH(\eta^2-H_2)dppe_2][NTf_2]$  have been determined experimentally and through electronic structure simulations using the GAMESS software. From these parameters, the triplet spin state of the dehydrogenated product [FeHdppe<sub>2</sub>][NTf<sub>2</sub>] was found to be the predominant dehydrogenation product at room temperature, with the absorption reaction accompanied by a triplet $\rightarrow$ singlet magnetic transition that greatly affects the thermodynamic stability of the hydrogen adduct. A partial quadrupole splitting value of -0.245 mm/s was identified for the dihydrogen ligand in these complexes, which differs significantly from the value of -1.04 mm/s for hydride ligands and may be used to help distinguish Kubas binding from hydride formation in hydrogen storage materials.

A modification of the Langmuir isotherm model used to describe the behavior of group 7 dihydrogen complexes was applied to absorption of hydrogen onto the singlet-state complexes [MHdppe<sub>2</sub>]<sup>+</sup> (M = Fe, Ru, Os). This model accurately described the reported behavior for the ruthenium and osmium complexes and illustrated the magnitude of the spin-state transition on the absorption behavior in the iron complex, which showed singlet-state enthalpy values similar to the manganese complex reported in the previous chapter despite experimental evidence for much stronger binding. Singlet-state absorption enthalpies demonstrated the familiar trend of increase as 5d > 3d > 4d, indicating a consistent trend within transition metals exhibiting the same spin state. However, the magnitude of the differences between the metal centers were significantly different, particularly in osmium complexes. Simulated isotherms for the ruthenium complex was found to have near ideal behavior for hydrogen absorption applications, showing great promise for the development of future organometallic hydrogen storage.

# Chapter 5 Conclusions

#### 5.1 Summary of the Current Work

In the preceding chapters, new methods and techniques based on the Sieverts technique used for the evaluation of hydrogen storage materials have been successfully employed to evaluate the thermodynamic properties of hydrogen binding in dihydrogen complexes. These *enabling technologies* provide methods for evaluation of the thermodynamic properties of these hydrogenation reactions that have, to date, proven elusive by traditional chemical methods because of the stability of known dihydrogen complexes and the instability of the dehydrogenated products. These methods may prove invaluable not only to the assessment and design of future hydrogen storage materials, but also to organometallic chemistry fields for the study of complexes that are sensitive to solvent addition or are only moderately stable at room temperature and require higher pressures to form.

Mössbauer spectroscopy has also been confirmed as a viable technique for distinguishing between dihydrogen and dihydride complexes, and for determining the identity of dehydrogenation products when solvent sensitivity is an issue.<sup>107</sup> This underutilized spectroscopic technique may prove useful in future chemical evaluations of dihydrogen complexes, given that the majority of known dihydrogen complexes have iron, ruthenium, osmium, or rhenium metal centers,<sup>88,90</sup> all of which are Mössbauer active atoms.

The Langmuir isotherm and computational studies using B3LYP/LANL2DZ-ECP DFT theory for geometry optimization and vibrational analysis of complexes, combined with MP2-SCS/LANL2DZ- ECP theory calculations for the total molecular energy, have been shown to be effective tools for producing highly accurate estimates of the equilibrium absorption behavior for dihydrogen complexes. Previous efforts using only ideal gas behavior van't Hoff equations have found difficulty in reproducing such behaviors, due to the difficulty in estimating entropic contributions to the thermal behavior.<sup>16,17,20</sup> The Langmuir isotherm, both in classical and statistical formulations, provides methods to avoid these difficulties and directly determine the thermodynamic properties from the equilibrium constant, providing more accurate determination of the enthalpy and entropy.

Enthalpy values obtained for the complexes in these studies were shown to be surprisingly high relative to estimates in the hydrogen storage literature,<sup>5</sup> and in all cases were comparable to tightly bound metal and complex hydride materials. However, significantly reduced entropy from the values expected<sup>16,17</sup> for formation of hydrides allows near room temperature cycling for these complexes. This unique behavior, combined with rapid kinetics observed in many dihydrogen complexes, holds significant promise for dihydrogen complexes and other Kubas materials for hydrogen storage applications.

The source of the reduced absorption entropy of  $\sim 30$  J/mol-K for all of the studied complexes relative to expectation has not been fully established. The rotational mode of the dihydrogen molecule in the complex was the only mode found to be significantly populated in dihydrogen complexes that would be absent in the corresponding hydride complexes, and accounted for up to  $\sim 11$  J/mol-K of the entropy change. The remaining contributions to the entropy change remain unknown but may be electronic in nature, associated with the spread of electronic charge throughout the large ligand framework. Investigations of the additional sources of entropy provide grounds for future study.

#### 5.2 Considerations for Future Studies

P-T-x behavior relevant for hydrogen storage has been found for hydrogenation of  $[Mn(CO)dppe_2]^+$ and  $[RuHdppe_2]^+$  cationic complexes. Although these complexes were only originally intended as models for evaluation of the Kubas mechanisms because of their weight and bulk, the surprising discovery of extremely rapid absorption kinetics in these materials provides renewed enthusiasm for the use of dihydrogen complexes as hydrogen storage materials.

In the following sections, some considerations are presented for the further exploration of dihydrogen complexes for hydrogen storage applications.

#### 5.2.1 Known Dihydrogen Complexes for Hydrogen Storage

The majority of materials exhibiting Kubas binding are high surface area or porous materials, such as MOFs or decorated carbons, designed to incorporate unsaturated metal centers as well as a high surface area for physisorption. However, as discussed previously, while such schemes have been shown to increase binding enthalpies in physisorption materials, it is unclear whether this is due to a greater binding potential throughout the entire material or whether the metal centers act independently of the physisorption sites, binding hydrogen with a higher enthalpy at the metal while weak binding interactions still predominate at the majority of the sites. If the latter, these schemes may not produce the desired binding properties until metal centers account for significant weight in the complex, which defeats the intention of the technique.

The bulk absorption properties found for hydrogen in  $[Mn(CO)dppe_2]^+$  have the advantage of producing equivalent binding sites for each hydrogen absorbed, leading to uniform and easily characterized absorption and desorption behavior. Development of higher weight dihydrogen complexes for hydrogen storage is therefore a viable method to provide additional control of operating properties. However, dihydrogen complexes containing significant percentages of removable hydrogen are rare.<sup>88</sup> Most complexes contain less than 1 wt % hydrogen; the complex  $CpMn(H_2)(CO)_2$ , a half-sandwich complex, has the highest reportable weight percent of dihydrogen for isolable dihydrogen complexes at 1.1 wt%.

Reports of dihydrogen complexes as potential hydrogen storage materials are further complicated by poor communication of fuel system requirements. Many complexes, such as  $Cr(CO)_5(H_2)$ , are reported as potential hydrogen storage materials but require extreme low temperatures to form and are unlikely to have stable unsaturated dehydrogenation products, which can cause potential release of carbon monoxide in fuel systems. Others, such as  $[Ru(H_2O)_5(H_2)]^+$  and  $[Os(NH_3)_5(H_2)]^+$ , exist only as solvated ions, which require a large portion of the weight of the storage material to be solvent. Additionally, complexes such as  $Ru(H)_2(H_2)_2(PCp_3)_2$  are reported for potential hydrogen storage applications. Up to 10 H<sub>2</sub> molecules may be reversibly extracted from this complex for a total of 1.7 wt% hydrogen release, but only through a substitution reaction with ethylene.<sup>161</sup> Only two H<sub>2</sub> molecules are labile through thermal decomposition, which reduces the usable percentage of hydrogen to 0.69 wt%.<sup>162</sup>

Significant material discovery efforts will be needed if dihydrogen complexes are to meet the USDOE gravimetric capacity requirements. However, it should be noted that these requirements are specific for vehicular and mobile systems. Stationary applications, such as fuel storage systems at refilling stations or electrical generation plants, may have more relaxed requirements for gravimetric capacity and could make use of currently known dihydrogen complexes. The relevant specification in these applications is the compression capability, designated by the volumetric storage capacity. This metric is poorly reported for dihydrogen complexes.

#### 5.2.2 Computational Discovery of New Complexes

Unfortunately, dihydrogen complexes suitable for hydrogen storage applications are unlikely to be discovered in the chemical literature. The requirement of 3–100 bar hydrogen delivery pressure set by the USDOE<sup>4</sup> for working fuel systems requires dihydrogen complexes that are *unstable* under ambient pressure and near room temperature conditions. Performance of the spectroscopic measurements necessary to identify dihydrogen complexes under pressurized conditions is extremely difficult, and limited to the work of a few specialized research groups. Few dihydrogen complexes have been synthesized under greater than one atmosphere of hydrogen pressure, although some, such as  $[RuH(H_2)dppe_2]^+$  have reported instabilities at room temperature<sup>96</sup> unless a hydrogen atmosphere is maintained. Unstable complexes may be formed at lower pressures and detected by solution NMR if lower temperatures are used; however, lower temperatures increase competition for the binding site by solvent, reducing the likelihood of observation of pure species and subsequent reporting. Additionally, such complexes may not be discovered simply through reluctance to explore the lower temperature parameter space; complexes that are unreactive to hydrogen at room temperature are less likely to be tested at lower temperatures because of generally slower kinetics and difficulties in experimental procedure.

The computational techniques presented in this paper not only provide methods for evaluating equilibrium behavior in known complexes, but also an opportunity for discovery and screening of new materials for hydrogen storage applications. Lightweight complexes that otherwise have not been observed to absorb hydrogen at room temperature but are otherwise of interest for their hydride character, such as Ni(dmpe)<sub>2</sub>, can be evaluated for direct hydrogenation and use as hydrogen storage opportunities. Trends have been developed for the relationship between the identity of the *trans* ligand and the H-H bond distance;<sup>168</sup> additional studies using the Langmuir isotherm can now be included to evaluate the full reaction coordinate. Complexes of the M(dmpe)<sub>2</sub> structure are known for 3d transition metals from Cr to Ni,<sup>164–167</sup> and appropriate calculations can provide trends across the periodic table. Extension of such studies can also provide additional chemical trends to guide materials design.

#### 5.2.3 Engineering Considerations

One of the promising potential benefits for dihydrogen complexes as hydrogen storage materials is the availability of the raw materials. Complexes in this study with relevant hydrogen storage properties were composed of earth-abundant 3d transition metals and organic molecules. This provides a significant advantage for these materials in energy applications over lithium-based or rare-earth derived storage materials, which are available in recoverable quantities in only a few locations and could face significant shortages if relied upon for energy infrastructure replacement. At the moment, phosphine complexes such as dppe are produced in small quantities and are quite expensive. However, organophosphorus compounds are produced in large quantities for pesticide and fire retardant applications, and the potential for large scale production is available for countries with developed chemical industries. Because of the high enthalpies and rapid kinetics observed for the complexes in this study and on the basis of similar enthalpy values found in previous studies, heat management is expected to be a critical component for operation of a dihydrogen complex hydrogen storage system. As mostly organic complexes and salts, dihydrogen complexes are expected to be thermal insulators and poor conductors of heat in the solid state, which can lead to rapid localized heating upon absorption, producing decomposition and potentially damaging fuel systems. Future studies for use of dihydrogen complexes as hydrogen storage materials should include evaluation of heat transfer properties in these materials and identification of proper heat management technologies.

#### 5.3 Concluding Remarks

The use of hydrogen for fuel applications has been a prominent goal of energy technology research for its potential for renewable production and clean combustion. Fuel storage is one of several critical components to the development of hydrogen energy infrastructure and, while research efforts have produced drastic improvements in the performance of storage materials, significant materials discovery and engineering development are still required before hydrogen can become a consistent part of the US energy profile.

The efforts presented in this work have been directed at developing methods and techniques for evaluating new storage materials. Application of these techniques, and further evaluation of dihydrogen complexes as storage materials, can lead not only to new materials but to the capability to rationally design dihydrogen complexes for specific hydrogen applications and rapidly improve energy storage systems toward inclusion in the national energy infrastructure.

## **Appendix: GAMESS Input Files**

In this section, the GAMESS input files used to calculate thermodynamic properties and Mössbauer parameters for the 3d organometallic clusters and the  $H_2$  molecule are reprinted. Files for the 3d transition metal organometallic fragments are representative of files for 4d and 5d fragments. The effective core potentials and LANL2DZ basis set are those defined for the "LANL2DZ ECP" basis set in the EMSL Basis Set Exchange database mhosted by Pacific Northwest National Laboratory<sup>169, 170</sup>

Because of their length, automatically generated molecular eigenvector printouts have been excluded from some of the files below. These values are generated after a completed run, and are used for improving performance on subsequent runs or restarts.

### 5.3.1 Geometry Optimization

```
! H2 molecule geometry optimization and vibrational analysis
 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN DFTTYP=B3LYP
    MAXIT=30 $END
 $SYSTEM MEMDDI=0 MWORDS=50 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=1.0e-7 $END
 $STATPT OPTTOL=0.0000001 TRMIN=0.01 NSTEP=100 HSSEND=.T. $END
 SFORCE METHOD=SEMINUM NVIB=2 VIBSIZ=0.01 SCLFAC=0.96 VIBANL=.TRUE.
   TEMP(1)=295 TEMP(2)=298 TEMP(3)=300 TEMP(4)=313 TEMP(5)=333
   \text{TEMP}(6) = 343 \text{ TEMP}(7) = 353 \text{ TEMP}(8) = 363 \text{ TEMP}(9) = 373
 $END
 $DATA
H2 molecule
C1
Η
      1.0
            -0.02189 0.00000 0.00000
Н
      1.0
            0.72189
                         0.00000
                                      0.00000
 $END
```

Input file for GAMESS - [Mn(CO)dppe2]+ geom. opt. ! \$CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN DFTTYP=B3LYP MAXIT=100 ICHARG=1 ECP=READ \$END \$SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. \$END \$BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 NPFUNC=1 \$END \$GUESS GUESS=MOREAD NORB=204 \$END \$SCF DIRSCF=.TRUE. CONV=5.0e-6 \$END \$STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 \$END \$DATA Mn(CO)dppe2 C1 25.0 7.67008 8.28046 -0.21132 Mn Ρ 15.0 6.82809 10.48811 0.40010 Ρ 15.0 10.06459 8.71101 0.34001 С 6.0 7.79196 12.04727 -0.03090 С 6.0 8.41215 12.82113 0.97456 С 6.0 9.12716 13.98499 0.63619 С 6.0 9.23721 14.38322 -0.70971 С 13.60362 6.0 8.63746 -1.71663 С 6.0 12.43929 7.92188 -1.38223С 6.0 6.10959 10.87473 2.10781 С 6.0 5.49090 12.12274 2.35226 С 6.0 4.91432 12.40192 3.60388 С 6.0 4.94926 11.43686 4.63110 С 6.0 5.55991 10.19105 4.39570 С 6.0 9.91252 6.13327 3.13868 С 6.0 11.07368 9.82148 -0.80511 С 6.0 -0.36890 11.55828 11.07234 С 6.0 12.30455 11.88731 -1.24034 С 6.0 12.57049 11.46497 -2.55567 С 10.22453 -3.00068 6.0 12.07386 С -2.13349 6.0 11.32339 9.41041 С 6.0 10.63577 9.24256 2.06158 С 6.0 12.00782 9.22436 2.40702 С 6.0 12.42099 9.58568 3.70144 С 6.0 11.47033 9.97577 4.66753 С 6.0 10.10641 10.00906 4.32674 С 6.0 9.69565 9.64348 3.02909 С 6.0 5.25128 10.64162 -0.66231 С 6.0 10.96897 7.03049 0.17398 1.0 Η 2.02002 8.33046 12.53592 14.57841 Η 1.0 9.58941 1.42145 -0.97007 Η 1.0 9.78134 15.28786 Η 1.0 8.72562 13.89542 -2.76002Η 1.0 7.48231 11.85019 -2.18217Η 1.0 5.46694 12.88327 1.57525 Η 1.0 4.44339 13.36630 3.77712 1.0 4.50770 11.65571 5.60011 Η Η 1.0 5.59048 9.44016 5.18185 1.0 6.59693 8.94168 2.97505 Η Η 1.0 11.36802 11.41498 0.64379 1.0 12.84704 Η 12.67562 -0.88906 12.09304 -3.22578 Η 1.0 13.15238 Η 1.0 12.26465 9.89324 -4.01840 Η 1.0 10.93921 8.46489 -2.507748.94998 Η 1.0 12.75969 1.67027 1.0 13.47787 9.56669 3.95459 Η Η 1.0 11.79230 10.25368 5.66818 Η 1.0 9.36538 10.31820 5.05981 Η 1.0 8.64286 9.68103 2.77689

Η	1.0	4.75163	11.59040	-0.43599
Н	1.0	5.55059	10.65836	-1.71560
Η	1.0	11.29153	6.91198	-0.86582
Н	1.0	11.86538	7.04421	0.80134
Ρ	15.0	8.47111	5.95830	-0.51579
Ρ	15.0	5.26265	7.78020	-0.52788
0	8.0	7.93872	9.30652	-2.96195
С	6.0	7.48214	4.46650	0.08401
С	6.0	6.82874	3.61342	-0.83270
С	6.0	6.11130	2.49310	-0.37448
C	6.0	6.02815	2.21806	1.00387
C	6 0	6 65748	3 07828	1 92346
C	6.0	7 37805	4 19713	1 46654
C	6.0	9 22559	5 32067	-2 13502
C	6.0	0 20326	1 02010	_2.15502
C	6.0	10 42220	3 55228	_2 22502
d	6.0	10.42320	J.JJZJO 4 JE070	-3.33593
d	6.0	10.47239	4.30070	-4.49152
d	6.0	9.69/54	5.04150	-4.40055
C	6.0	9.28041	6.12254	-3.29409
C	6.0	4.31683	6.68925	0.69969
C	6.0	3.84956	5.41090	0.32779
С	6.0	3.15366	4.61027	1.25248
С	6.0	2.92345	5.06957	2.56232
С	6.0	3.39811	6.33933	2.94378
С	6.0	4.09141	7.14233	2.01937
С	6.0	4.64766	7.14540	-2.19622
С	6.0	3.26451	7.15139	-2.49435
С	6.0	2.80264	6.69775	-3.74213
С	6.0	3.71690	6.22734	-4.70719
С	6.0	5.09182	6.20873	-4.41204
С	6.0	5.55321	6.66677	-3.16214
С	6.0	7.85001	8.86482	-1.85190
С	6.0	10.03155	5.88650	0.58752
С	6.0	4.32384	9.45135	-0.36984
Н	1.0	6.89343	3.80041	-1.90135
Н	1.0	5.62634	1.83620	-1.09275
Н	1.0	5.48293	1.34601	1.35628
н	1.0	6.59121	2.88014	2.99033
н	1.0	7.86006	4.84199	2.19845
н	1 0	9 75984	3 38513	-1 29242
н	1 0	10 86157	2 55759	-3 34557
н	1 0	10 95080	3 98985	-5 39535
и и	1 0	9 92570	6 27187	-5 35204
и и	1 0	8 85336	7 11771	-3 30762
п u	1.0	4 01117	F 02670	-3.30702
п u	1.0	7.01117 2.70206	2 62000	-0.07837
п 	1.0	2.79300	3.03090	0.94032
п 	1.0	2.3/992	4.45134	3.2/250
H TT	1.0	3.22392	6.70756	3.95225
H	1.0	4.43227	8.12299	2.33/09
Н	1.0	2.54094	7.48804	-1.75511
Н	1.0	1.73741	6.70845	-3.95889
Н	1.0	3.35947	5.88077	-5.67375
Н	1.0	5.80487	5.84664	-5.14857
Η	1.0	6.61586	6.65060	-2.95164
Η	1.0	10.51292	4.91053	0.46239
Η	1.0	9.73468	5.98446	1.63887
Η	1.0	3.91885	9.51865	0.64537
Η	1.0	3.47971	9.44409	-1.06554
\$E]	ND			

\$ECP

MN-ECP GEN	10 2	
3 d	potential	
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-60.2990287	2	64.6477389
-10.4217834	2	16.0960833
5 s-	-d potentia	1
3.000000	0	107.4127215
16.2591819	1	111.4958973
276.9373928	2	46.5568346
241.3174342	2	8.3688135
-146.4635329	2	7.7237489
5 p-	-d potentia	1
5.000000	0	80.0415103
5.7589756	1	105.6043646
285.2918654	2	40.8300466
143.4222647	2	8.0098457
-88.7031851	2	7.3390928
P-ECP GEN	LO 2	
5 d	potentia	1
-10.000000	1	462.1211423
-79.4864658	2	93.6863701
-28.3668251	2	21.2349094
-9.8577589	2	6.3388415
-1.0163783	2	2.0620684
5 s-	-d potentia	1
3.000000	0	78.0831823
12.9104154	1	58,9576810
150.0250298	2	36.0571255
71 7083146	2	11 2464453
23 0397012	2	2 6757561
б р-	-d potentia	1
5 000000	0	- 75 1617880
6 3446507	1	57 4544041
198 5585104	2	47 9481748
111 1470820	2	18 4588360
40 3944144	2	5 9414190
6 4483233	2	1 8487507
P-ECP	2	1.010/00/
C-ECP NONE		
C-ECP NONF		
C-ECP NONE		
C-ECP NONE		
C 101 1010E		

C-ECP	NONE
C-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H_FCD	NONE
H_FCD	NONE
H_FCD	NONE
	NONE
H FCP	NONE
H-ECP	NONE
P-ECP	
P-ECP	
O-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
-	

H-ECP	NONE
H-ECP	NONE
\$END	

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Input file for GAMESS - [Mn(CO)dppe2]+ agostic config geom opt
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Ρ
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Ρ
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Ρ
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\$E	ECP					
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H-ECP NONE		

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P-ECP	
P-ECP	
0-ECP	NONE
C-ECP	NONE
H-ECP	NONE

H-ECP	NONE
H-ECP	NONE
\$END	

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Input file for GAMESS - [FeHdppe2]+ geom. opt.
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 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN DFTTYP=B3LYP
    MAXIT=100 ICHARG=1 ECP=READ $END
 $SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=MOREAD NORB=198 $END
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 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
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с н	1 0	3 18088	11 08832	-3 22363
н Н	1 0	10 99926	5 71666	-3 45200
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C	6.0	9 17197	5 82051	
с u	1 0	J. <del>7</del> 6140	11 27/60	
п	1.0	4.70142	E 62096	0.77795
п	1.0	9.09015	5.02900	
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! Input file for GAMESS - [FeH(H2)dppe2]+ geom. opt.
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 $SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. $END
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 $GUESS GUESS=MOREAD NORB=199 $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 $END
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Н	1.0	9.32233	11.25651	2.55171
н	1.0	4,95592	5.34508	2.53983
C	6 0	9 19300	11 33734	-1 28976
C	6.0	5 13007	5 22590	-1 27751
с н	1 0	8 39288	12 04548	-1 49485
и и	1 0	5 94237	4 50711	-1 36072
C II	±.0	11 16/2/	9 38006	5 72654
d	6.0	2 07722	9.30900	5.72054
U TT	1 0	10 20020	0 21070	5.70255 E 42201
H	1.0	12.20830	9.31079	5.43301
н	1.0	2.02010	1.23920	5.48607
Ċ	6.0	3.94232	10.89194	-2.5/89/
C	6.0	10.35336	5.89933	-2.53493
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Н	1.0	11.16228	5.72307	-3.23952
С	6.0	4.80279	11.02968	-0.29954
С	6.0	9.48743	5.85391	-0.25381
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С	6.0	9.11825	6.40142	-2.98542
Н	1.0	5.31138	10.06452	-4.04078
н	1.0	8.97077	6.61807	-4.04077
C	 6 0	3 77554	11 33934	5 17694
C	6.0	10 57467	5 50355	4 82347
с н	1 0	2 73887	11 03557	5 05366
и и	1 0	11 58759	5 79747	4 55857
C	±.0	10 04258	9 36/99	-0 12722
C	6.0	1 22106	9.30499 7.21554	0.12151
с u	1 0	9.22100	0 E1120	0.43131
п	1.0	9.09104 4 21074	0.04420	0.55574
п		4.31974	0.23034	0.13307
C a	6.0	0.14907	LU.90575	4.70020
C C	6.0	8.10252	5.82371	4.60111
C	6.0	11.28207	9.52300	-0./8/98
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С	6.0	3.93495	4.98671	-1.97765
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Н	1.0	3.83385	4.08610	-2.57829
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 -207.3421649 2
                         8.6289082
     ----- p-d potential -----
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H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-FCD	NONE
H_FCD	NONE
C-FCD	NONE
	NONE
	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-FCD	NONE
H-FCD	NONE
	NONE
C FCP	NONE
	NONE
	NONE
H-ECP	NONE
н-ЕСР	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE

C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
\$END	

```
!
    Input file for GAMESS - [Fe(NO)(CN)5]2- geom opt
 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN
    MAXIT=100 ICHARG=-2 MULT=1 $END
 $SYSTEM MEMDDI=0 MWORDS=150 PARALL=.TRUE. $END
 $BASIS GBASIS=TZV NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $ELDENS IEDEN=1 WHERE=NUCLEI $END
 $ELFLDG IEFLD=2 WHERE=NUCLEI $END
 $ELPOT IEPOT=1 WHERE=NUCLEI $END
 $DATA
Nitroprusside ion
C1
Fe
      26.0
               1.00328
                          -0.01316
                                        0.07755
С
      6.0
              1.00322
                          0.73734
                                       1.90348
С
      6.0
              1.00337
                          1.85061
                                      -0.54892
С
      6.0
             -0.96674
                          0.10855
                                       0.03647
С
                                      -1.83045
      6.0
              1.00333
                         -0.51946
Ν
      7.0
              1.00323
                         -1.56123
                                       0.59897
С
      6.0
              2.97330
                          0.10841
                                       0.03665
0
      8.0
              1.00321
                         -2.64721
                                       0.96494
Ν
             -2.12905
                          0.09704
      7.0
                                       0.04043
Ν
      7.0
              1.00328
                         -0.90045
                                      -2.92858
Ν
      7.0
              4.13559
                          0.09721
                                       0.04030
Ν
      7.0
              1.00342
                          1.09715
                                       3.00873
Ν
      7.0
              1.00333
                          2.95140
                                      -0.91868
```

\$END

```
L
    Input file for GAMESS - [FeHdppe2]+ sing geom opt TZV
 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN MAXIT=200
    ICHARG=1 QMTTOL=1e-5 $END
 $SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. $END
 $BASIS GBASIS=TZV NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=MOREAD NORB=223 $END
 $SCF DIRSCF=.TRUE. DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $ELDENS IEDEN=1 WHERE=NUCLEI $END
 $ELFLDG IEFLD=2 WHERE=NUCLEI $END
 $ELPOT IEPOT=1 WHERE=NUCLEI $END
 $DATA
[FeHdppe2]+
C1
      26.0
Fe
               7.40615
                            8.30950
                                        2.15412
Ρ
     15.0
              7.70477
                          10.24504
                                       3.85524
Ρ
     15.0
              6.72702
                           6.36014
                                       3.83679
Ρ
     15.0
              7.55588
                          10.36726
                                       0.47140
Ρ
     15.0
              6.91908
                          6.40631
                                       0.45073
С
      6.0
                                       1.57944
              7.60132
                          11.86121
              7.03690
С
      6.0
                           4.87626
                                       1.49794
Н
                                       1.03319
      1.0
              7.98820
                          12.71078
Н
      1.0
              6.77270
                           4.00455
                                       0.91459
Н
      1.0
              6.59496
                          12.11035
                                       1.88663
Н
      1.0
              8.06715
                           4.75491
                                       1.80538
С
      6.0
              8.90466
                          10.00115
                                       5.22835
С
      6.0
              5.37402
                           6.59697
                                       5.06958
С
      6.0
             10.25380
                           9.82132
                                       4.93594
С
              4.03616
                           6.54880
                                       4.69142
      6.0
Н
      1.0
             10.59756
                           9.83558
                                       3.91990
Н
      1.0
              3.75619
                           6.30483
                                       3.68478
С
      6.0
              6.19191
                          10.71089
                                      -0.72434
С
      6.0
              8.14772
                           6.13254
                                      -0.89114
С
      6.0
                          10.55553
                                      -0.50527
              9.10112
С
              5.26879
                           6.21996
                                      -0.34075
      6.0
С
      6.0
              8.49243
                           9.93349
                                       6.55300
С
      6.0
              5.68012
                           6.92941
                                       6.38525
Н
      1.0
              7.45870
                          10.06201
                                       6.80679
Н
      1.0
              6.70255
                           6.97566
                                       6.70605
С
      6.0
              6.30346
                          10.30679
                                      -2.05063
С
      6.0
              7.83931
                           6.38446
                                      -2.22181
Н
      1.0
              7.21275
                           9.86116
                                      -2.40411
Н
      1.0
              6.85468
                                      -2.49912
                           6.70590
С
      6.0
                                       5.78367
              5.26481
                          12.95132
С
              8.84829
                                       6.22962
      6.0
                           3.78111
Н
      1.0
              5.37295
                          13.88826
                                       6.29628
Н
      1.0
              8.63900
                           2.92417
                                       6.84122
              6.37739
С
      6.0
                          12.31636
                                       5.27246
              7.82186
С
      6.0
                           4.42059
                                       5.56893
Н
      1.0
              7.34182
                          12.76856
                                       5.40532
Н
      1.0
              6.81909
                           4.05439
                                       5.68443
```

С	6.0	3.94669	11.46242	-1.18910
С	6.0	10.38758	5.54647	-1.55642
Н	1.0	3.03653	11.91868	-0.84824
Н	1.0	11.37525	5.21937	-1.29166
С	6.0	10.74995	9.54928	7.26584
С	6.0	3.34885	7.12698	6.91537
Н	1.0	11.46181	9.38136	8.05145
Н	1.0	2.57025	7.32101	7.62827
С	6.0	4.99640	10.53465	4.47375
С	6.0	9.38071	5,99656	4.65506
Н	1.0	4.87621	9.58665	3.98511
Н	1.0	9.59249	6.84665	4.03897
С	6.0	8.46777	11.58029	2.81361
Ċ	6.0	6.12633	4.99409	2.72679
H	1.0	8.59713	12.48189	3.39716
Н	1.0	6.10438	4.05960	3.27095
Н	1.0	9.45320	11.24919	2.51331
Н	1.0	5.11290	5.20600	2.41350
C	6.0	9,30853	11.67638	-1.31071
C	6.0	4,90683	5.06616	-1.03586
H	1.0	8,54494	12,42520	-1,40680
н	1.0	5,61279	4.26794	-1,16555
C	6.0	11,16937	9,60615	5.94792
C	6.0	3.03180	6.80723	5.60823
н	1 0	12 20788	9 48004	5 70668
н	10	2 00473	6 74992	5 30063
C	6.0	4.07420	11.06624	-2.50764
C	6.0	10.06762	5.79086	-2.88069
н	1.0	3,26389	11,21171	-3,19637
н	1.0	10.80514	5.65190	-3.64812
C	6.0	4,99490	11.28185	-0.30353
Ċ	6.0	9.43873	5.72436	-0.56801
H	1.0	4.86240	11.59810	0.71323
Н	1.0	9.71750	5.55472	0.45379
C	6.0	9.41028	9.70729	7.56465
Ċ	6.0	4.67613	7.19012	7.29964
Н	1.0	9.07629	9.66150	8.58387
Н	1.0	4.93205	7.43495	8.31309
C	6.0	4.00889	12.37973	5.63731
C	6.0	10.15017	4.24666	6.10362
H	1.0	3.14414	12.87628	6.03490
Н	1.0	10.94954	3.74635	6.61686
C	6.0	5.25436	10.48462	-2.93398
C	6.0	8.79452	6.21535	-3.20959
H	1.0	5.36296	10.17398	-3.95584
Н	1.0	8,53891	6,40931	-4.23408
C	6.0	3,87531	11,17107	4.98566
Ċ	6.0	10,41372	5.35024	5.32025
H	1.0	2,90868	10,71760	4.87813
Н	1.0	11,41828	5.71515	5.22175
C	6.0	10.09937	9.59882	-0.41255

С	6.0	4.33643	7.23881	-0.21454
Н	1.0	9.95801	8.73956	0.21088
Н	1.0	4.59396	8.14740	0.29518
С	6.0	6.25706	11.09795	4.60458
С	6.0	8.07843	5.53708	4.77182
С	6.0	11.28865	9.75735	-1.11098
С	6.0	3.06666	7.11320	-0.75863
Н	1.0	12.05420	9.00951	-1.03089
Н	1.0	2.36026	7.91428	-0.65453
С	6.0	10.48892	11.83153	-2.00460
С	6.0	3.64585	4.93893	-1.57962
Н	1.0	10.63674	12.69670	-2.62249
Н	1.0	3.38193	4.04587	-2.11362
С	6.0	11.48382	10.86832	-1.90373
С	6.0	2.72095	5.96391	-1.43875
Н	1.0	12.40375	10.99045	-2.44377
Н	1.0	1.74018	5.86326	-1.86344
Н	1.0	9.01216	7.98632	2.15653
<b>* - 1</b>	10			

\$END

```
L
    Input file for GAMESS - [FeHdppe2]+ trip geom opt TZV
 $CONTRL SCFTYP=UHF RUNTYP=OPTIMIZE EXETYP=RUN MAXIT=200 ICHARG=1
    MULT=3 $END
 $SYSTEM MEMDDI=0 MWORDS=150 PARALL=.TRUE. $END
 $BASIS GBASIS=TZV NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $DATA
[FeHdppe2]+
C1
Fe
      26.0
               7.61025
                            8.39110
                                         2.20466
Ρ
     15.0
              8.65176
                          10.09243
                                        3.58641
Ρ
     15.0
              5.85642
                           6.98590
                                        3.47884
Ρ
     15.0
              6.67404
                          10.29066
                                        0.96975
                           6.55855
Ρ
     15.0
              7.79668
                                        0.65262
С
      6.0
              7.00150
                          11.83019
                                        2.05855
С
      6.0
              6.98387
                           5.09043
                                        1.56155
Н
      1.0
              6.89920
                          12.73308
                                        1.44533
Н
      1.0
              6.78261
                           4.28970
                                        0.83893
Н
      1.0
              6.24536
                          11.86382
                                        2.85130
Н
      1.0
              7.75789
                           4.72804
                                        2.24523
С
      6.0
             10.50260
                           9.96278
                                        3.86133
С
      6.0
              4.03206
                           7.30346
                                        3.85577
С
      6.0
              4.87450
                          10.45040
                                        0.45925
С
      6.0
              9.52926
                           5.90761
                                        0.31194
С
      6.0
              7.62639
                          10.76365
                                       -0.59070
С
      6.0
              7.00584
                           6.52452
                                       -1.06229
С
      6.0
              8.41500
                          11.75583
                                        2.66839
С
      6.0
              5.69694
                                        2.33038
                           5.45873
Н
      1.0
              8.57527
                          12.58790
                                        3.36284
Н
      1.0
              5.35490
                           4.60279
                                        2.92590
Н
      1.0
              9.17431
                          11.81136
                                        1.87868
Н
      1.0
              4.88331
                           5.71257
                                        1.64112
С
              7.94005
                          10.48308
      6.0
                                        5.28853
С
      6.0
              6.47752
                           6.26773
                                        5.10778
Н
      1.0
              8.48135
                           7.35046
                                        2.94094
С
      6.0
              3.41169
                           8.45518
                                        3.32830
С
      6.0
              3.26562
                           6.39379
                                        4.61907
С
      6.0
              2.04185
                           8.69556
                                        3.55533
С
      6.0
              1.89758
                           6.63276
                                        4.84437
С
      6.0
              1.28220
                           7.78471
                                        4.31257
Н
      1.0
              3.98449
                                        2.73262
                           9.16078
      1.0
Н
              1.57621
                           9.58544
                                        3.13843
Н
                                        4.48896
      1.0
              0.22496
                           7.96812
Н
                                        5.43144
      1.0
              1.31581
                           5.92626
Н
      1.0
              3.73243
                           5.51058
                                        5.04898
С
      6.0
              4.32672
                           9.45466
                                       -0.38173
С
      6.0
              4.05383
                          11.51338
                                        0.89518
С
      6.0
                                       -0.77745
              2.97906
                           9.52226
С
      6.0
              2.70142
                          11.57451
                                        0.50400
```

С	6.0	2.16063	10.57984	-0.33217
Н	1.0	4.94374	8.63181	-0.73501
Н	1.0	2.57034	8.75283	-1.42744
Н	1.0	1.11796	10.62941	-0.63628
Н	1.0	2.07954	12.39929	0.84360
Н	1.0	4.44864	12.30176	1.52963
С	6.0	7.05114	11.58078	-1.58785
С	6.0	8.96175	10.32843	-0.73742
С	6.0	7.80586	11.96151	-2.71351
С	6.0	9.71826	10.71550	-1.86016
С	6.0	9.14103	11.53328	-2.85100
Н	1.0	6.01969	11.91264	-1.49970
Н	1.0	7.35362	12.58948	-3.47737
Н	1.0	9.72246	11.83049	-3.72035
Н	1.0	10.74730	10.37877	-1.96114
Н	1.0	9.41528	9.68255	0.01282
С	6.0	10.63805	6.48945	0.96029
С	6.0	9.72778	4.81276	-0.55944
С	6.0	11.93369	5.97991	0.74346
С	6.0	11.02110	4.30109	-0.76937
С	6.0	12.12727	4.88364	-0.11801
Н	1.0	10.49275	7.32425	1.64107
Н	1.0	12.78300	6.43692	1.24563
Н	1.0	8.88501	4.36462	-1.08101
Н	1.0	13.12640	4.48742	-0.28262
Н	1.0	11.16451	3.45535	-1.43740
С	6.0	7.67070	7.16695	-2.13190
С	6.0	5.75949	5.90649	-1.30592
С	6.0	5.19207	5.92598	-2.59589
С	6.0	7.10516	7.18497	-3.41884
С	6.0	5.86197	6.56526	-3.65589
Н	1.0	5.42662	6.57590	-4.65195
Н	1.0	7.63357	7.67792	-4.23094
Н	1.0	8.63574	7.64022	-1.97175
Н	1.0	5.22185	5.39682	-0.51115
Н	1.0	4.23723	5.43568	-2.77061
С	6.0	7.12143	5.01454	5.18763
С	6.0	6.31976	7.03038	6.28682
С	6.0	6.79094	6.54705	7.52023
С	6.0	7.59414	4.53034	6.42333
С	6.0	7.43185	5.29451	7.59356
Н	1.0	7.26014	4.39857	4.30384
Н	1.0	5.82289	7.99550	6.25340
Н	1.0	6.65624	7.14562	8.41788
Н	1.0	7.79422	4.91973	8.54765
Н	1.0	8.08326	3.56023	6.46875
C	6.0	8.77424	10.72337	6.40084
C	6.0	8.21205	11.06255	7.64684
C	6.0	6.53796	10.57382	5.44164
C	6.0	5.97955	10.92983	6.68319
C	6.0	6.81589	11.17523	7.79022

Н	1.0	6.38559	11.44524	8.75145
Н	1.0	4.89968	11.00583	6.78672
Н	1.0	8.86380	11.24064	8.49872
Н	1.0	9.85359	10.64467	6.30722
Н	1.0	5.87539	10.34946	4.60826
С	6.0	11.35945	11.07997	3.74514
С	6.0	11.04539	8.70768	4.21699
С	6.0	12.42575	8.57279	4.45675
С	6.0	12.74202	10.94024	3.97300
С	6.0	13.27757	9.68829	4.33195
Н	1.0	14.34441	9.58381	4.51397
Н	1.0	13.39339	11.80527	3.87706
Н	1.0	12.83127	7.60424	4.73884
Н	1.0	10.39187	7.84310	4.30450
Н	1.0	10.96745	12.05908	3.48341
\$END				

```
L
    Input file for GAMESS - [FeH(H2)dppe2]+ geom opt TZV
 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN MAXIT=200
    ICHARG=1 QMTTOL=1e-5 $END
 $SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. $END
 $BASIS GBASIS=TZV NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=MOREAD NORB=224 $END
 $SCF DIRSCF=.TRUE. DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $ELDENS IEDEN=1 WHERE=NUCLEI $END
 $ELFLDG IEFLD=2 WHERE=NUCLEI $END
 $ELPOT IEPOT=1 WHERE=NUCLEI $END
 $DATA
[FeH(H2)dppe2]+
C1
      26.0
Fe
               7.17513
                            8.35825
                                         2.15545
Ρ
     15.0
              7.63039
                          10.29775
                                       3.87960
Ρ
     15.0
              6.60684
                           6.39679
                                       3.84226
Ρ
     15.0
              7.42076
                          10.38563
                                       0.46716
Ρ
     15.0
              6.83509
                          6.38450
                                       0.42940
С
      6.0
              7.45019
                                       1.57745
                          11.87701
С
      6.0
              6.91722
                           4.88790
                                       1.52969
Н
              7.79822
      1.0
                          12.74045
                                       1.02662
Н
      1.0
              6.66864
                           3.99718
                                       0.97039
Н
      1.0
              6.44555
                          12.09653
                                       1.91486
Н
      1.0
              7.93809
                           4.77271
                                       1.86709
С
                           9.98947
      6.0
              8.93728
                                       5.14107
С
      6.0
              5.29442
                           6.61251
                                       5.12176
С
      6.0
             10.25426
                           9.78016
                                       4.74216
С
              3.94457
                           6.55510
                                       4.79075
      6.0
Н
      1.0
             10.51990
                                       3.70361
                           9.80669
Н
      1.0
              3.63208
                           6.31147
                                       3.79364
С
      6.0
              6.09393
                          10.73865
                                      -0.76703
С
      6.0
              8.19479
                           6.12010
                                      -0.78562
С
      6.0
                                      -0.46852
              8.99147
                          10.57226
С
              5.29016
                                      -0.53491
      6.0
                           6.09155
С
      6.0
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                           9.90670
                                       6.49295
С
      6.0
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                           6.93909
                                       6.42859
Н
      1.0
              7.62220
                          10.06262
                                       6.82990
Н
      1.0
              6.67639
                           6.99172
                                       6.71337
С
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                          10.31827
                                      -2.08581
С
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                                      -2.14446
Н
      1.0
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                           9.85667
                                      -2.40938
Н
      1.0
              7.03240
                                      -2.51508
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С
      6.0
                                       6.01157
              5.54164
                          13.14764
С
              8.80051
                           3.77994
                                       6.12747
      6.0
Н
                                       6.43519
      1.0
              5.73708
                          14.11409
Н
      1.0
              8.61048
                           2.91314
                                       6.73142
              6.54619
С
      6.0
                          12.47425
                                       5.34825
С
      6.0
              7.75368
                           4.43040
                                       5.51069
Н
      1.0
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                          12.92914
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Н
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                           4.06221
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С	6.0	3.87217	11.51467	-1.30157
С	6.0	10.50038	5.58801	-1.23956
Н	1.0	2.95936	11.98678	-0.99074
Н	1.0	11.46943	5.29539	-0.88182
С	6.0	10.92142	9.44592	7.02136
С	6.0	3.33078	7.11827	7.04004
н	1.0	11.68681	9.24330	7.74613
н	1.0	2.57627	7.30484	7.78016
C	6.0	5.05817	10.64942	4.93954
C	6.0	9.28311	6.02164	4.57477
H	1.0	4.85750	9.66742	4.55839
Н	1.0	9.47408	6.88367	3.96896
C	6.0	8 36135	11 60690	2 78203
c	6.0	5 97868	5 04332	2 73345
н	1 0	8 53063	12 51710	3 33965
н	1 0	5 91093	4 11364	3 28239
н	10	9 32895	11 25888	2 44663
н	10	4 97967	5 28840	2 39536
Ċ	0 6 0	9 22129	11 69153	-1 26998
c	6.0	5 02023	4 86049	-1 13345
н	1 0	8 46003	12 43952	-1 38968
н	1 0	5 73013	4 05846	-1 06964
Ċ	6.0	11 23892	9 52071	5 67627
c	6.0	2 97054	6 80228	5 74320
н	1 0	12 25226	9 37481	5 35276
н	1 0	1 93381	6 74033	5 47091
Ċ	6.0	4 02956	11 10069	-2 61133
c	6.0	10 29198	5 79416	-2 59220
н	1 0	3 23959	11 24729	-3 32294
н	1 0	11 09768	5 66140	-3 28891
Ċ	6.0	4 89434	11 33142	-0 38603
c	6.0	9 46388	5 75951	-0 34204
н	1 0	4 74073	11 66519	0.57201
н	1 0	9 65771	5 62162	0.02211
Ċ	6.0	9 61371	9 63333	7 42596
c	6.0	4 67041	7 18763	7 37870
н	1 0	9 35779	9 57636	8 46711
н	10	4 95987	7 42921	8 38383
Ċ	1.0 6 0	4 28465	12 57452	6 13684
c	6.0	10 00706	1 21750	5 96813
с н	1 0	3 50360	13 10058	6 65237
н	10	10 91290	3 73983	6 44854
Ċ	1.0 6 0	5 21363	10 10071	_2 00033
c	6.0	0 0/107	6 17217	-2.99955
с ц	1 0	5 34557	10.17550	-1 01/10
н Ц	1.0	2 27015	6 22691	4.01419
n C	L.0	0.0104J 1 01570	11 27277	5 60620
c	0.0 6 0	10 22611	5 36116	5 10570
с ц	1 0	3 00000 TO: 22044	10 86517	5 7127/
н	10	J.000000 11 32722	5 7272/	5 07202
C II	1.U 6 0	0 007/7	0 61656	-0 31281
C	0.0	J. JU(H)	2.01020	0.07004

С	6.0	4.36176	7.11218	-0.67700
Н	1.0	9.82698	8.75758	0.27264
Н	1.0	4.55402	8.08217	-0.26279
С	6.0	6.31613	11.21559	4.79259
С	6.0	7.98497	5.55998	4.72454
С	6.0	11.19666	9.77520	-1.00903
С	6.0	3.18293	6.91130	-1.38054
Н	1.0	11.95938	9.02715	-0.90683
Н	1.0	2.47991	7.71596	-1.48159
С	6.0	10.42132	11.84662	-1.92961
С	6.0	3.84996	4.65771	-1.83419
Н	1.0	10.58542	12.71029	-2.54562
Н	1.0	3.65983	3.70492	-2.29060
С	6.0	11.41421	10.88544	-1.79706
С	6.0	2.92438	5.68431	-1.95481
Н	1.0	12.34956	11.00811	-2.30985
Н	1.0	2.01408	5.52466	-2.50087
Н	1.0	8.77710	8.04223	2.15755
Н	1.0	5.08160	8.49069	1.91258
Н	1.0	5.19078	9.03890	2.39595
÷				

\$END

## 5.3.2 Vibrational Analysis

```
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    Input file for GAMESS - [Mn(CO)dppe2]+ vib anal
 $CONTRL SCFTYP=RHF RUNTYP=HESSIAN EXETYP=RUN MAXIT=100
    ICHARG=1 ECP=READ DFTTYP=B3LYP
 $END
 $SYSTEM MEMDDI=0 MWORDS=300 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 NPFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $FORCE METHOD=SEMINUM NVIB=2 SCLFAC=0.96 VIBANL=.TRUE.
   TEMP(1)=295 TEMP(2)=298 TEMP(3)=300 TEMP(4)=313
   TEMP(5)=333 TEMP(6)=343 TEMP(7)=353 TEMP(8)=363
   TEMP(9) = 373
 $END
 $DATA
Mn(CO)dppe2
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Ρ
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                          10.48348
                                       0.40231
Ρ
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             10.03343
                          8.68254
                                       0.46829
С
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                          12.00670
                                      -0.06697
С
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                          12.80365
                                       0.92860
С
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              9.18881
                          13.92344
                                       0.57182
С
      6.0
              9.37064
                          14.25496
                                      -0.78388
С
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              8.77771
                                      -1.78008
                          13.45677
                                      -1.42714
С
      6.0
              8.00270
                          12.33682
С
              6.09011
      6.0
                          10.95594
                                       2.08989
С
      6.0
              5.41715
                          12.19073
                                       2.24304
С
      6.0
              4.85152
                          12.54799
                                       3.47908
С
      6.0
              4.95272
                                       4.58227
                          11.67639
С
      6.0
              5.62140
                                       4.43943
                          10.44598
С
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              6.18261
                          10.08777
                                       3.19684
С
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С
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                          11.02012
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С
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                          11.87684
                                      -1.11772
С
             12.33958
                                      -2.48175
      6.0
                          11.54598
С
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             11.76096
                          10.35003
                                      -2.94897
С
      6.0
             11.09311
                           9.49230
                                      -2.05749
С
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             10.64164
                           9.22840
                                       2.17911
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      6.0
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                           9.04822
                                       2.55223
С
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             12.45650
                           9.47932
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                          10.10942
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                          10.30581
                                       4.34160
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С
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                          10.60882
                                      -0.65777
С
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                                       0.27383
      6.0
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Н
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                                       1.98092
Н
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                                       1.34994
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                          15.12063
                                      -1.05957
Н
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                          11.72992
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Н
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Н
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Н	1.0	5.71092	9.77276	5.28788
Н	1.0	6,70066	9.13493	3.10543
Н	1.0	11.51071	11.29539	0.82472
Н	1.0	12.68419	12,79936	-0.74656
н	10	12 85050	12 21231	-3 17239
н	1 0	11 81724	10 08961	-4 00281
н	1 0	10 64476	8 58127	-2 44477
н	1 0	12 70675	8 59982	1 86342
н	1.0	13 /0075	0.3331/	1.00342
н	10	11 93331	10 44584	5 67969
ц	1 0	0 54480	10.70870	5 02560
II L	1.0	9.34400	10.73073	2 91571
п	1.0	0.73402 1 71921	11 55750	2.01371
п	1.0	4.71034 5 51441	10 60044	1 71/12
	1.0	J.J1441 11 25777	10.00044	
п	1.0	11 01710	6.90072	-0.70025
П	15.0		0.98247	0.90750
P	15.0	8.44075	2.93852	-0.48843
P	15.0	5.25408	7.75887	-0.54741
0	8.0	7.87916	9.25753	-2.93164
C	6.0	7.44457	4.42678	0.03607
(	6.0	6.84619	3.59186	-0.93305
(	6.0	6.10695	2.46108	-0.54091
(	6.0	5.94889	2.15606	0.82408
(	6.0	6.52941	2.99370	1.79527
(	6.0	7.26987	4.12484	1.40452
C	6.0	9.23256	5.34581	-2.11054
C	6.0	9.92018	4.10880	-2.13543
C	6.0	10.53720	3.65761	-3.31516
С	6.0	10.47460	4.43437	-4.49026
C	6.0	9.78854	5.66144	-4.47391
С	6.0	9.17572	6.11731	-3.28966
С	6.0	4.37805	6.60626	0.66776
С	6.0	3.75902	5.41269	0.24249
С	6.0	3.14358	4.55928	1.17719
С	6.0	3.14423	4.88402	2.54591
С	6.0	3.76927	6.07002	2.97799
С	6.0	4.38428	6.92451	2.04473
С	6.0	4.62751	7.15670	-2.22522
С	6.0	3.26925	7.31135	-2.58869
С	6.0	2.80543	6.85034	-3.83361
С	6.0	3.68904	6.21332	-4.72764
С	6.0	5.03906	6.04682	-4.36835
С	6.0	5.50333	6.51842	-3.12498
С	6.0	7.82093	8.83046	-1.81339
С	6.0	9.97754	5.85093	0.65043
С	6.0	4.31352	9.41607	-0.32056
Η	1.0	6.96963	3.80229	-1.99186
Η	1.0	5.66034	1.82250	-1.29933
Н	1.0	5.38356	1.27781	1.12612

Н	1.0	6.40	0590		2.770	)34	2.	85201
Н	1.0	7.70	0859		4.754	173	2.	17588
Н	1.0	9.9	5301		3.483	354	-1.	24677
Н	1.0	11.0	5700		2.708	360	-3.	31765
Н	1.0	10.94	4763		4.083	353	-5.	40401
Н	1.0	9.7	2549		6.267	707	-5.	37444
Н	1.0	8.6	5213		7.069	963	-3.	30953
Н	1.0	3.74	4910		5.142	293	-0.	80890
Н	1.0	2.6	5763		3.644	129	0.	83320
Н	1.0	2.6	5894		4.222	232	3.	26577
Н	1.0	3.7	7553		6.330	004	4.	03384
Н	1.0	4.84	4723		7.842	247	2.	39876
Н	1.0	2.5	5812		7.769	962	-1.	90574
Н	1.0	1.7	5890		6.978	388	-4.	09912
Н	1.0	3.3	3012		5.858	327	-5.	69035
н	1.0	5.7	2990		5.556	587	-5.	05009
н	1.0	6.5	4792		6.390	)60	-2.	86795
н	1.0	10.4	5594		4.878	301	0.	53073
н	1 0	9 6	5357		5 935	540	1	69541
н	10	3 9	7098		9 470	018	<u>0</u>	71924
н	10	3 42	2691		9 422	236	-0	96062
 \$I	=ND	0.1			5.122		•••	50002
\$1	=CP							
M	N-FCP GEN	1(	2	2				
3		d n	- ntent	ial	_			
-10 0000000 1 357 3914469								
	-60 29902	287		2	64	64773	29 29	
	-10 42178	234		2	16	09608	33	
5	10.42110	s-d	note	∽ n+ia	1		-	
5	3 00000	100	poce	0	107	41272	15	
	16 25918	R19		1	111	49589	73	
	276 93730	328		- 2	46	556834	16	
	241 31743	20 847		2	-0. 8	368813	70 25	
_	-1/6 /6353	270		2	7	72374	20	
5	-1-0055.	n_d	note	∠ n+ia	, , 1		-	
5	5 00000	700 700	poter	0 0	80	041510	22	
	5 75802	756		1	105	604364	16	
	285 20186	50		2	40	830040	-0 	
	143 42226	547		2	-0- 8	00084	50 57	
	-88 70319	251	,	2	7	220002	28	
D.	-00.70310	10		2	( ,		_0	
5		ы д	note	∠ n+ia	1		_	
J	_10 00000	200	pote	1	162	1211/	22	
	-10.00000	58		1 2	402.	686370	2.5 21	
	-78 36683	251	·	2 2	95. 21	231000	24	
	0 9577	200		2 7	6	22001	15	
	-9.0377.	209 705		2 2	0. 2	062069	27	
5	-1.0102	ده د م	noto	c ntia	 1	002000	54	
J	2 0000	s-u 200	poter	0 0	70	08210	22	
	17 010/	151		ย 1	(0. 50	02260	10	
	150 02502	208		1 2	2C 20.	05712	55	
	170.0720	200		<u> </u>	. טכ	ωσιτζ	رر	

7′	1 7083146	2	11 2464453
2	3 0397012	2	2 6757561
6	n-d	notential	
С г	5 0000000	0	75 1617880
6	5 3446507	1	57 4544041
109	2 5585101	2	<i>A7</i> 0 <i>A</i> 817 <i>A</i>
111	1 1470820	2	18 /588360
11.	2011111	2	5 0/1/100
44	5 1102722	2	1 8487507
	5.4403233	2	1.040/30/
	NONE		
	NUNE		
C-ECP	NUNE		
C-ECP	NONE		
C-ECP	NUNE		
C-ECP	NUNE		
H-ECP	NONE		
H-FCP	NONE		

H-ECP	NONE
H-ECP	NONE
P-FCP	
P-FCP	
0-ECP	NONE
C-FCP	NONE
C-FCP	NONE
C-FCP	NONE
	NONE
	NONE
	NONE
C-ECP	NONE
H-ECP	NONE

H-ECP	NONE	
H-ECP	NONE	
\$END		

```
L
    Input file for GAMESS - [Mn(CO)dppe2]+ agostic vib anal
 $CONTRL SCFTYP=RHF RUNTYP=HESSIAN EXETYP=RUN DFTTYP=B3LYP
    MAXIT=150 ICHARG=1 ECP=READ $END
 $SYSTEM MEMDDI=0 MWORDS=150 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2
   TEMP(1)=295 TEMP(2)=298 TEMP(3)=300 TEMP(4)=313 TEMP(5)=333
   TEMP(6)=343 TEMP(7)=353 TEMP(8)=363 TEMP(9)=373
 $END
 $DATA
Mn(CO)dppe2
C1
Mn
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                           6.28789
                                        5.69874
Ρ
     15.0
              8.01900
                           5.20284
                                       6.52822
Ρ
     15.0
              3.96610
                           7.20457
                                       4.77793
Ρ
     15.0
              4.48275
                           5.24569
                                       7.31666
Ρ
     15.0
              7.47139
                          6.87365
                                       3.80446
0
      8.0
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                          8.73462
                                       7.27172
С
                                       8.33794
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              8.53389
                           5.20945
С
      6.0
              9.00099
                          6.40752
                                       8.92414
С
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              9.37543
                          6.43610
                                      10.27997
С
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              9.27948
                           5.27194
                                      11.06670
С
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                          4.07888
                                      10.48855
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С
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С
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                                       3.20369
С
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                           5.68514
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С
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                          4.53087
                                       1.76519
С
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                          4.19831
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С
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                          5.01077
                                       2.98091
С
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                          8.95877
                                       4.11124
С
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                                       5.00273
С
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С
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                                      11.11280
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                                      11.94216
С
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C	6.0	3.58434	1.36811	6.00419
C	6.0	4.17197	0.54423	6.98315
C	6.0	4.84608	1.13205	8.07073
Ċ	6.0	4 92731	2 53244	8 18395
c	6.0	7 02143	6 06276	2 15756
c	6.0	6 40426	6 81404	1 13201
c	6.0	6 01802	6 19510	-0 07088
c	6.0	6 23807	1 81825	-0 26487
c	6.0	6 84512	4.01025	0.20407
c	6.0	7 22038	4.00198	1 96293
c	6.0	7 02226	9 62600	2 20627
c	6.0	7.92220 0.01777	0.00090	2.20027
C	0.0	0.01///	0.04551 10.14246	2.22955
C	0.0	9.21724	10.14540	1.87002
C	6.0	8.72974	11.25283	2.59805
C	6.0	7.84051	11.05097	3.66830
C	6.0	7.43674	9.74799	4.02229
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C	6.0	2.75366	7.22978	6.25144
Н	1.0	6.36785	3.19858	5.09725
Н	1.0	4.61062	4.75280	3.20248
Н	1.0	9.07261	7.32194	8.34007
Н	1.0	9.74146	7.36221	10.71654
Н	1.0	9.57188	5.29354	12.11354
Н	1.0	8.72972	3.17573	11.08938
Н	1.0	8.08682	3.11346	8.69559
Н	1.0	10.36943	3.30844	6.74975
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Н	1.0	8.99318	-0.19218	4.62961
Н	1.0	6.78012	0.90984	4.24445
Н	1.0	1.19642	7.39177	3.61161
Н	1.0	-0.09402	5.95259	2.06415
Н	1.0	0.95060	3.90565	1.08015
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Н	1.0	4.04017	9.89639	6.06928
Н	1.0	3.84447	12.20316	5.21669
Н	1.0	3.53141	12.62798	2.77144
Н	1.0	3.42257	10.69905	1.18550
Н	1.0	3.58926	8.37519	2.03048
Н	1.0	6.16425	6.99596	9.11667
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Н	1.0	4.41078	6.38388	13.00547
Н	1.0	2.79993	4.74016	12.02234
Н	1.0	2.86652	4.22506	9.60229
Н	1.0	3.20372	3.37906	5.34251
Н	1.0	3.05441	0.92580	5.16375
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Н	1.0	5.3	0182		0.503	358	8.83195
Н	1.0	5.4	3330		2.969	959	9.04025
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Н	1.0	5.9	4926		4.343	395	-1.19923
Н	1.0	7.0	2827		2.999	934	0.61460
Н	1.0	7.6	9693		4.067	703	2.73133
Н	1.0	9.1	9586		7.999	969	1.65568
Н	1.0	9.9	0437		10.290	019	1.04694
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Н	1.0	9.2	7648		5.142	134	3.74610
Н	1.0	9.9	5751		6.77	326	3.66171
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Н	1.0	10.4	0675		5.604	401	5.91655
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	276.93739	28		2	46	. 556834	46
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	143.42226	47		2	8	.00984	57
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P	-ECP GEN	10		2			
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	-28.36682	51		2	21	.234909	94
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5		- s-d	pote	enti	al		-
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P-ECP							
P-ECP							
P-ECP							
O-ECP NONE							
C-ECP NONE							

C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-FCP	NONE
H-ECP	NONE
H-FCP	NONE
H-ECP	NONE

H-ECP	NONE
H-ECP	NONE
\$END	

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L
    Input file for GAMESS - [Mn(H2)(C0)dppe2] + vib anal
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 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=MOREAD NORB=205 $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2
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    TEMP(5)=333 TEMP(6)=343 TEMP(7)=353 TEMP(8)=363
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 $END
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Ρ
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С
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                                       0.96929
С
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                          13.92583
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С
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                          13.55812
                                      -1.72116
С
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                          12.39751
                                      -1.37986
С
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      6.0
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С
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                          12.20460
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Н	1.0	12.70545	12.83506	-0.81250
Н	1.0	13.13198	12.13301	-3.17519
Н	1.0	12.19623	9.96741	-4.00694
Н	1.0	10.87809	8.52094	-2.50886
Н	1.0	12.76014	8.88775	1,69797
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н	10	11 80980	10 21800	5 69004
н	10	9 38515	10.21000	5 07802
н	10	8 65448	9 69031	2 79720
н	1.0	1 765/1	11 53268	-0 11573
Ц	1.0	5 57637	10 57036	-0.773
Ц	1.0	11 33035	6 96520	-0.00108
п	10	11 20527	7 04599	0.30108
П	15.0	11.09J0/ 0 E12E4	7.04J00 E 04207	0.77144
Р П	15.0	0.JLJJ4	5.94507	-0.50221
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0	8.0	7.91183	9.28937	-2.88259
C	6.0	7.52694	4.44538	0.02702
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(	6.0	6.05/10	2.20175	0.93507
(	6.0	6.71600	3.03656	1.85/4/
C	6.0	7.44372	4.15268	1.40612
С	6.0	9.23311	5.33421	-2.20904
С	6.0	9.79132	4.03508	-2.27776
С	6.0	10.39128	3.57597	-3.46305
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С	6.0	9.88975	5.69953	-4.53872
С	6.0	9.29064	6.16188	-3.34951
С	6.0	4.32183	6.64454	0.74035
С	6.0	3.82768	5.37662	0.36970
С	6.0	3.11709	4.59018	1.29594
С	6.0	2.89676	5.05564	2.60493
С	6.0	3.39871	6.31543	2.98558
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С	6.0	4.67751	7.09314	-2.15619
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С	6.0	2.83863	6.65365	-3.71305
С	6.0	3.75749	6.20138	-4.68230
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С	6.0	5.58691	6.63306	-3.12738
С	6.0	7.84068	8.83405	-1.78076
С	6.0	10.09607	5.84994	0.50388
С	6.0	4.34801	9.39419	-0.33860
Н	1.0	6.89503	3.81985	-1.95823
Н	1.0	5.61639	1.85964	-1.16031
Н	1.0	5.50375	1.33293	1.28270

	1 0	C C	C 7 7 7	2	02057	2 02102	
п	1.0	0.0	0121	2.0	72020	2.92105	
н	1.0	7.9	4590	4.	//83I	2.14128	
Н	1.0	9.7	4715	3.	37165	-1.41727	
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Н	1.0	10.9	0583	4.0	05296	-5.51695	
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н	1 0	8 8	7666	7	16245	-3 33638	
ц	1 0	3 0	2110	1	00867	-0 63624	
	1.0	2.5		т. Э	C1 770	-0.03024	
п	1.0	2.7	5790	5.0	01779	0.99100	
Н	1.0	2.3	4048	4.4	44980	3.31603	
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Н	1.0	3.4	0419	5.3	86442	-5.65381	
н	1.0	5.8	4950	5	84186	-5.12391	
н	1 0	6.6	1939	6	62192	-2 91673	
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п	1.0	9.0	1000	5.0	12200	1.50515	
н	1.0	3.9	4833	9.4	47299	0.67791	
Н	1.0	3.5	0022	9.	37783	-1.02964	
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\$E	ECP						
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F	-10.42170	C	noto	د . سبن ما	10.05	00000	
5	2 00000	s-u	ροτε		-		
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	20 70210	251		2	7 22	00020	
п		10		2	1.55	90928	
Р- г	-ECP GEN	10		2			
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5	-1.01637	783 s-d	pote	2 ntial	2.06	20684	

12.9104154 150.0250298 71.7083146 23.0397012	1 2 2 2	58.9576810 36.0571255 11.2464453 2.6757561
6 p-d	potential	
5.000000	0	75.1617880
6.3446507	1	57.4544041
198.5585104	2	47.9481748
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P-ECP		
C-ECP NONE		
H-ECP NUNE		
H-ECP NONE		
H-ECP NUNE		
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H-ECP	NONE
P-ECP	
P-ECP	
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H-ECP	NONE

H-ECP	NONE
H-ECP	NONE
\$END	

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    Input file for GAMESS - [FeHdppe2]+ vib anal
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 $GUESS GUESS=MOREAD NORB=198 $END
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 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2
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   TEMP(6)=343 TEMP(7)=353 TEMP(8)=363 TEMP(9)=373
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 $DATA
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Ρ
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Ρ
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С
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С	6.0	3.39954	7.29479	7.01234
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С	6.0	4.85137	10.40700	4.27969
С	6.0	9.50658	6.21346	4.40679
Н	1.0	4.71591	9.52263	3.66077
Н	1.0	9.68999	7.03274	3.71935
С	6.0	8.38819	11.52183	2.83306
С	6.0	6.09952	5.11298	2.77129
Н	1.0	8.37847	12.39948	3.48781
Н	1.0	6.17048	4.22113	3.40250
Н	1.0	9.43063	11.28733	2.59248
Н	1.0	5.04403	5.27121	2.52325
С	6.0	9.27002	11.32606	-1.33870
С	6.0	4.97831	5.36924	-1.19685
Н	1.0	8.46965	12.03736	-1.53158
Н	1.0	5.74679	4.62617	-1.39961
С	6.0	11.20621	9.49627	5.84253
С	6.0	3.04974	7.04665	5.67098
Н	1.0	12.25963	9.43363	5.58015
Н	1.0	2.00325	7.01445	5.37700
С	6.0	4.00109	10.92016	-2.51903
С	6.0	10.20389	5.89668	-2.73773
Н	1.0	3.19244	11.09448	-3.22437
Н	1.0	10.97056	5.71956	-3.48792
С	6.0	4.90887	11.07736	-0.25892
С	6.0	9.46885	5.84593	-0.41173
Н	1.0	4.77161	11.37866	0.77773
Н	1.0	9.69639	5.63926	0.63164
С	6.0	9.43955	9.47623	7.51580
С	6.0	4.75773	7.32504	7.38124
Н	1.0	9.12286	9.39704	8.55301
Н	1.0	5.03713	7.51107	8.41558
С	6.0	3.87674	12.18593	5.63306
С	6.0	10.34552	4.48964	5.91799
Н	1.0	3.00441	12.69072	6.04093
Н	1.0	11.17749	3.97907	6.39706
С	6.0	5.19974	10.31968	-2.94915
С	6.0	8.94574	6.40127	-3.11601
Н	1.0	5.31871	10.02645	-3.98949
Н	1.0	8.73822	6.61769	-4.16124
С	6.0	3.71907	11.05922	4.80525
С	6.0	10.58435	5.55325	5.02905
Н	1.0	2.72441	10.68411	4.57718
Н	1.0	11.60286	5.87230	4.82122

С	6.0	10.1	3014		9.	352	278	-0	.19282
С	6.0	4.2	3828		7.	419	951	-0	.11071
Н	1.0	9.9	8747		8.	537	'41	0	. 50870
Н	1.0	4.4	3571		8.	282	214	0	.52127
С	6.0	6.1	4925		10.	873	354	4	.57462
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С	6.0	11.3	5621		9.	497	<b>'</b> 49	-0	.87106
С	6.0	2.9	7261		7.	283	35	-0	.71356
Н	1.0	12.1	5776		8.	785	546	-0	.69024
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С	6.0	3.7	1579		5.	233	818	-1	.80054
Н	1.0	10.6	3049		12.	284	19	-2	.72031
Н	1.0	3.5	1867		4.	388	856	-2	.45612
С	6.0	11.5	4150		10.	552	227	-1	.78279
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Н	1.0	12.4	8825		10.	663	858	-2	.30587
Н	1.0	1.7	3300		6.	084	10	-2	.02704
Н	1.0	8.6	3123		8.	027	'56	2	.14222
\$E	END								
\$E	ECP								
FE	E-ECP GEN	1	0	2	<u>,</u>				
3		- d p	otent	tio	ıl	-			
	-10.00000	00		1	3	392.	61497	87	
	-63.26675	18		2		71.	17569	79	
	-10.96133	38		2		17.	73202	81	
5		- s-d	pote	ent	ial			-	
	3.00000	00		0	1	L26.	05718	95	
	18.17291	.37		1	1	L38.	12642	51	
	339.12311	64		2		54.	20988	58	
	317.10680	12		2		9.	28379	66	
-	-207.34216	49		2		8.	62890	82	
5		- p-d	pote	ent	ial			-	
	5.00000	00		0		83.	17594	90	
	5.95359	30		1	1	L06.	05599	38	
	294.26655	27		2		42.	82849	37	
	154.42446	35		2		8.	77018	05	
	-95.31642	49		2		8.	03978	18	
P-	-ECP GEN	10		2					
5		- d	pote	ent	ial			-	
	-10.00000	00		1	2	162.	12114	23	
	-79.48646	58		2		93.	68637	01	
	-28.36682	51		2		21.	23490	94	
	-9.85775	89		2		6.	33884	15	
	-1.01637	83		2		2.	06206	84	
5		- s-d	pote	ent	ial			-	
	3.00000	00		0		78.	08318	23	
	12.91041	54		1		58.	95768	10	
	150.02502	98		2		36.	05712	55	
	71.70831	46		2		11.	24644	53	
	23.03970	12		2		2.	67575	61	

6	p-d	potential	
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6	5.3446507	1	57.4544041
198	3.5585104	2	47.9481748
111	1470820	2	18.4588360
40	3944144	2	5 9414190
6	5 4483233	2	1 8487507
P_FCP		L	1.040/50/
P-FCP			
	NONE		
	NONE		
C-FCP	NONE		
H-FCP	NONE		
H-FCP	NONE		
C-FCP	NONE		
C-FCP	NONE		
H-FCP	NONE		
H-ECP	NONE		
(-FCP	NONE		
(-FCP	NONE		
H-FCP	NONE		
H-FCP	NONE		
(_FCP	NONE		
C_FCP	NONE		
H-FCP	NONE		
H-FCP	NONE		
C-FCP	NONE		

C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
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C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-FCD	NONE

C-ECP	NONE
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H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
H-ECP	NONE
H-ECP	NONE
\$END	

```
L
    Input File for GAMESS - [FeH(H2)dppe2]+ vib anal
 $CONTRL SCFTYP=RHF RUNTYP=HESSIAN EXETYP=RUN DFTTYP=B3LYP
    MAXIT=200 ICHARG=1 ECP=READ $END
 $SYSTEM MEMDDI=0 MWORDS=150 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=MOREAD NORB=199 $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2
  TEMP(1)=295 TEMP(2)=298 TEMP(3)=300 TEMP(4)=313 TEMP(5)=333
  TEMP(6)=343 TEMP(7)=353 TEMP(8)=363 TEMP(9)=373
 $END
 $DATA
[FeH(H2)dppe2]+
C1
Fe
      26.0
               6.97167
                           8.39792
                                        2.15327
Ρ
     15.0
              7.54614
                          10.05383
                                       3.76618
Ρ
     15.0
              6.64272
                          6.67353
                                       3.76011
Ρ
     15.0
              7.35390
                          10.12366
                                       0.54925
Ρ
     15.0
              6.83222
                          6.64351
                                       0.54574
                          11.75571
С
      6.0
              7.43224
                                       1.53864
С
      6.0
              6.83850
                          5.00383
                                       1.53105
Н
      1.0
              7.85410
                          12.54044
                                       0.90219
Н
      1.0
              6.46173
                          4.20151
                                       0.88973
                                       1.82650
Н
      1.0
              6.41868
                          12.05642
Н
                          4.76484
      1.0
              7.87203
                                       1.80209
С
      6.0
              8.86552
                          9.74790
                                       5.07736
С
      6.0
              5.35766
                          6.88789
                                       5.12295
С
      6.0
             10.22197
                          9.64709
                                       4.69699
С
      6.0
              3.98417
                          6.88905
                                       4.78906
Н
      1.0
             10.51330
                          9.73493
                                       3.65276
Н
      1.0
              3.66063
                          6.72680
                                       3.76263
С
      6.0
              6.02789
                          10.48040
                                      -0.74263
С
      6.0
              8.23188
                          6.36007
                                      -0.68614
С
              8.97986
      6.0
                          10.25492
                                      -0.39300
С
      6.0
              5.27669
                           6.38382
                                      -0.50292
С
      6.0
              8.52137
                          9.60631
                                       6.43870
С
      6.0
              5.73567
                          7.10461
                                       6.46570
Н
      1.0
              7.48523
                          9.69195
                                       6.75457
Н
      1.0
              6.78546
                          7.10441
                                       6.74542
С
      6.0
              6.19450
                          10.09710
                                      -2.09106
С
      6.0
              8.06734
                          6.63618
                                      -2.06000
Н
              7.12366
                          9.63826
                                      -2.41801
      1.0
      1.0
              7.11279
                                      -2.43652
Н
                          6.99409
С
              5.51274
                                       6.14814
      6.0
                          12.84594
С
      6.0
              8.99841
                          4.11143
                                       6.08724
Н
      1.0
              5.77436
                          13.73602
                                       6.71517
Н
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              8.82745
                          3.29289
                                       6.78220
С
      6.0
              6.51645
                          12.12801
                                       5.47512
С
      6.0
                                       5.52282
              7.90265
                          4.78760
Н
      1.0
              7.54700
                          12.46793
                                       5.55040
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Н	1.0	6.89431	4.48314	5.79481
С	6.0	3.78699	11.30822	-1.28366
С	6.0	10.53277	5.65456	-1.12847
Н	1.0	2.86212	11.78432	-0.96654
Н	1.0	11.48110	5.26708	-0.76337
С	6.0	10.87122	9.29365	7.02321
С	6.0	3.39033	7.29636	7.12061
н	1.0	11.64194	9.12619	7.77149
н	1.0	2.63583	7.44472	7.88914
С	6.0	4.85498	10.53670	4,68639
C	6.0	9.42735	6.22956	4.28478
Н	1.0	4.58614	9.63399	4.14808
Н	1.0	9.58976	7.04663	3.58961
C	6.0	8.29618	11.50808	2.78266
Ċ	6.0	5.97585	5.16974	2.78957
H	1.0	8.33587	12.39529	3.42174
Н	1.0	6.02400	4.28176	3.42847
Н	1.0	9.32163	11.24704	2.50031
Н	1.0	4.92672	5.34544	2.52624
C	6.0	9,18193	11.30107	-1.32399
Ċ	6.0	5.14661	5.21233	-1.28699
H	1.0	8.37713	11,99435	-1.55903
н	1.0	5.96210	4,49457	-1.34482
C	6.0	11.21902	9.42856	5.66498
Ċ	6.0	3.00660	7.08846	5.78138
H	1.0	12,26129	9.36715	5.36108
н	1.0	1.95332	7.07367	5.51133
C	6.0	3 96273	10 92987	-2 62907
c	6.0	10 36154	5 92608	-2 50007
н	1 0	3 17409	11 10953	-3 35540
н	10	11 17594	5 75047	-3 19852
C	6.0	4 81120	11 08016	-0 34577
c	6.0	9 47738	5 87812	-0 22596
н	1 0	4 64948	11 38176	0.68722
н	1 0	9 63741	5 67683	0.83076
C	6.0	9 51965	9 38033	7 40529
c	6.0	4 75694	7 30474	7 45769
н	1 0	9 24212	9 28092	8 45197
н	1 0	5 06197	7 46264	8 48937
C	6.0	4 17232	12 41216	6 09076
c	6.0	10 31409	4 49104	5 75115
н	1 0	3 39617	12 96740	6 61181
н	1.0	11 16096	3 96629	6 18674
C	6.0	5 16865	10 32265	-3 02777
c	6.0	9 12777	6 42196	-2 96087
н	1 0	5 31265	10 02769	-4 06438
н	1 0	8 98674	6 63413	-4 01705
Ċ	4.0 6 0	3 84811	11 25267	5 26200
c	6.0	10 52116	5 55152	L 251/2
н	1 0	2 82038	10 89992	5 32751
н	1.0	11,53631	5,85452	4,59285
	2.0		2.00.00	

C	6.0	10.0	3728		9.370	060	-0.10588
c	6.0	4 2	1477		7 300	332	-0 47990
н	1 0	9.8	8499		8 568	228	0 60859
ц	1.0	J.0 ∕ 3	0933		8 230	77	0.00000
C II	1.0	т.J С 1	0522		10.250	)57	0.000 <del>1</del> 3
c	0.0	0.1	1177		10.905	127 070	4.72772
C	6.0	8.1	1122		5.853	378	4.61600
C	6.0	11.2	8280		9.522	277	-0.74613
С	6.0	3.0	3553		7.064	156	-1.21190
Н	1.0	12.0	8956		8.828	324	-0.52513
Н	1.0	2.2	2831		7.792	238	-1.18859
С	6.0	10.4	2546		11.451	L41	-1.96198
С	6.0	3.9	7094		4.966	519	-2.01620
Н	1.0	10.5	7147		12.257	709	-2.67729
Н	1.0	3.8	8637		4.060	)64	-2.61205
C	6.0	11 4	8052		10 561	130	-1 67400
c	6.0	2 0	0727		5 800	150	_1 07580
с u	1.0	12.5	0121 1170		10 67	766	2 16901
п	1.0	1 0	4170 0611		T0.0//		-2.10691
п	1.0	1.9	9011		5.090	509	-2.55/1/
н	1.0	8.4	4932		8.074	148	2.15763
Н	1.0	5.2	6921		8.364	157	2.02598
Н	1.0	5.4	2761		9.126	583	2.24373
\$E	END						
\$E	ECP						
FE	E-ECP GEN	1	0	2			
3		d p	oten	tia	1-		
	-10.0000	000		1	392.	.614978	87
	-63.2667	518		2	71.	175697	79
	-10.9613	338		2	17.	732028	81
5		s-d	pot	ent	ial		_
•	3 0000	200	Pee	0	126	057189	95
	18 1729	137		1	138	12642	53
	330 1731	164		2	54	20022	58
	217 1069	10 <del>1</del> 212		2	ТС О	20300.	56
	207 2421	C 4 O		2	9.	C20000	
	-207.34210	049		Ζ.	× , ×	.628908	82
5	 - 0000	p-a	pot	ent	101	175040	-
	5.00000	000		0	83.	.175945	90
	5.95359	930		1	106.	.05599:	38
	294.2665	527		2	42.	. 828493	37
	154.4244	635		2	8.	.770180	<b>0</b> 5
	-95.31642	249		2	8.	.039781	18
P	-ECP GEN	10		2			
5		d	pot	ent	ial		-
	-10.0000	000		1	462.	.121142	23
	-79.4864	658		2	93.	686370	01
	-28.36682	251		2	21.	234909	94
	-9.8577	589		2	6	33884	15
	-1 0163	783		2	2	062065	 R4
5		. 55 c_4	no+	∟ ≏n+	ial		_
5	2 0000	300 300	PUC	n	70	082191	73
	12 0104	151		1	(0) 50	05760	10
	150 0250	200		т Э	20.	057100. 1057101	10
	130.0230	∠ 70		2	. מכ	באבוכש.	נו

71.7083146	2	11.2464453
23.0397012	2	2.6757561
6 p-d	potential	
5.0000000	0	75.1617880
6.3446507	1	57.4544041
198.5585104	2	47.9481748
111.1470820	2	18.4588360
40.3944144	2	5.9414190
6.4483233	2	1.8487507
P-FCP		
P-ECP		
P-ECP		
C-ECP NONE		
C-ECP NONE		
H-ECP NONE		
C-ECP NONE		
H-ECP NONE		
H-ECP NONE		
C-ECP NONE		
H-ECP NONE		
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C-ECP NONE		
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C-ECP	NONE
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H-ECP	NONE
H-ECP	NONE
C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
\$END	

## 5.3.3 Total Energy Calculation

```
!
    Input file for GAMESS - H2 molecule total energy
 $CONTRL SCFTYP=RHF RUNTYP=OPTIMIZE EXETYP=RUN MPLEVL=2
    MAXIT=30 $END
 $SYSTEM MEMDDI=12 MWORDS=50 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=1.0e-7 $END
 $STATPT OPTTOL=0.0000001 TRMIN=0.01 NSTEP=100 HSSEND=.T. $END
 $FORCE METHOD=SEMINUM NVIB=2 VIBSIZ=0.01 SCLFAC=0.96 VIBANL=.TRUE.
   TEMP(1)=295 TEMP(2)=298 TEMP(3)=300 TEMP(4)=313 TEMP(5)=333
   TEMP(6)=343 TEMP(7)=353 TEMP(8)=363 TEMP(9)=373
 $END
 $DATA
H2 molecule
C1
Н
      1.0
             -0.02189
                         0.00000
                                     0.00000
Н
      1.0
             0.72189
                         0.00000
                                     0.00000
 $END
```

```
!
    Input file for GAMESS - [Mn(CO)dppe2]+ total energy
 $CONTRL SCFTYP=RHF RUNTYP=ENERGY EXETYP=RUN MAXIT=100
    ICHARG=1 ECP=READ MPLEVL=2
 $END
 $SYSTEM MEMDDI=5000 MWORDS=75 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $FORCE METHOD=SEMINUM NVIB=2 SCLFAC=0.96 VIBANL=.TRUE. $END
 $DATA
Mn(CO)dppe2
C1
Mn
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               7.63915
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 $GUESS GUESS=HUCKEL $END
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Н	1.0	3.53141	12.62798	2.77144
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Н	1.0	3.58926	8.37519	2.03048
Н	1.0	6.16425	6.99596	9.11667
Н	1.0	6.09363	7.50022	11.52913
Н	1.0	4.41078	6.38388	13.00547
Н	1.0	2.79993	4.74016	12.02234
Н	1.0	2.86652	4.22506	9.60229
Н	1.0	3.20372	3.37906	5.34251
Н	1.0	3.05441	0.92580	5.16375
Н	1.0	4.10021	-0.53765	6.90361
Н	1.0	5.30182	0.50358	8.83195
Н	1.0	5.43330	2.96959	9.04025
Н	1.0	6.24204	7.88106	1.25935

н	1.0	5.94	1926		4.347	395	-1.1	9923
н	1 0	7 02	2827		2 990	934	06	1460
н	10	7 69	9693		4 067	703 703	27	3133
н	10	9 10	1586		7 990	169	1 6	5568
н	1.0	9.12	1437	1	n 290	10 10	1.0	1691
LI LI	1.0	9.90	1022	1	0.290 7 759	70	2.0	2610
п	1.0	9.0-	1023 1070	1	1 000	010 170	4.5	2040
н	1.0	7.4: C 7		T	1.895	0/8 200	4.2	4020
н	1.0	6.74	1937		9.610	036	4.8	4839
н	1.0	9.2	648		5.141	_34	3.7	4610
Н	1.0	9.95	5751		6.773	326	3.6	6171
Н	1.0	9.51	L734		7.129	918	6.0	9420
Н	1.0	10.40	)675		5.604	-01	5.9	1655
Н	1.0	2.23	3990		5.113	869	6.2	1082
Н	1.0	2.11	L949		5.848	30	7.8	1634
Н	1.0	3.10	0701		7.977	'32	6.9	7058
Н	1.0	1.76	5213		7.540	062	5.9	0377
Н	1.0	5.55	5414		6.788	351	-0.8	5473
\$1	END							
\$1	ECP							
M	N-ECP GEN	10	)	2				
3		d no	tent	ial	-			
0	-10 0000	700 700	, conc	1	357	39144	69	
	-60 29902	287		2	64	64773	89	
	-10 42179	221		2	16	04113	22	
5	10.42170	h a	noto	c ntia	1	05000		
J	2 0000	s-u	pore	ni lu a	L 107	41272	- 15	
	10 25019	000		0 1	107.	41272		
	10.25910	219		1 2		49589	15	
	2/6.93/39	928		2	46.	55683	46	
	241.3174:	342		2	8.	36881	.35	
-	-146.4635:	329		2	_ 7.	/23/4	89	
5		p-d	pote	ntia	1		-	
	5.00000	000		0	80.	04151	.03	
	5.75897	756		1	105.	60436	46	
	285.29186	554		2	40.	83004	66	
	143.42226	547		2	8.	00984	57	
	-88.70318	351		2	7.	33909	28	
P	-ECP GEN	10		2				
5		d	pote	ntia	1		-	
	-10.00000	000		1	462.	12114	23	
	-79.48646	558		2	93.	68637	01	
	-28.36682	251		2	21.	23490	94	
	-9.85775	589		2	6.	33884	15	
	-1.01637	783		2	2	06206	84	
5		s-d	pote	ntia	1		_	
5	3 00000	300	P	0	- 78	08318	23	
	12 010/	154		5 1	52	95762	10	
	150 02501	208		⊥ 2	26	05710 05710	55	
	1 7002 VL	146		د ۲	JU. 11	21611	52	
	22 0207	140 21 2		۲ ک	11. 2	24044	-JJ C1	
~	23.03976	01Z .		2	ے <sub>ا</sub>	0/5/5	υT	
6		p-d	pote	ntia	L 		-	
	5.00000	000		0	75.	16178	80	

(	6.3446507	1	57.4544041
198	8.5585104	2	47.9481748
11	1 1470820	2	18 4588360
11	2011111	2	5 0/1/100
40	0.3944144 C 4402222	2	1 0407507
	0.4483233	Ζ	1.8487307
P-ECP			
P-ECP			
P-ECP			
0-ECP	NONE		
C-ECP	NONE		
C-FCP	NONE		
C-FCP	NONE		
	NONE		
	NONE		
	NONE		
	NONE		
	NONE		
C-ECP	NONE		
C-FCP	NONF		
C-FCP	NONE		

C-ECP	NONE
C-ECP	NONE
H-ECP	NONE
	NUNL

H-ECP	NONE
H-ECP	NONE
\$END	

```
L
    Input file for GAMESS - [Mn(H2)(C0)dppe2]+ total energy
 $CONTRL SCFTYP=RHF RUNTYP=ENERGY EXETYP=RUN MPLEVL=2
    MAXIT=150 ICHARG=1 ECP=READ $END
 $SYSTEM MEMDDI=5000 MWORDS=50 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2 $END
 $DATA
Mn(CO)(H2)dppe2
C1
Mn
      25.0
               7.67730
                            8.22523
                                       -0.12511
Ρ
     15.0
              6.83663
                          10.45490
                                       0.43054
Ρ
     15.0
             10.06420
                           8.69477
                                       0.35876
С
      6.0
              7.81434
                          12.00004
                                      -0.02869
С
      6.0
              8.45704
                          12.76586
                                       0.96929
С
      6.0
              9.17441
                          13.92583
                                       0.62454
С
      6.0
              9.26547
                          14.32884
                                      -0.72140
С
                                      -1.72116
      6.0
              8.64309
                          13.55812
С
                                      -1.37986
      6.0
              7.92416
                          12.39751
С
      6.0
              6.09032
                          10.93832
                                       2.11054
С
      6.0
              5.47288
                          12.20460
                                       2.24797
С
      6.0
              4.86593
                          12.57934
                                       3.45919
С
      6.0
              4.86874
                          11.69562
                                       4.55705
С
              5.48313
                          10.43646
                                       4.43179
      6.0
С
      6.0
              6.08638
                          10.06183
                                       3.21416
С
      6.0
             11.05912
                          9.83221
                                      -0.77533
С
             11.57075
                          11.06415
                                      -0.31746
      6.0
С
      6.0
             12.31376
                          11.89022
                                      -1.18137
С
      6.0
             12.55061
                          11.49776
                                      -2.51139
С
      6.0
             12.02709
                          10.27664
                                      -2.97837
С
                                      -2.11819
      6.0
             11.28000
                           9.45228
С
      6.0
             10.64005
                           9.21270
                                       2.08456
С
                           9.17540
                                       2.43234
      6.0
             12.01118
С
      6.0
             12.42944
                           9.53383
                                       3.72608
С
      6.0
             11.48466
                                       4.68986
                           9.94225
С
      6.0
             10.12176
                           9.99760
                                       4.34669
С
      6.0
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                           9.63487
                                       3.04989
С
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                          10.58247
                                      -0.65027
С
      6.0
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                          7.03561
                                       0.14125
Н
      1.0
              8.38717
                          12.47921
                                       2.01520
Н
              9.65371
                          14.51239
                                       1.40474
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Н
      1.0
              9.81252
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Н
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Н
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                          12.90947
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              4.39684
                          13.55611
                                       3.54631
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      1.0
              4.40213
                          11.98683
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Н
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Н
      1.0
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Н	1.0	11.40396	11.38286	0.70696
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Н	1.0	13.13198	12.13301	-3.17519
Н	1.0	12.19623	9.96741	-4.00694
Н	1.0	10.87809	8.52094	-2.50886
Н	1.0	12.76014	8.88775	1.69797
Н	1.0	13.48610	9.49909	3.97945
Н	1.0	11.80980	10.21800	5.69004
Н	1.0	9.38515	10.32174	5.07802
Н	1.0	8.65448	9.69031	2.79720
Н	1.0	4.76541	11.53268	-0.44573
Н	1.0	5.57637	10.57936	-1.70180
Н	1.0	11.33035	6.96520	-0.90108
Н	1.0	11.89587	7.04588	0.77144
Ρ	15.0	8.51354	5.94307	-0.56221
Р	15.0	5.28721	7.72144	-0.48258
0	8.0	7.91183	9.28937	-2.88259
С	6.0	7.52694	4.44538	0.02702
С	6.0	6.84736	3.61578	-0.89193
С	6.0	6.12196	2.49801	-0.43963
С	6.0	6.05710	2.20175	0.93507
С	6.0	6.71600	3.03656	1.85747
С	6.0	7.44372	4.15268	1.40612
С	6.0	9.23311	5.33421	-2.20904
С	6.0	9.79132	4.03508	-2.27776
С	6.0	10.39128	3.57597	-3.46305
С	6.0	10.44267	4.40832	-4.59982
С	6.0	9.88975	5.69953	-4.53872
С	6.0	9.29064	6.16188	-3.34951
С	6.0	4.32183	6.64454	0.74035
С	6.0	3.82768	5.37662	0.36970
С	6.0	3.11709	4.59018	1.29594
С	6.0	2.89676	5.05564	2.60493
С	6.0	3.39871	6.31543	2.98558
С	6.0	4.10931	7.10199	2.06062
С	6.0	4.67751	7.09314	-2.15619
С	6.0	3.29497	7.09395	-2.45847
С	6.0	2.83863	6.65365	-3.71305
С	6.0	3.75749	6.20138	-4.68230
С	6.0	5.13185	6.18857	-4.38438
С	6.0	5.58691	6.63306	-3.12738
С	6.0	7.84068	8.83405	-1.78076
С	6.0	10.09607	5.84994	0.50388
С	6.0	4.34801	9.39419	-0.33860
Н	1.0	6.89503	3.81985	-1.95823
Н	1.0	5.61639	1.85964	-1.16031
Н	1.0	5.50375	1.33293	1.28270
Н	1.0	6.66727	2.82057	2.92183
Н	1.0	7.94590	4.77831	2.14128
Н	1.0	9.74715	3.37165	-1.41727
Н	1.0	10.81444	2.57527	-3.49980

н	10	10	90583		4 052	206	-5 51695
ц	1.0	10. Q	02120		6 350	152	-5 40007
ш	1.0	9. o	92100		7 162		- 3 10507
п	1.0	o. 2	000100		7.102	245	-3.33038
н	1.0	3.	98149		4.998	362	-0.63624
н	1.0	2.	73790		3.61/	/79	0.99106
Н	1.0	2.	34048		4.449	980	3.31603
Н	1.0	3.	23191		6.688	312	3.99363
Н	1.0	4.	47629		8.074	-04	2.37833
Н	1.0	2.	56722		7.417	<i>'</i> 01	-1.71719
Н	1.0	1.	77373		6.660	)52	-3.93190
Н	1.0	3.	40419		5.864	42	-5.65381
н	1 0	5	84950		5 841	86	-5 12301
 Ц	1.0	5. 6	64020		6 621	02	2 01672
п	1.0	10.	04939		4 905	-92	-2.91075
н	1.0	10.	00000		4.893		0.31441
н	1.0	9.	81518		5.886	15	1.56315
Н	1.0	3.	94833		9.472	299	0.67791
Н	1.0	3.	50022		9.377	783	-1.02964
Н	1.0	7.	33331		8.014	52	1.61113
Н	1.0	7.	62802		7.308	300	1.41007
\$E	END						
\$E	CP						
MN	I-FCP GEN		10	2			
2		- А	notent	-ial	_		
J	10 00000	- u 20	potem	1	257	20144	60
	-10.000000	שט		1 2	557.	59144	09
	-60.299028	57		2	64.	64773	89
_	-10.42178	34		2	_ 16.	09608	33
5		- s-	d pote	entic	1		-
	3.00000	00		0	107.	41272	15
	16.259182	19		1	111.	49589	73
	276.937392	28		2	46.	55683	46
	241.317434	42		2	8.	36881	35
-	-146.463532	29		2	7.	72374	89
5		- n-	d pote	entio	1		_
•	5 00000	20		0	80	04151	03
	5 75807	56		1	105	60/36	46
	205 201061	50		1 7	105.	00700	66
	203.29100	)4 4 7		2	40.	00004	
	143.422264	+/ - 4		2	ð. 7	00984	-57
_	-88.70318	51	-	2	7.	33909	28
P-	-ECP GEN	1	0	2			
5		- d	pote	entic	ıl		-
	-10.00000	00		1	462.	12114	-23
	-79.48646	58		2	93.	68637	01
	-28.36682	51		2	21.	23490	94
	-9.857758	89		2	6.	33884	15
	-1 016378	22		2	2	06206	.84
5	1.010510	55	d not	∟ n+ia	ב. יו	00200	
J	3 00000	30 20		0 0	70	 00210	72
	12 01044			ย 1	(Ö.	0E2C0	10
	12.91041	04 00		1 2	58.	95/68	10
	150.025029	98		2	36.	05/12	55
	71.708314	16		2	11.	24644	53
	23.039702	12		2	2.	67575	61

6	n-d	potential	
Ŭ	5 0000000	0	75 1617880
	5.0000000	1	F7 4544041
10	0.5440507	1	57.4544041
193	8.5585104	Z	47.9481748
11	1.1470820	2	18.4588360
40	0.3944144	2	5.9414190
(	6.4483233	2	1.8487507
P-ECP			
C-ECP	NONE		
C-FCP	NONE		
	NONE		
	NONE		
	NONE		
C-ECP	NONE		
C-FCP	NONE		
	NONE		
	NONE		
	NONE		
C-ECP	NONE		
C-ECP	NONE		
C-ECP	NONE		
H-ECP	NONE		
H-FCP	NONE		
H_FCP	NONE		
	NONE		
	NONE		
H-ECP	NUNE		
H-ECP	NUNE		
H-ECP	NONE		
H-ECP	NONE		

H-ECP	NONE
H-ECP	NONE
P-ECP	
P-ECP	
0-ECP	NONE
C-ECP	NONE
C-FCP	NONE
C-FCP	NONE
C-FCP	NONE
	NONE
C-FCP	NONE
	NONE
C-FCP	NONE
H-FCP	NONE
H-FCP	NONE
H_FCP	NONE
H-FCP	NONF
H-FCP	NONF
H-FCP	NONE
H-FCP	NONE
H-FCP	NONF
H-FCP	NONE
H-FCP	NONE
H-FCP	NONF

H-ECP	NONE	
H-ECP	NONE	
\$END		

```
L
    Input file for GAMESS - [FeHdppe2]+ total energy
 $CONTRL SCFTYP=RHF RUNTYP=ENERGY EXETYP=RUN
    MAXIT=200 ICHARG=1 ECP=READ MPLEVL=2 $END
 $SYSTEM MEMDDI=4000 MWORDS=50 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.T. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2 $END
 $DATA
[FeHdppe2]+
C1
Fe
      26.0
               7.17168
                            8.36120
                                        2.15485
Ρ
     15.0
              7.63169
                          10.03577
                                       3.76677
Ρ
     15.0
              6.74479
                           6.62234
                                       3.75390
                          10.13150
Ρ
     15.0
              7.48298
                                       0.56603
Ρ
     15.0
              6.90199
                           6.63269
                                       0.54478
С
      6.0
              7.56996
                          11.76737
                                       1.55802
С
      6.0
              6.94193
                          4.97393
                                       1.49593
Н
                                       0.92891
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              8.02431
                          12.53977
Н
      1.0
              6.55830
                           4.18026
                                       0.84642
Н
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              6.55350
                          12.08928
                                       1.80977
Н
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С
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                           9.78535
                                       5.15714
С
      6.0
              5.41708
                           6.85150
                                       5.07310
С
      6.0
             10.24656
                           9.68658
                                       4.83255
С
      6.0
              4.05199
                           6.83013
                                       4.70696
Н
      1.0
             10.57850
                           9.75114
                                       3.79854
Н
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              3.75373
                           6.63591
                                       3.67845
С
      6.0
                          10.48553
                                      -0.68796
              6.11872
С
      6.0
              8.19918
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                                      -0.78829
С
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                          10.26428
                                      -0.42313
С
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                           6.46508
                                      -0.34510
С
      6.0
                                       6.50626
              8.47746
                           9.67266
С
                                       6.41823
      6.0
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                           7.10802
Н
      1.0
              7.42813
                           9.75344
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Н
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С
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С
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Н
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Н
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                                      -2.46041
С
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С
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                           4.08797
                                       6.18725
Н
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                          13.52104
                                       6.58246
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              8.82847
                                       6.87367
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                                       5.66211
                           4.42663
Н
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                                       5.78957
С
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                          11.29830
                                      -1.16995
С
      6.0
             10.46219
                           5.61962
                                      -1.38116
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6.0	10.80594	9.39066	7.18856
6.0	3.39954	7.29479	7.01234
1.0	11.54769	9.24560	7.97003
1.0	2.62587	7.45763	7.75855
6.0	4 85137	10 40700	4 27969
6.0	9 50658	6 21346	4 40679
1 0	4 71591	9 52263	3 66077
1 0	9 68999	7 03274	3 71935
6.0	8 38810	11 52183	2 83306
6.0	6 00052	5 11208	2.05500
1 0	0.03332	12 20048	2 / 0701
1.0	6 17040	1 22112	2 10750
1.0	0.17040	4.22113	2 50240
1.0	9.45005	LI.20755	2.39240
1.0	5.04403	5.27121	2.52325
6.0	9.27002	11.32606	-1.33870
6.0	4.97831	5.36924	-1.19685
1.0	8.46965	12.03736	-1.53158
1.0	5.74679	4.62617	-1.39961
6.0	11.20621	9.49627	5.84253
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1.0	12.25963	9.43363	5.58015
1.0	2.00325	7.01445	5.37700
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6.0	10.20389	5.89668	-2.73773
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6.0	4.90887	11.07736	-0.25892
6.0	9.46885	5.84593	-0.41173
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1.0	9.69639	5.63926	0.63164
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6.0	4.75773	7.32504	7.38124
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6.0	3.87674	12.18593	5.63306
6.0	10.34552	4.48964	5.91799
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6.0	4.23828	7.41951	-0.11071
1.0	9.98747	8.53741	0.50870
	$\begin{array}{c} 1.0\\ 1.0\\ 6.0\\ 1.0\\ 1.0\\ 1.0\\ 1.0\\ 1.0\\ 1.0\\ 1.0\\ 1$	1.0 $2.93994$ $1.0$ $11.42992$ $6.0$ $3.39954$ $1.0$ $11.54769$ $1.0$ $2.62587$ $6.0$ $4.85137$ $6.0$ $9.50658$ $1.0$ $4.71591$ $1.0$ $9.68999$ $6.0$ $8.38819$ $6.0$ $6.09952$ $1.0$ $8.37847$ $1.0$ $6.17048$ $1.0$ $9.43063$ $1.0$ $5.04403$ $6.0$ $9.27002$ $6.0$ $4.97831$ $1.0$ $8.46965$ $1.0$ $5.74679$ $6.0$ $11.20621$ $6.0$ $3.04974$ $1.0$ $12.25963$ $1.0$ $2.00325$ $6.0$ $4.00109$ $6.0$ $10.20389$ $1.0$ $3.19244$ $1.0$ $10.97056$ $6.0$ $4.90887$ $6.0$ $9.43955$ $6.0$ $4.75773$ $1.0$ $9.12286$ $1.0$ $5.03713$ $6.0$ $9.43955$ $6.0$ $4.75773$ $1.0$ $9.12286$ $1.0$ $5.19974$ $6.0$ $10.34552$ $1.0$ $3.71907$ $6.0$ $10.58435$ $1.0$ $2.72441$ $1.0$ $1.60286$ $6.0$ $10.13014$ $6.0$ $4.23828$ $1.0$ $9.98747$	1.0 $2.93994$ $11.76946$ $1.0$ $11.42992$ $5.22616$ $6.0$ $10.80594$ $9.39066$ $6.0$ $3.39954$ $7.29479$ $1.0$ $11.54769$ $9.24560$ $1.0$ $2.62587$ $7.45763$ $6.0$ $4.85137$ $10.40700$ $6.0$ $9.50658$ $6.21346$ $1.0$ $4.71591$ $9.52263$ $1.0$ $9.68999$ $7.03274$ $6.0$ $8.38819$ $11.52183$ $6.0$ $6.09952$ $5.11298$ $1.0$ $8.37847$ $12.39948$ $1.0$ $6.17048$ $4.22113$ $1.0$ $9.43063$ $11.28733$ $1.0$ $5.04403$ $5.27121$ $6.0$ $9.27002$ $11.32606$ $6.0$ $4.97831$ $5.36924$ $1.0$ $8.46965$ $12.03736$ $1.0$ $5.74679$ $4.62617$ $6.0$ $11.20621$ $9.49627$ $6.0$ $3.04974$ $7.04665$ $1.0$ $12.25963$ $9.43363$ $1.0$ $2.00325$ $7.01445$ $6.0$ $4.00109$ $10.92016$ $6.0$ $4.0019$ $10.92016$ $6.0$ $4.90887$ $11.07736$ $6.0$ $9.43955$ $9.47623$ $6.0$ $4.75773$ $7.32504$ $1.0$ $9.69639$ $5.63926$ $6.0$ $4.75773$ $7.32504$ $1.0$ $9.43955$ $9.47623$ $6.0$ $4.03452$ $4.48964$ $1.0$ $3.00441$ $12.69072$ $1.0$

н	10	44	3571		8 282	214	0 52127
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C	6.0	0.1	C110		2.014	140	4.07070
C	6.0	11.3	5621		9.49/	′49	-0.8/106
C	6.0	2.9	7261		7.285	335	-0.71356
Н	1.0	12.1	5776		8.785	546	-0.69024
Н	1.0	2.2	0434		8.030	)91	-0.53269
С	6.0	10.4	9353	1	1.466	572	-2.01671
С	6.0	3.7	1579		5.233	318	-1.80054
н	1.0	10.6	3049	1	2.284	19	-2.72031
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c	6.0	11 5	4150	1	0 552	27	_1 78279
c	6.0	2 7	0020	-	6 190	27/	1 55942
	0.0	12 1	0025	1	0.10		-1.33642
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Н	1.0	8.6	3123		8.027	756	2.14222
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\$E	ECP						
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	-63 26675	18		2	71	17569	79
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F	-10.90133.	20		ے نے بے	⊥/. .1	13202	01
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	3.00000	00		0	126.	.05718	95
	18.172913	37		1	138.	12642	51
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	317.106803	12		2	9.	28379	66
-	207.342164	19		2	8.	62890	82
5		- p-d	pote	entic	ıl		-
	5.00000	00	•	0	83.	17594	90
	5,953593	30		1	106.	05599	38
	294 266552	77		2	42	82849	37
	154 12116	-'		2	2	77018	05
	05 21642	10		2	0. 0	02070	10
Б	-93.510424	+9 10		2	٥.	01650	10
P-	ECP GEN	. 10		Ζ	-		
5		- d	pote	entic	1		-
	-10.00000	00		1	462.	12114	23
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	12 91041	54		1	58	95768	10
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L
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    MAXIT=200 ICHARG=1 ECP=READ $END
 $SYSTEM MEMDDI=5000 MWORDS=75 PARALL=.TRUE. $END
 $BASIS GBASIS=LANL2DZ EXTFIL=.T. NDFUNC=1 $END
 $GUESS GUESS=HUCKEL $END
 $SCF DIRSCF=.TRUE. CONV=5.0e-6 DAMP=.T. $END
 $STATPT OPTTOL=0.0001 NSTEP=100 TRMIN=0.01 $END
 $FORCE METHOD=SEMINUM VIBSIZ=0.010000 SCLFAC=0.960000 NVIB=2 $END
 $DATA
[FeH(H2)dppe2]+
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                            8.39792
                                        2.15327
Ρ
     15.0
              7.54614
                          10.05383
                                       3.76618
Ρ
     15.0
              6.64272
                           6.67353
                                       3.76011
                                       0.54925
Ρ
     15.0
              7.35390
                          10.12366
Ρ
     15.0
              6.83222
                          6.64351
                                       0.54574
С
      6.0
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                          11.75571
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С
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                           5.00383
                                       1.53105
Н
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Н
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Н
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С
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С
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С
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С
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С
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                                       6.46570
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Н
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С	6.0	3,39033	7,29636	7.12061
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Н	1.0	2.63583	7.44472	7.88914
c	6.0	4 85498	10 53670	4 68639
c	6.0	9 42735	6 22956	4 28478
н	1 0	4 58614	9 63399	4 14808
н	1 0	9 58976	7 04663	3 58961
Ċ	6.0	8 29618	11 50808	2 78266
c	6.0	5 07585	5 16074	2 78057
с u	1 0	9 22597	12 20520	2 12171
п u	1.0	6 02400	1 29176	2 12917
п U	1.0	0.02400	4.20170	2.42047 2.50021
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C	6.0	11.21902	9.42856	5.66498
C	6.0	3.00660	7.08846	5.78138
Н	1.0	12.26129	9.36/15	5.36108
Н	1.0	1.95332	7.07367	5.51133
C	6.0	3.96273	10.92987	-2.62907
C	6.0	10.36154	5.92608	-2.50007
Н	1.0	3.17409	11.10953	-3.35540
Н	1.0	11.17594	5.75047	-3.19852
С	6.0	4.81120	11.08016	-0.34577
С	6.0	9.47738	5.87812	-0.22596
Н	1.0	4.64948	11.38176	0.68722
Н	1.0	9.63741	5.67683	0.83076
С	6.0	9.51965	9.38033	7.40529
С	6.0	4.75694	7.30474	7.45769
Н	1.0	9.24212	9.28092	8.45197
Н	1.0	5.06197	7.46264	8.48937
С	6.0	4.17232	12.41216	6.09076
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Н	1.0	3.39617	12.96740	6.61181
Н	1.0	11.16096	3.96629	6.18674
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С	6.0	9.12777	6.42196	-2.96087
Н	1.0	5.31265	10.02769	-4.06438
Н	1.0	8.98634	6.63413	-4.01795
С	6.0	3.84811	11.25267	5.36390
С	6.0	10.52446	5.55153	4.85142
Н	1.0	2.82038	10.89992	5.32751
Н	1.0	11.53631	5.85452	4.59285
С	6.0	10.03728	9.37060	-0.10588
С	6.0	4.21477	7.30932	-0.47990
Н	1.0	9.88499	8.56838	0.60859

Н	1.0	4.300	)44	8.	2307	<b>'</b> 7	0.08649	
С	6.0	6.195	532	10.	9695	57	4.72772	
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С	6.0	11.282	280	9.	5227	<b>'</b> 7	-0.74613	
С	6.0	3.035	553	7.	0645	6	-1.21190	1
Н	1.0	12.089	956	8.	8282	24	-0.52513	
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С	6.0	10.425	546	11.	4514	1	-1.96198	
С	6.0	3.970	)94	4.	9661	9	-2.01620	1
Н	1.0	10.571	L47	12.	2570	9	-2.67729	
Н	1.0	3.886	537	4.	0606	54	-2.61205	
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